Electrochemical properties of lead fluoride electrode in fluoride shuttle battery

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Abstract

Recently, the reversible charge-discharge reaction of BiF₃ electrode for fluoride shuttle battery (FSB) that can be used as a promising candidate for next-generation battery are observed using a liquid-based electrolyte. In this study, we investigate the electrochemical performance of PbF₂ as an active material for the FSB. To increase the electronic conductivity, the PbF₂ was mixed with carbon, and the composite material between PbF₂ and carbon, PbF₂/C, is formed. High charge and discharge capacities are obtained for PbF₂/C during the first cycle. Although the charge and discharge capacities gradually decreased, the charge-discharge reactions occurred in the second and third cycles. To confirm the progress of the charge-discharge reactions, the crystal structure change of the active material during charging and discharging in the first and second cycles is evaluated using X-ray diffraction (XRD). From the XRD results, the formation of Pb and PbF₂ during discharging and charging can be confirmed, indicating that the discharge (PbF₂ + 2e⁻ \rightarrow Pb + 2F⁻) and charge (Pb + 2F⁻ \rightarrow PbF₂ + 2e⁻) reactions progress in both the first and second cycles.

1. Introduction

Lithium-ion batteries (LIBs) are extensively used as secondary batteries for many electronic devices [1-3]. To increase the use of secondary batteries, many research institutions have investigated a variety of next-generation high-energy-density batteries such as sodium, magnesium, potassium, lithium/oxygen batteries [4–7]. Recently, batteries that utilize anion migration to perform the electrochemical reaction have been developed [8–15]. In 2011, Reddy et al. reported a new all-solid battery which utilize the migration of fluoride ions [8]. For this battery, the metal fluoride and metal were used as the active material, and fluoride ions migrated between metal fluoride and metal through a solid electrolyte. Since this study was conducted, the progress of electrochemical reaction of several metal fluorides using a solid electrolyte has been reported [8-11]. Furthermore, the battery using a liquid-based electrolyte has also been researched [12, 13]. Our group has devoted attention to a battery that utilizes the fluoride-ion shuttling in a liquid-based electrolyte [16, 17]. However, the supporting electrolyte salt such as cesium fluoride (CsF) hardly dissolved in the organic solvent [17]. To increase the solubility of CsF in the organic solvent, anion acceptor (AA) was added in the organic solvent [17]. Using the liquid-based electrolyte comprising organic solvent (bis[2-(2-methoxyethoxy)ethyl] ether (tetraglyme: G4)), supporting electrolyte salt (CsF), and AA (fluorobis(2,4,6-trimethylphenyl)borane (FBTMPhB)), the discharge and charge reactions of BiF₃ successfully progressed, and the progress of both the reactions were confirmed using X-ray diffraction (XRD). Using the XRD results, the

discharge reaction of BiF₃ was observed to completely progress; however, the charge reaction progressed partially [17]. To obtain high capacities during both charging and discharging, high ionic and electronic conductivities are required for the active material. The PbF₂-based compounds exhibit fluoride ion conductivity [18–21]; therefore, PbF₂ was selected as an active material for the fluoride shuttle battery (FSB). It has been reported that the high adhesion state with carbon improved the charge and discharge capacities of metal fluorides for the active materials of LIBs due to the increased electronic conductivity [22–32]. This method can be used for the active material of the FSB. Thus, in the current study, the composite electrode of PbF₂ and carbon, PbF₂/C, was formed using a ball mill (FRITSCH; PULVERISETTE 7). The electrochemical performance of PbF₂/C was evaluated using the electrolyte comprising of G4, CsF, and FBTMPhB. Furthermore, to confirm the progress of the charge-discharge reactions, the crystal structure change of the active material was confirmed using XRD.

2. Experimental

2.1. Preparation of the electrolyte

The CsF (Tokyo Chemical Industry Co., Ltd), FBTMPhB (Tokyo Chemical Industry Co., Ltd), and G4 (KISHIDA CHEMICAL Co., Ltd) were used as the fluoride salt, the AA, and the organic solvent, respectively. The G4 containing 0.45 mol dm⁻³ CsF and 0.5 mol dm⁻³ FBTMPhB was used

as the electrolyte. The electrolyte was prepared in a glove box under high purity argon.

2.2. Preparation of the active material

The PbF₂ (Aldrich) was used as the active material. This was pulverized for 1 h in a planetary ball mill at 1100 rpm, and was then 80 wt% pulverized PbF₂; further, 20 wt% acetylene black (AB) was mixed for 1 h in the planetary ball mill at 1100 rpm to produce PbF₂/C. The PbF₂ and AB were sealed in the planetary ball mill under a high purity argon atmosphere.

2.3. Characterization of the active material

For PbF₂/C, the crystal structure, particle configuration, and solubility in the electrolyte were investigated by X-ray diffraction (XRD), scanning electron microscopy (SEM), and atomic absorption spectrometry (AAS). The XRD measurements were performed with a Rigaku Smartlab. XRD patterns were recorded in the 2θ range from 20° to 70° using a Rigaku diffractometer with Cu K α radiation (λ = 1.54 Å). The SEM measurements were performed on a HITACHI SU6600 instrument, and the energy of the incident electrons was 20 keV. The AAS measurements were performed using a polarized Zeeman atomic absorption spectrophotometer (Hitachi).

