

# **Mechanism of reductive fluorination by PTFE-decomposition fluorocarbon gases for WO<sub>3</sub>**

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## ABSTRACT.

Reductive fluorination, which entails the substitution of  $O^{2-}$  from oxide compounds with  $F^-$  from fluoropolymers, is considered a practical approach for preparing transition-metal oxyfluorides. However, the current understanding of the fundamental reaction paths remains limited due to the analytical complexities posed by high-temperature reactions in glassware. Therefore, to expand this knowledgebase, this study investigates the reaction mechanisms behind the reductive fluorination of  $WO_3$  using polytetrafluoroethylene (PTFE) in a Ni reactor. Here, we explore varied reaction conditions (temperature, duration, and F/W ratio) to suppress the formation of carbon byproducts, minimize the dissipation of fluorine-containing tungsten (VI) compounds, and achieve a high fluorine content. The gas-solid reaction paths are analyzed using infrared spectroscopy which revealed tetrafluoroethylene ( $C_2F_4$ ), hexafluoropropene ( $C_3F_6$ ), and iso-octafluoroisobutene ( $i-C_4F_8$ ) to be the reactive components in the PTFE-decomposition gas during the reactions with  $WO_3$  at 500 °C.  $CO_2$  and  $CO$  are further identified as gaseous byproducts of the reaction evincing that the reaction is prompted by difluorocarbene ( $:CF_2$ ) formed after the cleavage of  $C=C$  bonds in  $i-C_4F_8$ ,  $C_3F_6$ , and  $C_2F_4$  upon contact with the  $WO_3$  surface. The solid-solid reaction path is established through a reaction between  $WO_3$  and  $WO_{3-x}F_x$  where solid-state diffusion of  $O^{2-}$  and  $F^-$  is discerned at 500 °C.

## 1. INTRODUCTION

Transition-metal oxides have gained widespread utility as multi-functional materials on the grounds of their ubiquity and chemical, structural and electronic functionalities that facilitate the uses as catalysis, energy materials, and even next-generation electronics.<sup>1-3</sup> Moreover, their oxide frameworks allow the introduction of different anionic species to form mixed anion compounds, some of which have been noted to have vastly disparate properties.<sup>4-12</sup> For transition-metal compounds, the valence states and electronic properties of the central atoms are influenced by the surrounding anions, and thus modifying the anions even in non-stoichiometric amounts, can generate new properties. Prime examples of such compounds are oxyfluorides which are formed through the substitution of  $O^{2-}$  anions with  $F^{-}$  anions. Since fluorine has one proton and electron more than oxygen, the anion substitution introduces an additional electron into the conduction bands of the parent compound – a mechanism that has been linked to increased conductivities in some of the transition-metal oxyfluorides reported. This advantage of  $F^{-}$  anion modification is further complemented by the similarity in the ionic sizes of the  $F^{-}$  and  $O^{2-}$  anions which minimize the structural deformation during the process.<sup>4,6,13-22</sup>

Fluorination techniques have become subjects of interest in recent studies focusing

on oxyfluorides. Although various fluorinating reagents such as elemental  $F_2$ , HF, and  $SF_4$  have been reported, most of them tend to be oxidative or substitutive. In addition, these reagents are highly toxic and corrosive, which makes them difficult to handle. As a solution, fluoropolymers such as polytetrafluoroethylene (PTFE)<sup>13-15,23-30</sup> or polyvinylidene difluoride<sup>15,31-36</sup> has drawn interest. This fluorination technique facilitates the simultaneous exchange of the  $O^{2-}$  from the transition-metal oxides and the  $F^-$  from the fluorocarbon gases while reducing the oxidation state of the central atom. In particular, PTFE has been adapted for the reductive fluorination of transition-metal oxides such as  $Nb_2O_5$ ,  $Ta_2O_5$ , and  $MoO_3$ , whereby the degree of fluorination and the resultant oxyfluoride structures can be controlled by adjusting the quantity of PTFE employed or varying the reaction temperatures.<sup>13</sup>

Among the transition-metal oxides investigated so far, reductive fluorination is particularly suited for  $WO_3$  – a material of focus in both experimental and theoretical researches.<sup>14,23,37-42</sup>  $WO_3$  has been found to undergo complicated structural evolutions when subjected to certain conditions. For instance, when heat is applied, the low-temperature monoclinic (II) phase transforms into a monoclinic (I) phase via a triclinic structure. The monoclinic (I) phase, which is stable at room temperature, transforms into an orthorhombic phase when heated to around 330 °C, after which it changes to a

tetragonal phase.<sup>43,44</sup> Likewise, substituting the  $O^{2-}$  in  $WO_3$  with  $F^-$  has been found to form  $WO_{3-x}F_x$ -type fluorinated  $WO_3$  with different structural configurations depending on the amount of F. Previous studies have shown that fluorinating  $WO_3$  using PTFE produces a  $WO_{3-x}F_x$  whose structure changes from a monoclinic lattice to orthorhombic, tetragonal, and cubic ones with increasing degrees of fluorination.<sup>14,23,37-39</sup> The benefits of high degrees of fluorination are represented by the cubic phase of  $WO_{3-x}F_x$  ( $x = 0.45$ ), which manifests superconducting properties and significantly smaller resistivity compared with  $WO_3$  and the tetragonal phase of  $WO_{3-x}F_x$  ( $x = 0.11$ ).<sup>14</sup> The efficacy of reductive fluorination was confirmed by a study that achieved a highly fluorinated  $WO_{3-x}F_x$  ( $x = 0.60$ ) through spark plasma sintering a mixture of  $WO_3$  and PTFE powder, which provided photocatalytic properties.<sup>23</sup> It is worth noting that the resulting  $WO_{3-x}F_x$  ( $x = 0.60$ ) was comparable to a previous  $x = 0.66$  product achieved through fluorination using HF under extreme conditions: 3000 atm and 700 °C in a sealed gold ampoule.<sup>37</sup> XPS spectroscopy also revealed the  $WO_{3-x}F_x$  synthesized using PTFE comprised W atoms with the mixed oxidation states of +6 and +5.<sup>23</sup>

