Magnetodielectric effect in EuZrO₃

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Following recent report on antiferromagnetic ordering in EuZrO₃ we performed dielectric measurements of this material as a function of temperature and magnetic field. Dielectric constant of dense EuZrO₃ ceramics is 30.1 at 300 K. It gradually decreases upon cooling without any quantum paraelectric behavior; however, below T_N =4.1 K it shows a pronounced drop that qualitatively resembles that observed in EuTiO₃. We report that dielectric constant of EuZrO₃ is magnetic field dependent. The magnitude of the magnetodielectric effect in both EuTiO₃ and EuZrO₃ is discussed in the light of the recently proposed coupling of the Eu–O–Eu superexchange interactions with electrons involved in partially covalent Ti(Zr)–O bond. © 2010 American Institute of Physics. [doi:10.1063/1.3456730]

Recent interest in magnetoelectric materials has brought to light several compounds containing divalent europium. For example, EuTiO₃, a G-type antiferromagnet below Néel temperature, $T_{\rm N}$ =5.5 K, demonstrates a strong magnetocapacitance effect.^{1,2} In view of very low electrical conductivity of EuTiO₃ at $T \le 60$ K, observed magnetocapacitance is attributed solely to the magnetic field dependence of dielectric constant, $\varepsilon(H)$. Most recently, electric field induced magnetoelectric coupling in EuTiO₃ has been reported by Shvartsman *et al.*³

Although theoretical support for magnetoelectric coupling in EuTiO₃ is abundant,⁴ a direct proof of magnetic dependence of the optical phonon mode(s) has yet to be found. Recent analysis of the infrared active phonon modes in EuTiO₃ indicates that, similar to SrTiO₃, the (soft) phonon mode that contributes the largest dielectric weight to the phonon spectrum and presumably is responsible for the magnetoelectric effect, is Slater-type mode.⁵ Remarkably, this mode does not involve magnetically active (Eu²⁺) ions, as opposed to the Last-type mode where Eu²⁺ ions vibrate against the TiO₆ octahedra. According to Cohen,⁶ the properties of the Slater-type mode in the ABO_3 perovskites with A=Sr, Ba, and Pb and B=Ti are largely dependent on the overlap between the O 2p and B 3d orbitals, i.e., on the degree of covalency of the B-O bond. Recently, Goian *et al.*,⁵ have suggested that the magnetoelectric effect in EuTiO₃ can be understood in terms of the impact of Eu-O-Eu superexchange on the electron density involved in (partially) covalent Ti-O bonding. One way to examine this hypothesis is to find another $Eu^{2+}B^{4+}O_3$ perovskite with similar magnetic structure but different degree of covalency of the B-O bond.

Here we report on synthesis, crystal structure, and physical properties of EuZrO₃ perovskite with a special emphasis on its magnetodielectric effect. Until recently, EuZrO₃ was considered paramagnetic with a cubic (space group $Pm\bar{3}m$)

perovskite structure.⁷ Recently Viallet *et al.*⁸ and Zong *et al.*⁹ have revised the crystal structure of EuZrO₃ and found that it crystallizes in orthorhombic (space group Pbnm) symmetry similar to that of SrZrO₃. Most importantly, Zong et al. found that $EuZrO_3$ develops an antiferromagnetic (AFM) ordering below $T_N \approx 4.1$ K. Although direct confirmation of the type of the AFM structure in EuZrO₃ has yet to be provided by neutron diffraction, the indirect evidence indicates that EuZrO₃ and EuTiO₃ are magnetically very similar. In particular, analysis of the sign and magnitude of the magnetic exchange constants, J_i , between nearest, J_1 , and nextnearest, J_2 , Eu²⁺ ions in EuZrO₃ (Ref. 9) favors the G-type AFM structure, similar to that found in EuTiO₃.^{10,11} On the other hand, one may expect the valence electronic structure of these two perovskites to be quite different. Namely, in contrast to partially covalent Ti-O bond in EuTiO₃, the Zr-O bond in EuZrO₃ is more ionic, similar to the case of SrZrO₃. This makes EuZrO₃ an important reference material for understanding of the magnetodielectric effect in Eu²⁺-based perovskites.