2.4. Characterization of electrolyte

To evaluate the stability of the electrolyte during cycling, cyclic voltammetry (CV) was conducted using a three-electrode electrochemical cell (EC FRONTIER CO., Ltd. VB7) comprising a platinum foil working electrode, a platinum mesh counter electrode, and a silver rod immersed in acetonitrile containing 0.1 mol dm⁻³ silver nitrate and 0.1 mol dm⁻³ tetraethylammonium perchlorate as the reference electrode (0.587 V vs. standard hydrogen electrode) [33]. The CV of the prepared electrolyte was performed at room temperature using a multipotentiostat (Biologic VMP-300) over a -4.0 to 2.0 V (vs. ref.) potential range at a sweep rate of 0.1 mV s⁻¹ during the three cycles. The assembly of electrochemical call and CV measurements were performed in a glove box filled with high purity argon.

2.5. Electrochemical performance of PbF₂/C

Firstly, the PbF₂/C, AB, and polyvinylidene difluoride (75: 10: 15 wt%) were mixed, and Nmethyl-2-pyrrolidone (NMP) was added. The resulting slurry was coated on an Al current collector, and the NMP was evaporated at 110°C. The charge and discharge capacities of the PbF₂/C electrode were measured using a three-electrode electrochemical cell consisting of a PbF₂/C electrode as the working electrode, a platinum mesh as the counter electrode, and a silver rod immersed in acetonitrile containing 0.1 mol dm⁻³ silver nitrate and 0.1 mol dm⁻³ tetraethylammonium perchlorate as the reference electrode. The charge and discharge measurements were performed using a multipotentiostat (Biologic VMP-300) within a potential range from -2.0 to -0.6 V (vs. ref.) at 0.025C ($\sim 20 \ \mu A \ cm^{-2}$) and room temperature in an argon atmosphere. The specific capacity was obtained by dividing the capacity with the weight of the active material (PbF₂).

2.6. Crystal structure of PbF₂/C in the pristine and during the charge-discharge cycle

The crystal structures of PbF₂/C during the charge-discharge processes in the first and second cycles were evaluated using XRD. After the PbF₂/C electrodes were charged and discharged, they were washed with G4 followed by dimethyl carbonate to eliminate the residual electrolyte. The XRD patterns were recorded in the 2θ range from 25° to 45°.

2.7. Dissolution amount of Pb in the electrolyte during the charge-discharge processes

The dissolution amount of Pb in the electrolyte in the first fully charged and discharged states was measured by AAS. After the PbF_2/C electrode was charged and discharged, the electrochemical cell was disassembled, and the electrolyte was recovered from the cell. The concentration of Pb in the electrolyte was measured.

3. Results and discussions

3.1. Characterization of PbF_2/C

The XRD pattern of PbF₂/C powder is illustrated in Fig. 1. All the diffraction peaks are assigned to the cubic phase with space group Fm3m. The SEM images of PbF₂ and PbF₂/C powders are depicted in Fig. 2. The particle size of PbF₂ (before mixing with AB) is 0.5–2 µm (Fig. 2(a)). After mixing with AB, smaller particles of AB are observed on the PbF₂ particle surfaces (Fig. 2(b)). This indicates that AB particles are adhered to the surface of the PbF₂ particles after mixing.

3.2 Stability of electrolyte during the cycling

To evaluate the stability of the electrolyte during cycling, CV was performed. The CVs of the prepared electrolyte during the first, second, and third cycles are depicted in Fig. 3. There was a large reduction peak below -2.2 V (vs. ref.) and a small oxidation peak between -0.3 and 0.5 V (vs. ref.) during the first cycle. The shapes of the CVs during the second and third cycles are similar to that during the first cycle. This indicates that the prepared electrolyte could be used repeatedly between -2.2 and -0.3 V (vs. ref.).

3.3. Electrochemical performance of PbF₂/C

The charge and discharge curves of PbF_2/C are depicted in Fig. 4. The charge and discharge capacities of PbF_2/C during the first cycle were 175 and 240 mAh g⁻¹, respectively. The discharge capacity exceeds the theoretical capacity (219 mAh g⁻¹), which indicates that side reaction, such as

decomposition of the electrolyte, occurs during discharging. The charge/discharge capacities during the second and third cycles are 128/183 and 88/124 mAh g⁻¹, respectively. Although the charge/discharge capacities gradually decreased during cycling, the charge/discharge reactions progressed during the second and third cycles. Furthermore, the shapes of the charge/discharge curves changed from the first to second cycle. The discharge onset potential changed from -1.05 (first cycle) to -1.35 V (second cycle). Charging started at -1.35 V during the first, second, and third cycles. In contrast, a plateau was observed at -0.9 V during the first cycle; however, the charge potential gradually increased during the second and third cycles.