The pivotal role of PTFE in reductive fluorination has led to numerous investigations into its decomposition behavior under different conditions.<sup>45-65</sup> According to previous work, the decomposition of PTFE generally starts around 260 °C and significantly

escalates over 400 °C.<sup>48</sup> From a chemical perspective, the decomposition process begins after the C–C bonds in PTFE are broken, leading to the formation of difluorocarbene (:CF<sub>2</sub>). Subsequently, various fluorocarbon gases such as tetrafluoroethylene (C<sub>2</sub>F<sub>4</sub>), perfluorocyclobutane (*c*-C<sub>4</sub>F<sub>8</sub>), hexafluoropropene (C<sub>3</sub>F<sub>6</sub>), octafluoroisobutene (*i*-C<sub>4</sub>F<sub>8</sub>), and hexafluoroethane (C<sub>2</sub>F<sub>6</sub>) are formed depending on atmospheres and temperatures.<sup>45,46</sup> Regarding the roles of these decomposition products in reductive fluorination, some reports have suggested that the process is mainly induced by C<sub>2</sub>F<sub>4</sub> to form CO<sub>2</sub>.<sup>15,23</sup> Despite the knowledge amassed on the subject, the correlation between the decomposition products from PTFE and the reductive fluorination of transition-metal oxides such as WO<sub>3</sub> has not been verified to the best of our knowledge. As such, this study attempts to elucidate the underlying reaction mechanism between WO<sub>3</sub> and PTFE-decomposition gas. Besides the gas-solid reactions, a solid-solid reaction between WO<sub>3</sub> and WO<sub>3-x</sub>F<sub>x</sub> was also conducted to affirm the mutual diffusion of F<sup>-</sup> and O<sup>2-</sup> in the solid state.

## 2. EXPERIMENTAL SECTION

**Apparatus and materials.** All volatile materials were handled in a reaction manifold made of SUS-316 stainless steel and tetrafluoroethylene-perfluoroalkylvinylether copolymer.<sup>66</sup> The gaseous materials in the reaction manifold were evacuated through a

soda lime chemical trap, and thereafter, a cold trap cooled to  $-196\text{ }^{\circ}\text{C}$  by liquid nitrogen. The non-volatile materials were, when necessary, handled under a dry Ar atmosphere in a glove box. A Ni reactor ( $100\text{ cm}^3$ ) equipped with a corrosion-resistant valve was used for reactions. The reactor lid was air-cooled during reactions. A tube made of  $\text{CaF}_2$  crystal was used for infrared (IR) measurements of PTFE-decomposition gas at  $500\text{ }^{\circ}\text{C}$ . Additional details of the measurement methods will be provided in the next subsection. Approximately 100 mg of  $\text{WO}_3$  powder (FUJIFILM Wako Chemicals, mean diameter  $\sim 100\text{ nm}$ ) was pressed into a pellet with a diameter of 6 mm and a thickness of 0.7 mm ( $\sim 30\%$  porosity). PTFE powder (FUJIFILM Wako Chemicals, MW 5,000 $\sim$ 20,000) was used as purchased.

**Analyses.** Infrared spectra of gaseous samples were recorded using an ALPHA II spectrometer (Bruker Optics Laboratories, Inc.) in the transmission mode with the aid of a gas cell with AgCl windows. Powder X-ray diffraction (XRD) patterns were obtained in a Bragg-Brentano geometry using a Smartlab diffractometer (Rigaku Corp., D-tex Ultra 250) equipped with a Si-strip high-speed detector ( $\text{Cu } K\alpha$  radiation, 40 kV-30 mA). X-ray photoelectron spectroscopy (XPS) data were obtained with a JPS-9030 MC spectrometer (JEOL, Ltd.,  $\text{Mg}K\alpha$  radiation). The samples for XPS analysis were fixed on

indium metal foil. The baseline was corrected by the Shirley method and fitting of the resulting spectrum was performed using pseudo Voigt function. Thermal stability of the sample was investigated by a thermogravimetric differential thermal analyzer (ThermoPlusEvo2 TG8120) at a scan rate of 5 °C min<sup>-1</sup> under an Ar flow atmosphere. The sample for thermogravimetry was loaded on an Al pan. Elemental analysis was performed Organic Microanalysis Laboratory, Kyoto University. Carbon content was analyzed by combustion analysis (J-Science Lab JM11). Fluorine content was analyzed by ion chromatography (ThermoFisher ICS-1600) after complete combustion. The F content by this method was validated by comparing several obtained values with the ones by the fluoride ion selective electrode method.<sup>67</sup>

**Decomposition behavior of PTFE under vacuum at different decomposition and measurement temperatures.** The decomposition gases of PTFE were analyzed by IR measurements at 25 °C and 500 °C. For the IR measurements at 500 °C, a stainless-steel block with a horizontal through-hole and a vertical blind hole was placed in the IR spectrometer within the confines of the glove box. The outside of the heating block was cooled with blown Ar to avoid damaging the spectrometer. The tube made of CaF<sub>2</sub> was placed in the vertical blind hole and gradually heated up to 500 °C using a heater built



into the stainless-steel block. The PTFE powder, dried overnight under vacuum at 150 °C, was placed in the CaF<sub>2</sub> tube, whereafter, the IR spectrum of the decomposition gas was immediately obtained. For the IR measurements at 25 °C, PTFE powder on a Ni boat was loaded into the Ni reactor. The Ni reactor was heated to 150 °C under vacuum overnight to dry the PTFE powder. The reactor was then isolated from the reaction manifold and heated to the respective target temperatures (450, 500, or 550 °C) for more than 12 h. After the PTFE decomposition, the Ni reactor was cooled to 25 °C overnight. The volatile gas was subsequently transferred to a gas cell for IR measurements.