 $EuZrO_3$ was prepared from Eu_2O_3 and ZrO_2 (both 99.9%) pure, sourced from Kanto Chemicals, Japan). The stoichiometric mixture of Eu₂O₃ and ZrO₂ was treated at 1300 °C for 20 h in pure hydrogen at a flow rate of 80 $\text{ cm}^3/\text{min}$, with intermediate regrinding until a yellow, single phase EuZrO₃ product was obtained. Phase purity was confirmed by powder x-ray diffraction (Rigaku Ultima III x-ray diffractometer, $\lambda = 0.15406$ nm radiation). Lattice parameters were obtained from Rietveld refinement of the x-ray data using RIETAN 2000. Magnetic susceptibility in the 2-100 K range was measured using superconducting quantum interference devise (Quantum Design, MPMS, USA). EuZrO₃ ceramic with 98 % density was obtained by spark plasma sintering (SPS) (Dr. Sinter, SPS-1050, Japan) at 1450 °C under 130 MPa uniaxial pressure and base vacuum pressure of 2×10^{-2} Pa.¹² To release the residual lattice strain after SPS, the samples were annealed at 1300 °C in H₂ gas for 20 h. For dielectric measurements, Au electrodes were spattered

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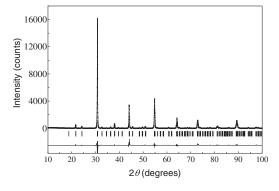


FIG. 1. Room temperature x-ray diffraction pattern of EuZrO₃ (+). Calculated diffraction pattern from Rietveld refinement of EuZrO₃ (solid line). The vertical bars indicate the positions of expected Bragg peaks. The difference between observed and calculated data is shown at the bottom of the plot.

on the EuZrO₃ disks of 7 mm diameter and 1 mm thickness. Dielectric properties were measured with Agilent E4980 Precision LCR Meter in the temperature interval of 2.5–300 K and magnetic field of 0–80 kOe utilizing a home-made dielectric measurements probe coupled with the Physical Property Measurement System (QD, USA).

Powder x-ray analysis (Fig. 1) has confirmed orthorhombic (space group *Pbnm*) symmetry of EuZrO₃ with lattice constants a=5.8019(4), b=5.8236(5), and c=8.2051(9) Å. The x-ray pattern was refined to the reliability factor R_p =11.2% in close agreement with Ref. 8. According to the lattice parameters, the absolute density of EuZrO₃ is 6.976 g cm⁻³.

Temperature dependence of magnetic susceptibility, χ , measured at H=50 Oe and shown in Fig. 2 indicates an AFM phase transition at $T_N \approx 4.1$ K. Linear fit of the $\chi^{-1}(T)$ dependence yields a Weiss temperature $\theta_W \approx +0.1$ K. According to Ref. 9, positive θ_W is a rare case that indicates coexistence of both ferromagnetic and AFM interactions in EuZrO₃. Remarkably, positive $\theta_W \approx +3.3$ K has also been found in EuTiO₃, thus supporting similar G-type AFM magnetic structure.

Since EuZrO₃ and SrZrO₃ are structurally identical with the difference in the lattice constants of $\approx 0.04\%$,¹³ one may expect similar dielectric properties. Indeed, room tempera-

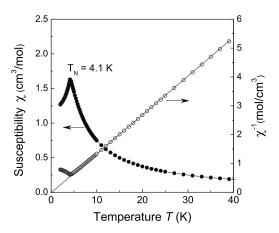


FIG. 2. Temperature dependence of zero field cooled molar magnetic susceptibility, χ , and χ^{-1} of EuZrO₃. The AFM transition at $T_N \approx 4.1$ K is detected from the $\chi(T)$ anomaly. Linear fit of the $\chi^{-1}(T)$ dependence yields a positive Weiss temperature, $\theta_W \approx +0.1$ K.