3.4. Confirmation of progress of the charge and discharge reactions

To confirm the progress of the charge and discharge reactions, the crystal structures of the PbF₂ electrode in the charged and discharged states were confirmed using XRD. The XRD patterns of the PbF₂/C electrode during charging and discharging in the first cycle are depicted in Fig. 5. In the pristine state, all the diffraction peaks are indexed to PbF₂. The intensities of peaks assigned to PbF₂ decreased, and new peaks appeared at $2\theta = 31.3^{\circ}$ and 36.3° in the partially discharged state (100 mAh g⁻¹). These peaks are assigned to Pb metal (cubic phase). As discharging progresses, the peaks assigned to PbF₂ vanish and the intensities of peaks assigned to Pb increase. These results indicate that the discharge reaction (PbF₂ + 2e⁻ \rightarrow Pb + 2F⁻) progressed. Conversely, the intensities of peaks assigned

to Pb decrease and those assigned to PbF₂ increase in the partially charged state (100 mAh g⁻¹). As charging processes, the peaks assigned to Pb vanish and the intensities of peaks assigned to PbF₂ increase, indicating that the charge reaction (Pb + 2F⁻ \rightarrow PbF₂ + 2e⁻) progressed. The XRD results indicate that PbF₂ and Pb are formed during charging and discharging, respectively.

The crystal structures of the PbF₂/C electrode in the fully charged and discharged states during the second cycle are also evaluated using XRD, and the results are presented in Fig. 6. The peaks assigned to the Pb metal were observed in the fully discharged state. Further, the peaks in the fully charged states are indexed to PbF₂. These results indicate that Pb was formed during discharging and that PbF₂ was formed during charging in the second cycle. The XRD results indicate that the charge and discharge reactions progressed during the second cycle.

3.5. Mechanism of capacity degradation during cycling

The XRD results for both the first and second cycles (Figs. 5 and 6) indicate that the charge and discharge reactions progressed completely during both the cycles. However, the charge and discharge capacities gradually decreased during cycling. Furthermore, the dissolution of active material was employed in the electrochemical reaction. This means that the active material could be dissolved in the electrolyte. Thus, we expected that the active material may be dissolved in the electrolyte and be lost from the electrode. Further, the dissolution amount of Pb in the electrolyte during charging and discharging was evaluated by AAS. The dissolution amount of Pb in the electrolyte in the fully charged and discharged states during the first cycle is presented in Table 1. The dissolution amount of Pb considerably increased from the fully discharged (1%) to the fully charged states (13%). This indicates that Pb dissolved slightly during discharging. In contrast, the dissolution of Pb progressed during charging. It can therefore be concluded that the dissolution of Pb progress during charging to following reaction:

 $Pb \rightarrow Pb^{2+} + 2e^{-}$

The progress of above reaction causes the loss of active material from the electrode. The content of active material in the electrode directly affected the capacity; therefore, we attributed the capacity degradation to the loss of active material from the electrode. The progress of Pb dissolution might be influenced by the electrolyte composition; therefore, the type and concentration of AA should be investigated to suppress the dissolution of Pb. It will improve the cycling performance of FSB.

4. Conclusion

The electrochemical properties of PbF₂/C for the FSB were investigated using the developed electrolyte. The electrochemical reactions of the PbF₂/C electrode were observed during the charge-discharge cycling. Although the charge-discharge capacities gradually decreased during the cycling, the charge-discharge reactions were observed in the following cycles. To confirm the progress of the

charge and discharge reactions, the crystal structures of the PbF₂/C electrode during charging and discharging in both the first and second cycles were evaluated using XRD. The XRD results indicated the complete formation of PbF₂ and Pb during charging and discharging, respectively, in both the first and second cycles. The AAS results indicated that the active material was partially lost after charging. Therefore, we attributed the capacity degradation to the loss of active material due to the dissolution of Pb.

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Table

 Table 1
 Dissolution amount of Pb in the electrolyte in the fully charged and discharged states during

the first cycle.

State	Dissolution amount of Pb (%)
Completely discharged	1
Completely charged	13

Figure



Figure 1 XRD patterns of the PbF_2/C powder.



Figure 2 SEM images of the (a) PbF_2 and (b) PbF_2/C powders.



Figure 3 Cyclic voltammograms of the prepared electrolytes over the -4.0 to 2.0 V (vs. ref.) potential range during the first (black), second (red), and third (blue) cycles [The data were reproduced using the data of ref.[17]].



Figure 4 Charge and discharge curves of PbF_2/C at 0.025C during the first (black), second (red), and

third cycles within the potential range from -2.0 to -0.6 V (vs. ref.).



Figure 5 XRD patterns of PbF_2/C in the pristine, partially discharged (100 mAh g⁻¹), fully discharged,

partially charged (100 mAh g⁻¹), and fully charged states during the first cycle.



Figure 6 XRD patterns of PbF₂/C in the fully charged and discharged states during the second cycle.