**One-step reactions of WO<sub>3</sub> with PTFE-decomposition gases.** A WO<sub>3</sub> pellet (approximately 100 mg) and PTFE powder (F/W = 3.56, 6.00, 8.00, 10.00, or 12.00; where F/W is the molar ratio of the F in PTFE ((C<sub>2</sub>F<sub>4</sub>)<sub>n</sub>) to the W in WO<sub>3</sub>) were loaded onto separate Ni boats and placed inside the Ni reactor. The reactor was heated to 150 °C under vacuum overnight to dry the starting materials. The reactor was subsequently isolated from the reaction manifold and heated to the respective target temperatures (450, 500, or 550 °C) for designated durations (4, 12, or 36 h). After each reaction, the reactor was cooled to 25 °C overnight, whereafter the volatile gas was transferred to the gas cell for IR measurements and the solid product was isolated for further analyses. Caution:

Fluorocarbon gases formed by PTFE decomposition are hazardous and must be carefully handled to prevent any exposures.<sup>45</sup>

**Two-step reaction of WO<sub>3</sub> with PTFE-decomposition gases.** For this investigation, two Ni reactors were employed to separate the reaction between WO<sub>3</sub> and PTFE decomposition gases. One reactor loaded with PTFE powder on a Ni boat was used for PTFE decomposition (F/W = 8.00). The other reactor enclosed the WO<sub>3</sub> pellet on a Ni boat and was used the reaction between the PTFE decomposition gas and WO<sub>3</sub>. Both the reactors were heated to 150 °C under vacuum overnight to dry the starting materials and isolated from the reaction manifold. The PTFE-decomposition process was conducted at 500 °C for 12 h, after which the Ni reactor was cooled to 25 °C overnight. A portion of the volatile gas was transferred to the gas cell for IR measurements to ascertain the gas components. Subsequently, the volatile gases both in the first Ni reactor and in the gas cell were entirely transferred to the second Ni reactor containing WO<sub>3</sub> by cooling it to -196 °C using liquid nitrogen. After the gas transfer, the second reactor was isolated from the reaction manifold and heated to 500 °C for 36 h. Once the reaction was complete, the reactor was cooled down to 25 °C overnight. The volatile gas was transferred to the gas cell for IR measurements, and the solid product was isolated for further analyses.

**Solid-solid reaction of WO<sub>3</sub> and WO<sub>3-x</sub>F<sub>x</sub>.** WO<sub>3</sub> and WO<sub>3-x</sub>F<sub>x</sub> ( $x = 0.36$ , synthesized by the reaction of WO<sub>3</sub> and PTFE at 500 °C for 36 h with the F/W ratio of 6.00 as described in Table 1) powders were mixed using a pestle and a mortar and pressed into a pellet. The obtained pellet was placed on a Ni boat and loaded into the Ni reactor for drying under vacuum at 150 °C overnight. The reactor was isolated from the reaction manifold and heated to 500 °C for 24 h. After the reaction, the Ni reactor was cooled down to 25 °C, and the solid product was isolated for further analyses.

**Quantum chemical calculations.** Theoretical calculations were performed based on density functional theory (DFT) and Møller-Plesset perturbation theory (MP). The exchange-correlation functional for DFT was B3LYP<sup>68</sup>, and the 2<sup>nd</sup> order perturbation was adopted for MP (MP2 method).<sup>69</sup> The calculations were performed using the Gaussian16 program.<sup>70</sup> The total energies, electronic structures, and bond frequencies of C<sub>2</sub>H<sub>4</sub>, :CF<sub>2</sub>, C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, *i*-C<sub>4</sub>F<sub>8</sub>, 1-octafluorobutene (1-C<sub>4</sub>F<sub>8</sub>), 2-*cis*-octafluorobutene (2-*cis*-C<sub>4</sub>F<sub>8</sub>), and 2-*trans*-octafluorobutene (2-*trans*-C<sub>4</sub>F<sub>8</sub>) were estimated using the 6-311+G(d,p) basis set. The optimized geometries had no imaginary frequency. Wiberg bond indices were calculated as the bond orders via the natural atomic orbital analysis

### 3. RESULTS AND DISCUSSION

**Reactions of WO<sub>3</sub> with PTFE-decomposition gases at different conditions.** In the present study, Ni reactors were selected for reductive fluorination reactions due to their low reactivity with PTFE-decomposition gases, which provides higher durability than quartz reactors (see Supporting Information for the IR spectra (Figure S1) and additional discussion on the PTFE-deposition gas reactivity with quartz at 550 °C). As shown by the schematic in Figure 1(a), the WO<sub>3</sub> and PTFE samples were loaded onto separate boats in the same Ni reactor heated in a one-step reaction process. Table 1 highlights the reaction parameters *i.e.*, temperature, duration, and F/W ratio (elemental ratio of F in PTFE ((C<sub>2</sub>F<sub>4</sub>)<sub>n</sub>) to W in WO<sub>3</sub>) alongside data derived from the reaction products ( $R_w$  (wt%): weight of WO<sub>3-x</sub>F<sub>x</sub> after reaction/weight of WO<sub>3</sub> before reaction;  $C_F$  (wt%): fluorine content in products;  $x$ : composition of WO<sub>3-x</sub>F<sub>x</sub> calculated from  $C_F$ ; and  $C_c$  (wt%): carbon content in products).

The effect of temperature on the reactivity was investigated by comparing the reaction products obtained under different reaction temperatures with the reaction duration and F/W ratio parameters (36 h and F/W = 3.56) kept constant. A reaction temperature of

550 °C was selected to compare the reaction products to those obtained by a previous study that utilized a quartz reactor.<sup>14</sup> At that temperature, the reaction product have a  $C_F$  of 3.41 % ( $WO_{2.58}F_{0.42}$ ), which is comparable to the  $WO_{2.55}F_{0.45}$  previously reported. The product had a low  $R_w$  of 25.3% despite the high fluorine content. The large decrease in weight indicates that the fluorinated product partially sublimed as was confirmed by the presence of fluorine-containing tungsten species around the lid (cooler parts) of the Ni reactor. On the other hand, the reaction product in the Ni boat was found to have a high  $C_c$  of 3.87 %, revealing that the reaction between the PTFE-decomposition gas and  $WO_3$  at 550 °C forms carbon byproduct – consistent with the black color of the product. The structure of this residual carbon was analyzed by XRD in the case of the reaction with excess PTFE ( $F/W = 54.2$ ) at 550 °C which gave a high carbon content of 73.7 %. The resulting XRD pattern indicates its amorphous structure characterized by broad peaks around 23° and 42° corresponding to 002 and 100 diffraction lines, respectively (see Figure S2 for the corresponding XRD pattern). The solubility of carbon into the  $WO_{3-x}F_x$  host is neglected in the following characterization.