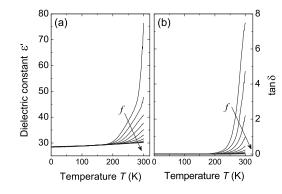


FIG. 3. Temperature dependence of (a) dielectric constant, ε' and (b) dielectric loss, tan δ , of EuZrO₃ at selected frequencies *f*=0.1, 0.3, 0.7, 1.7, 4.4, 11.3, 29.2, 75.3, 194, and 500 kHz.

ture ε' of EuZrO₃ determined in this work and that of SrZrO₃ reported in Ref. 13 are 30.1 and 30.4, respectively. Dielectric constant of EuZrO₃ is weakly temperature dependent [Fig. 3(a)] with $d\varepsilon'/dT \approx 6.8 \times 10^{-3}$ K⁻¹. The ε' gradually decreases from 30.1 at 300 K to 28.45 at 5 K with no indication of the quantum paraelectric behavior. This supports our original assumption of the predominantly ionic Zr–O bond which does not favor formation of the low-energy (soft) phonon mode in EuZrO₃. Above 175 K, ε' shows low-frequency dispersion which, we believe, is attributed to the finite electronic conductivity also evidenced from the large tan δ values [Fig. 3(b)]. From the linear tan δ vs. ω^{-1} dependence (not shown) we conclude that above 175 K the imaginary part of the dielectric constant is dominated by the electronic conductivity, σ' , according to

$$\varepsilon^* = \varepsilon' + i \frac{\sigma'}{\varepsilon_0 \omega},\tag{1}$$

where ε_0 is the electric constant and ω is the angular frequency, $\omega = 2\pi f$.

The low-temperature behavior of ε' is shown in Fig. 4(a) as a function of applied magnetic field. Dielectric constant of EuZrO₃ at H=0 demonstrates an anomalous drop below $T \approx 4$ K, the temperature corresponding to the AFM phase transition. Pronounced change in the $\varepsilon'(T)$ dependence is found upon application of magnetic field. Low-*T* dielectric constant increases with magnetic field and saturates at $\varepsilon' \approx 28.48$ for $H \ge 30$ kOe. Qualitatively, magnetodielectric effect in EuZrO₃ is similar to that reported for EuTiO₃. On the quantitative scale, the magnitude of the magnetodielectric effects in EuZrO₃ and EuTiO₃ is quite different. Katsufuji *et al.*¹ proposed a simple equation to describe the magnetodielectric effect as follows:

$$\varepsilon'(H) = \varepsilon'(H = 0)(1 + \alpha \langle \mathbf{S}_i \cdot \mathbf{S}_i \rangle), \qquad (2)$$

where $\varepsilon'(H=0)$ is dielectric constant at zero magnetic field, $\langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle$ is the spin pair correlation between nearest neighbor Eu²⁺, and α is the coupling constant between spin correlation and dielectric constant. In order to estimate $\langle \mathbf{S}_i \cdot \mathbf{S}_j \rangle$ in EuZrO₃, we have performed a mean-field calculation based on a Heisenberg model under an assumption that the 4*f* spins of Eu²⁺(S=7/2) are located on a simple cubic lattice with a G-type AFM ordering.⁹ The Heisenberg Hamiltonian of the spin system, with restriction to the nearest-neighbor and next-nearest-neighbor interactions, is given by

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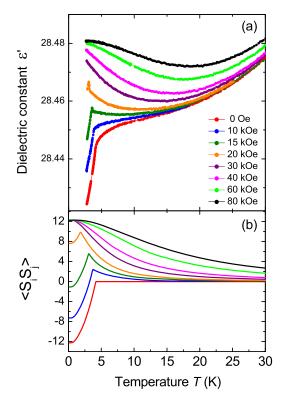


FIG. 4. (Color online) (a) Low-temperature behavior of dielectric constant of $EuZrO_3$ measured at 200 kHz in magnetic field of 0, 10, 15, 20, 30, 40, 60, and 80 kOe. (b) Spin correlation function for cubic $EuZrO_3$ calculated using a mean-field approximation.