Further, the reaction product attained at 450 °C (36 h and  $F/W = 3.56$ ) was investigated to ascertain the effect of lower reaction temperatures. Here, the sublimation of  $WO_{3-x}F_x$  and the carbon formed were significantly suppressed, yielding a high  $R_w$  of 98.3% and a

low  $C_c$  of 0.15%. Even so, the reaction yielded a lower fluorine content ( $C_F$  of 0.22 % corresponding to  $WO_{2.97}F_{0.03}$ ), signifying that the low reaction temperature was not sufficient to complete the PTFE decomposition; as corroborated by the residual PTFE, which had melted and resolidified in the Ni boat in the course of the reaction. This product (450 °C) was light blue and was comparable to a previously reported product with a low fluorination degree (light green for  $x = 0.04$  (orthorhombic structure)).<sup>37</sup> In an attempt to increase the fluorination degree, reaction products from the higher reaction temperature of 500 °C (36 h and  $F/W = 3.56$ ) were examined. The fluorinated product was dark blue with a higher  $R_w$  of 76.8 %, a slightly lower  $C_F$  of 2.94 % ( $WO_{2.64}F_{0.36}$ ), and a significantly lower  $C_c$  of 0.32 wt%, compared to the product obtained at 550 °C. Thermogravimetric analysis revealed that  $WO_{2.64}F_{0.36}$  has a high thermal stability, exhibiting a gentle weight loss up to 500 °C (1% and 5% weight loss at 195 °C and >500 °C, respectively) (see Figure S3 for the thermogravimetric curve of  $WO_{2.64}F_{0.36}$ ). The presence of  $W^{5+}$  in  $WO_{2.64}F_{0.36}$  is confirmed by XPS analysis (see Figures S4 for XPS spectrum of  $WO_{2.64}F_{0.36}$ ). The  $W^{6+}/W^{5+}$  ratio estimated by peak fitting is consistent with the composition of  $WO_{2.64}F_{0.36}$  ( $W^{6+}/W^{5+} = 0.64/0.36$ ), suggesting the negligible contribution of anionic vacancy which can affect the  $W^{6+}/W^{5+}$  ratio.

Table 1. Experimental conditions and results of the reactions between WO<sub>3</sub> and the PTFE-decomposition gas

| Entry | Temperature<br>[°C] | F/W <sup>a</sup> | Time<br>[h] | R <sub>w</sub> <sup>b</sup><br>[wt%] | C <sub>F</sub> <sup>c</sup><br>[wt%] | Composition<br>x in WO <sub>3-x</sub> F <sub>x</sub> <sup>d</sup> | C <sub>C</sub> <sup>e</sup><br>[wt%] | Product<br>color |
|-------|---------------------|------------------|-------------|--------------------------------------|--------------------------------------|---|--------------------------------------|------------------|
| 1     | 450                 | 3.56             | 36          | 98.3                                 | 0.22                                 | 0.03  | 0.15                                 | Light blue       |
| 2     | 500                 | 3.56             | 4           | 92.9                                 | 1.36                                 | 0.17  | < 0.10                               | Blue             |
| 3     | 500                 | 3.56             | 12          | 92.6                                 | 1.46                                 | 0.18  | 0.16                                 | Blue             |
| 4     | 500                 | 3.56             | 36          | 76.8                                 | 2.94                                 | 0.36  | 0.32                                 | Dark blue        |
| 5     | 500                 | 6.00             | 36          | 76.9                                 | 2.91                                 | 0.36  | 0.16                                 | Dark blue        |
| 6     | 500                 | 8.00             | 36          | 76.7                                 | 2.58                                 | 0.32  | 0.13                                 | Dark blue        |
| 7     | 500                 | 10.00            | 36          | 63.4                                 | 3.55                                 | 0.44  | 0.52                                 | Light purple     |
| 8     | 500                 | 12.00            | 36          | 41.0                                 | 3.49                                 | 0.43  | 2.20                                 | Bluish black     |
| 9     | 550                 | 3.56             | 36          | 25.3                                 | 3.41                                 | 0.42  | 3.87                                 | Black            |

<sup>a</sup> F/W: elemental ratio of F in PTFE ((C<sub>2</sub>F<sub>4</sub>)<sub>n</sub>) to W in WO<sub>3</sub>, <sup>b</sup> R<sub>w</sub>: weight of WO<sub>3-x</sub>F<sub>x</sub> (after reaction) / weight of WO<sub>3</sub> (before reaction), <sup>c</sup> C<sub>F</sub>: fluorine content in product, <sup>d</sup> the composition x in WO<sub>3-x</sub>F<sub>x</sub> is determined based on C<sub>F</sub>, <sup>e</sup> C<sub>C</sub>: carbon content in product.

For insight into the effect of temperature on the structural configurations of the fluorinated products, the XRD patterns of the reaction products obtained at 450, 500, and 550 °C, for the reaction duration of 36 h and F/W ratio of 3.56, were compared to pristine WO<sub>3</sub>, as shown in Figure 1(b). The pristine WO<sub>3</sub> possesses a monoclinic structure as previously reported according to a Rietveld refinement (see Figure S5 and Table S1 for detailed crystallographic parameters). The reaction product obtained at 500 °C, with a composition of WO<sub>2.64</sub>F<sub>0.36</sub> (Table 1), is assigned to a single cubic phase. Rietveld refinement confirmed that this phase belongs to the ReO<sub>3</sub>-type structure with O/F-disordering in congruence with a previous report (see Figure S6 and Table S2 for detailed crystallographic parameters).<sup>14,23</sup> At the higher temperature of 550 °C, the reaction yielded a product in the cubic phase. Although some other peaks are also observed in the pattern (around  $2\theta = 25.8^\circ$  and  $36\text{--}38^\circ$ ), they were not able to be indexed as a superlattice by considering the tilting of the octahedra. On the other hand, the XRD pattern of the reaction product obtained at 450 °C only exhibited slight differences from the pristine WO<sub>3</sub> pattern, revealing that the structure is not a cubic phase.

The influence of reaction time was investigated by implementing reactions at the temperature of 500 °C and F/W ratio of 3.56 at varied durations of 4, 12, and 36 h (Table 1). The 4 and 12 h reaction durations yielded products with significantly lower  $C_F$  values



(1.36% and 1.46%, respectively) than those obtained in 36 h (2.94%). This XRD analysis confirms that the reaction duration of 36 h is a requisite to achieve a complete transition to the cubic phase (Figure S7 for XRD patterns of the products with different reaction durations). On another note, the PTFE on the Ni boat was observed to completely disappear in all reaction cases. This indicates that even the reaction duration of 4 h was sufficient to facilitate complete PTFE decomposition. These results suggest that the rate-determining step is not the decomposition of PTFE, but rather the reaction between  $\text{WO}_3$  and the PTFE-decomposition gas.

The influence of F/W ratio on the reaction between the PTFE-decomposition gas and  $\text{WO}_3$  was evaluated by varying the F/W ratio while maintaining the reaction temperature at 500 °C and reaction duration at 36 h. XRD analyses of the reaction products confirmed that the cubic phase was maintained in all the cases regardless of the F/W ratio applied (see Supporting Information; Figure S8 for the XRD patterns of the products with different F/W (3.56, 6.00, 8.00, 10.00, and 12.00) and Figures S9–S12 and Tables S3–S6 for detailed crystallographic parameters by Rietveld refinement). Although only a limited  $x$  range is available in the present case, there is a trend that the  $a$  lattice parameter increasing with increasing  $x$  in  $\text{WO}_{3-x}\text{F}_x$ . The correlation between the F/W ratio and the  $C_F$ ,  $C_C$ , and  $R_w$  values of the reaction products is summarized in Figure 1(c) (see Table 1

for detailed data). The trend suggests that an increase in the F/W ratio increases the  $C_F$  and  $C_C$  values and decreases  $R_w$  values of the reaction products. Even so, the F/W ratios of over 3.56 did not significantly enhance the F content, but merely accelerated the formation of carbon byproduct and the partial sublimation of the product. The carbon byproduct was visually confirmed by the bluish-black color of the product obtained by F/W = 12.00 ( $C_C = 2.20\%$ ). The sublimation was corroborated by the color change of the deposits to yellow upon exposure to air around the lid of the Ni reactor, although the large amounts of white repolymerized PTFE from  $C_2F_4$  in the PTFE decomposition gas were also seen when the F/W ratio was increased (see Figure S13 for photos of the lid after the reaction at 500 °C for 36 h with F/W = 12.00).

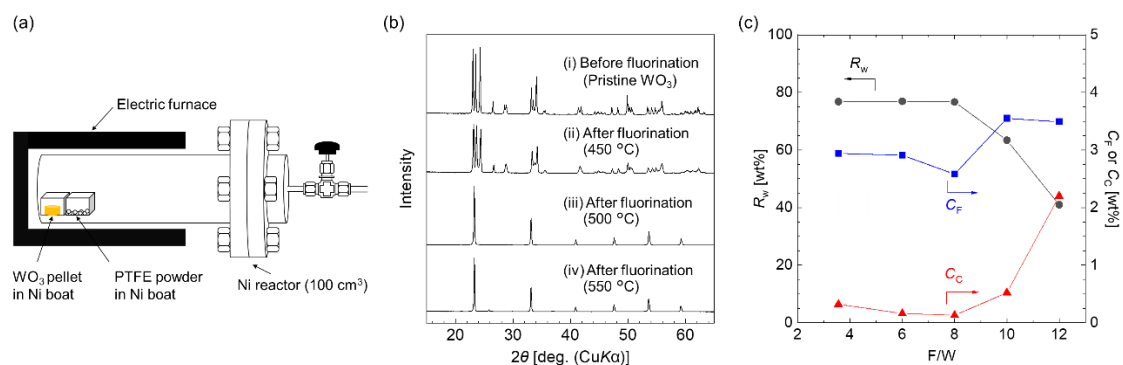


Figure 1. (a) Schematic of the one-step reaction between  $\text{WO}_3$  and PTFE. (b) XRD patterns of (i) pristine  $\text{WO}_3$  and  $\text{WO}_{3-x}\text{F}_x$  obtained at (ii) 450 °C, (iii) 500 °C, and (iv) 550 °C. The reaction time and F/W ratio were 36 h and 3.56, respectively, in all the cases. See Supporting Information for detailed crystallographic data obtained by Rietveld refinement (Figure S5 and Table S1 for (i) pristine  $\text{WO}_3$  and Figure S6 and Table S2 for (iii)  $\text{WO}_{2.64}\text{F}_{0.36}$ ). (c) The F/W ratio dependency of  $R_w$ ,  $C_F$ , and  $C_C$  in reactions of  $\text{WO}_3$  and the PTFE-decomposition gas. Reaction temperature: 500 °C and reaction time: 36 h.

**PTFE-decomposition gases at different decomposition and measurement temperatures.** Identifying the components of the PTFE-decomposition gas is crucial to elucidating the reductive fluorination mechanisms. Indeed, PTFE decomposition has been the subject of many researches,<sup>45-65</sup> which have prescribed the following decomposition mechanism:(i) chain cleavage of PTFE, (ii)  $\text{:CF}_2$  formation, (iii)  $\text{C}_2\text{F}_4$  formation from two  $\text{:CF}_2$ , (iv-A) *c*- $\text{C}_4\text{F}_8$  formation from two  $\text{C}_2\text{F}_4$  molecules, (iv-B)  $\text{C}_3\text{F}_6$  formation from  $\text{C}_2\text{F}_4$  and  $\text{:CF}_2$ , and (v) *i*- $\text{C}_4\text{F}_8$  formation from  $\text{C}_3\text{F}_6$  and  $\text{:CF}_2$ .<sup>45,46</sup> It is worth noting that the decomposition mechanism of PTFE is complicated and generally depends on the reaction conditions. In fact, certain conditions have also been reported to generate other chemical species, such as  $\text{C}_2\text{F}_6$ . As such, it is important to investigate mechanism using the present experimental system for clarity.