$$H = -2J_1 \sum_{\langle i,j \rangle} \mathbf{S}_i \cdot \mathbf{S}_j - 2J_2 \sum_{\{i,j\}} \mathbf{S}_i \cdot \mathbf{S}_j,$$
(3)

where \mathbf{S}_i is the operator of the 4*f* spin localized at the site *i*. The sums of the first and the second terms run over the nearest-neighbor $\langle i, j \rangle$, and the next-nearest-neighbor pairs $\{i, j\}$, respectively. In our calculation, we used $J_1/k_B = -0.032$ K and $J_2/k_B = 0.017$ K which yield $T_N = 4.1$ K and $\theta_W = 0.1$ K, where k_B is the Boltzmann constant. Figure 4(b) shows temperature dependence of $\langle \mathbf{S}_i, \mathbf{S}_j \rangle$ calcu-

lated from the product of $\langle S \rangle$ on the two sublattices under different *H*.

From the data given in Figs. 4(a) and 4(b) and using Eq. (2), one finds for EuZrO₃ the magnetodielectric coupling constant $\alpha \approx 1.1 \times 10^{-4}$, which is a factor of 27 lower than that of EuTiO₃ (i.e., $\alpha \approx 2.74 \times 10^{-3}$).¹

In conclusion, at low T, EuZrO₃ shows magnetodielectric effect whose features are qualitatively similar to that reported for EuTiO₃, albeit the former perovskite is not a quantum paraelectric. That said, the magnitude of the magnetodielectric coupling in EuZrO₃ is much smaller than that of EuTiO₃. In accord with the hypothesis of Goian *et al.*,⁵ we attribute this difference to stronger covalency of the Ti–O bond in EuTiO₃ which favors formation of the soft phonon mode which, in turn, facilitates coupling with the low-energy magnetic excitations.

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- ¹T. Katsufuji and H. Takagi, Phys. Rev. B **64**, 054415 (2001).
- ²S. Kamba, D. Nuzhnyy, P. Vaněk, M. Savinov, K. Knízek, Z. Shen, E. Šantavá, K. Maca, M. Sadowski, and J. Petzelt, EPL **80**, 27002 (2007).
 ³V. V. Shvartsman, P. Borisov, W. Kleemann, S. Kamba, and T. Katsufuji,
- V. V. Snvartsman, P. Borisov, W. Kleemann, S. Kamba, and I. Katsuruji, Phys. Rev. B **81**, 064426 (2010).
- ⁴H. Wu, Q. Jiang, and W. Z. Shen, Phys. Rev. B **69**, 014104 (2004); R. Ranjan, H. S. Nabi, and R. Pentcheva, J. Phys.: Condens. Matter **19**, 406217 (2007).
- ⁵V. Goian, S. Kamba, J. Hlinka, P. Vaněk, A. A. Belik, T. Kolodiazhnyi, and J. Petzelt, Eur. Phys. J. B **71**, 429 (2009).
- ⁶R. E. Cohen, Nature (London) **358**, 136 (1992).
- ⁷A. H. Shafer, J. Appl. Phys. **36**, 1145 (1965).
- ⁸V. Viallet, J.-F. Marucco, J. Saint, M. Herbst-Ghysel, and N. Dragoe, J. Alloys Compd. **461**, 346 (2008).
- ⁹Y. Zong, K. Fujita, H. Akamatsu, S. Murai, and K. Tanaka, J. Solid State Chem. 183, 168 (2010).
- ¹⁰T. R. McGuire, M. W. Shafer, R. J. Joenk, H. A. Alperin, and S. J. Pickart, J. Appl. Phys. **37**, 981 (1966).
- ¹¹J. E. Greedan, C. L. Chien, and R. G. Johnston, J. Solid State Chem. **19**, 155 (1976).
- ¹²S. Grasso, Y. Sakka, and G. Maizza, Sci. Technol. Adv. Mater. **10**, 053001 (2009).
- ¹³D. Souptel, G. Behr, and A. M. Balbashov, J. Cryst. Growth **236**, 583 (2002).