To determine the components of the PTFE-decomposition gas, IR spectroscopy was performed on the PTFE-decomposition gas produced at 450, 500, and 550 °C; measured after the gas was cooled to 25 °C. A summary of the PTFE-decomposition gas is provided in Table 2. As shown in Figure 2(a-i, ii, and iii), all cases exhibited complicated bands between 1000–1400  $\text{cm}^{-1}$ . However, the bands above 1700  $\text{cm}^{-1}$  and below 1000  $\text{cm}^{-1}$  were clear enough to identify the chemical components of the gas. Each IR spectrum is marked by a band representing the  $\text{CF}_2$  symmetric stretching mode of *c*- $\text{C}_4\text{F}_8$  at 962

$\text{cm}^{-1}$ .<sup>51,73-76</sup> Additionally, the bands appearing at 1751 and 1795  $\text{cm}^{-1}$  are ascribed to the C=C stretching mode of *i*-C<sub>4</sub>F<sub>8</sub><sup>51,77</sup> and C<sub>3</sub>F<sub>6</sub>,<sup>51,73,78,79</sup> respectively.

PTFE decomposition at 450 °C gives a gas that mainly consists of *c*-C<sub>4</sub>F<sub>8</sub> while C<sub>3</sub>F<sub>6</sub> and *i*-C<sub>4</sub>F<sub>8</sub> participate as a minor species. When the decomposition temperature was raised to 500 °C, the *i*-C<sub>4</sub>F<sub>8</sub> species was seen to increase while the *c*-C<sub>4</sub>F<sub>8</sub> and C<sub>3</sub>F<sub>6</sub> components decreased. Comparing the intensities of the bands appearing at 962 and 1795  $\text{cm}^{-1}$  reveals that decomposition at 500 °C leads to a more significant decrease in the *c*-C<sub>4</sub>F<sub>8</sub> than C<sub>3</sub>F<sub>6</sub>. This trend becomes more prevalent when the temperature is increased to 550 °C, resulting in *i*-C<sub>4</sub>F<sub>8</sub> as the main component and *c*-C<sub>4</sub>F<sub>8</sub> and C<sub>3</sub>F<sub>6</sub> as minor species. It is worth noting that C<sub>2</sub>F<sub>4</sub>, which is typically represented by a band at 1325  $\text{cm}^{-1}$ ,<sup>51,52</sup> was not identified in the spectra, presumably because it repolymerized back to PTFE when the gas was cooled to 25 °C before the IR spectroscopy. The influence of temperature on the components of the PTFE-decomposition gas aligns with a previous report, although the formation temperatures of the gaseous species were different, probably due to variations in the experimental conditions such as the N<sub>2</sub> flow atmosphere and the sealed quartz reactor used in the previous work.<sup>50</sup>

As mentioned above, the IR spectra obtained after cooling the PTFE-decomposition gases to 25 °C do not entirely reflect the gaseous components involved in the reactions

with  $\text{WO}_3$ . This deduction is also confirmed by the formation of repolymerized PTFE around the lid of the Ni reactor after the reactions despite the absence of  $\text{C}_2\text{F}_4$  bands in the IR spectra measured at  $25^\circ\text{C}$ . The  $\text{C}_2\text{F}_4$  gaseous component is particularly difficult to identify at  $25^\circ\text{C}$  due to the formation of overlapping absorption bands and repolymerization. It is interesting to note that although numerous studies have proposed that  $\text{C}_2\text{F}_4$  is formed immediately after the decomposition of PTFE,<sup>45,46,48,51,52,56</sup> none of them have provided spectroscopic evidence of  $\text{C}_2\text{F}_4$  at high temperatures of around  $500^\circ\text{C}$ : to the best of our knowledge.

Therefore, for further characterization of the PTFE decomposition components, the gas was subjected to *in situ* IR spectroscopy at  $500^\circ\text{C}$  using a  $\text{CaF}_2$  container under a dry Ar atmosphere. The corresponding spectra are illustrated in Figure 2(b). Apart from the broad absorption bands propagated by the high measurement temperature,<sup>73</sup> the *in situ* spectrum displayed other differences from the one measured at  $25^\circ\text{C}$  (Figure 2(a-ii)), even though the decomposition temperature was the same ( $500^\circ\text{C}$ ). For instance, the characteristic bands corresponding to the C=C stretching modes of  $\text{C}_3\text{F}_6$  and *i*- $\text{C}_4\text{F}_8$ , which are expected to appear at  $1795$  and  $1751\text{ cm}^{-1}$ , respectively, are absent in the *in situ* spectrum. Additionally, two broad bands that can be assigned to the  $\text{CF}_2$  symmetric and asymmetric stretching modes of  $\text{C}_2\text{F}_4$  appear at  $1000$  and  $1400\text{ cm}^{-1}$  (reported at  $1183$  and

1325  $\text{cm}^{-1}$  in previous literature).<sup>51,52,73</sup> It is also worth mentioning that the absorption bands for the singlet ground state of  $:\text{CF}_2$ , which typically appear at 1225 and 1114  $\text{cm}^{-1}$  in congruity with the symmetric and asymmetric stretching modes<sup>80,81</sup> are not clearly visible in the *in situ* spectrum, although they might still exist as minor components. Nonetheless, the results from this investigation demonstrate that  $\text{C}_2\text{F}_4$  is formed in the initial step of PTFE decomposition at the high temperature of around 500 °C and further confirm the mechanisms proposed in previous works.<sup>45,46</sup>

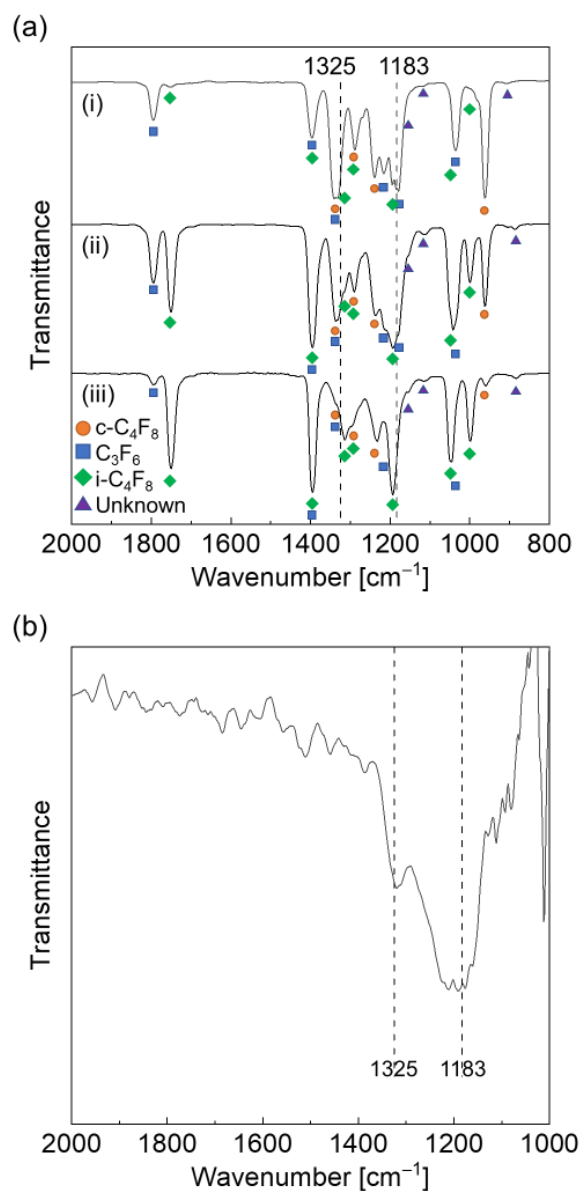
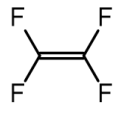
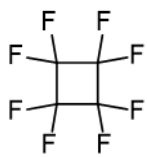
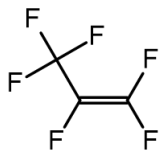



Figure 2. Infrared spectra of PTFE-decomposition gases. (a) Spectra of the gases decomposed at (i) 450 °C, (ii) 500 °C, and (iii) 550 °C (measured after cooling down to 25 °C). (b) *In situ* IR spectrum of the gas decomposed and measured at 500 °C. The wavenumbers of 1183 and 1325  $\text{cm}^{-1}$  correspond to the  $\text{CF}_2$  symmetric and asymmetric stretching modes of  $\text{C}_2\text{F}_4$ , respectively.<sup>51,52,73</sup> Although the stretching modes of  $:\text{CF}_2$  may appear in this region (1225 and 1114  $\text{cm}^{-1}$  for the singlet ground state of  $:\text{CF}_2$  according to the literature),<sup>80,81</sup> they are not clearly observed in this spectrum.



Table 2. Components of PTFE-decomposition gases at different decomposition and measurement temperatures

| Gaseous species  | Decomposition / measurement temperatures [°C] |           |          |                                 |
|--|---|-----------|----------|---------------------------------|
|  | 450 / 25                                      | 500 / 25  | 550 / 25 | 500 / 500<br>( <i>in situ</i> ) |
| <br><b>C<sub>2</sub>F<sub>4</sub></b>     | N   | N         | N        | Major                           |
| <br><b>c-C<sub>4</sub>F<sub>8</sub></b>   | Major   | Secondary | Minor    | N                               |
| <br><b>C<sub>3</sub>F<sub>6</sub></b>   | Intermediate                                  | Secondary | Minor    | N                               |
| <br><b>i-C<sub>4</sub>F<sub>8</sub></b> | Minor   | Major     | Major    | N                               |

Major: primary major species, Secondary: secondary major species, Minor: minor species, N: not observed.

**Reactivities of fluorocarbon gases from decomposed PTFE with WO<sub>3</sub>.** Based on the IR spectroscopy results in the section above, PTFE decomposition at 500 °C initially produces C<sub>2</sub>F<sub>4</sub> and possibly :CF<sub>2</sub>, which are then converted to *c*-C<sub>4</sub>F<sub>8</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub>. Although the direct reaction between the :CF<sub>2</sub> and WO<sub>3</sub> is plausible, it has been previously reported to have a short lifetime (0.1 sec at most).<sup>80</sup> However, the present configuration of WO<sub>3</sub> and PTFE in individual Ni boats makes the occurrence of this reaction highly unlikely. This is further substantiated by the *in situ* IR spectrum (Figure 2(b)), which clearly demonstrates that the reductive fluorination of WO<sub>3</sub> involves C<sub>2</sub>F<sub>4</sub> as a main reaction species and not :CF<sub>2</sub>. Even so, the decomposition of PTFE at 650, 700, and 750 °C has been previously reported to form *c*-C<sub>4</sub>F<sub>8</sub> and C<sub>3</sub>F<sub>6</sub> in the span of several seconds.<sup>46</sup> Thus, investigating the contribution of *c*-C<sub>4</sub>F<sub>8</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> components in the present reductive fluorination would be pivotal to clearly depict the underlying reaction mechanisms.

In order to determine the reactions between the *c*-C<sub>4</sub>F<sub>8</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> species and WO<sub>3</sub>, the one-step reaction was divided into two steps wherein the PTFE-decomposition gas was cooled to 25 °C in a first step and reacted with WO<sub>3</sub> in a second step (F/W = 8.00), as illustrated by the schematic in Figure 3(a). In this configuration, the PTFE-decomposition gas was prepared by decomposing PTFE at 500 °C for 12 h and cooling it

to 25 °C overnight. In the second reaction step, WO<sub>3</sub> reacted with the PTFE-decomposition gas at 500 °C for 36 h.

As shown in Figure 3(b), the XRD pattern of the two-step reaction product was compared to the patterns obtained from pristine WO<sub>3</sub> and the one-step reaction product (same as Figure S8(d) and S10) to ascertain the resemblances between their structures. The two-step reaction product manifests a pattern identical to the one-step reaction product – an indication that it also embodies the ReO<sub>3</sub>-type cubic phase. This observation further evinces that some, if not all, of the *c*-C<sub>4</sub>F<sub>8</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> also contribute to the reductive fluorination of WO<sub>3</sub> in a similar manner as C<sub>2</sub>F<sub>4</sub>.

The involvement of the PTFE-decomposition species in the two-step reaction was investigated through the IR spectroscopy of the residual gas obtained after the reaction. Figure 3(c) shows the IR spectrum of the two-step reaction residual gas alongside spectra of the PTFE-decomposition gas obtained and cooled in the absence of WO<sub>3</sub> (same as Figure 2(a-ii)) and the one-step reaction residual gas (500 °C, 36 h, and F/W = 8.00 in Table 1). A comparison between the IR spectra from the two-step reaction residual gas (Figure 3(c-ii)) and the one-step reaction residual gas (Figure 3(c-iii)) indicates that the reactions are nearly identical. Further investigation of the spectra in Figures 3(c-i) and 3(c-ii) reveals that the absorption bands assigned to *i*-C<sub>4</sub>F<sub>8</sub> (1751 cm<sup>-1</sup>) disappeared while

those ascribed to  $C_3F_6$  ( $1795\text{ cm}^{-1}$ ) became significantly weaker after the two-step reaction. On the other hand, the intensity of the bands assigned to *c*- $C_4F_8$  (*i.e.*,  $962$  and  $569\text{ cm}^{-1}$ ) notably remained stable before and after the two-step reaction. These results indicate that the  $C_3F_6$  and *i*- $C_4F_8$  contributed to the reductive fluorination reaction at  $500\text{ }^\circ\text{C}$ , whereas the *c*- $C_4F_8$  remained inactive throughout the process. Additionally, the disappearance of the *i*- $C_4F_8$  bands after the two-step reaction affirms that *i*- $C_4F_8$  has a higher reactivity to  $WO_3$  than  $C_3F_6$ .

Besides the fluorocarbon components of the two-step reaction residual gas, the IR spectrum also manifested bands assigned to the bending, and asymmetric stretching modes of  $CO_2$  are found at  $669$  and  $2350\text{ cm}^{-1}$ ,<sup>82,83</sup> alongside bands assigned to the stretching mode of  $CO$  around  $2120$  and  $2170\text{ cm}^{-1}$ .<sup>83,84</sup> It should be pointed out that although previous works have reported the formation of  $CO_2$  during reactions involving PTFE and metal oxides such as  $CuO$ ,  $V_2O_5$ , and  $Fe_2O_3$ , the occurrence of  $CO$  has only been reported in a few cases, and has not even been proposed for the case of  $WO_3$  to date.<sup>52</sup> However, the present study reveals that  $CO$  is formed even during the one-step reaction at  $450\text{ }^\circ\text{C}$ , which results in a product with a high oxygen content and low fluorination degree (see Figure S14 for IR spectrum of the residual gas after the one-step reaction at  $450\text{ }^\circ\text{C}$  for 36 h with  $F/W = 3.56$ ). This indicates that  $CO$  does not generate as

a result of the decreasing O species in  $WO_{3-x}F_x$  during the fluorination reactions but is instead generated concurrently with  $CO_2$  throughout the reaction process

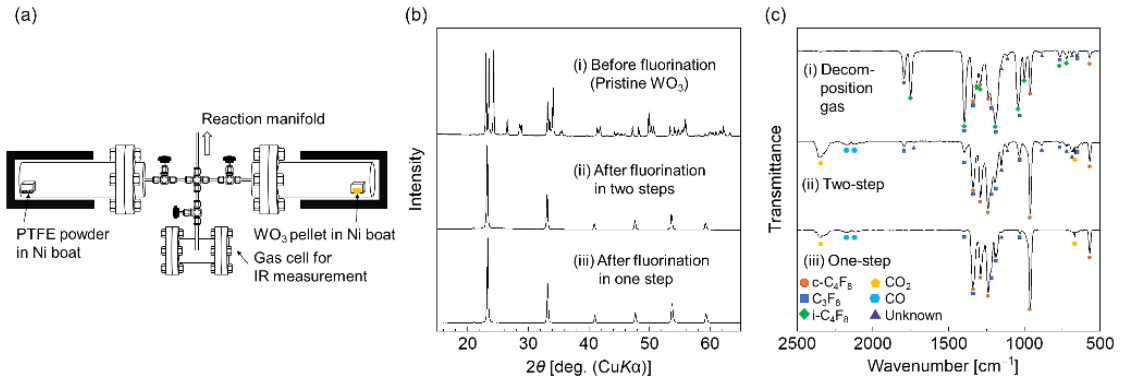


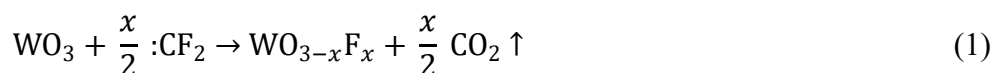
Figure 3. (a) Schematic of the two-step reaction between  $WO_3$  and PTFE. The PTFE-decomposition gas obtained at  $500\text{ }^\circ\text{C}$  was cooled to  $25\text{ }^\circ\text{C}$  in the left reactor and transferred to the right reactor containing  $WO_3$  for reaction at  $500\text{ }^\circ\text{C}$ . (b) XRD patterns of (i) pristine  $WO_3$  and those after fluorination in (ii) the two steps, and (iii) the one step under the reaction temperature of  $500\text{ }^\circ\text{C}$ , and reaction hours of 36 h, and  $F/W = 8.00$ . (c) Infrared spectra of PTFE-decomposition gases before and after reaction. (i) PTFE-decomposition gas at  $500\text{ }^\circ\text{C}$  measured after cooling to  $25\text{ }^\circ\text{C}$  in the absence of  $WO_3$ , (ii) the residual gas after the two-step reaction of (i) with  $WO_3$  (Figure 3(a)), and (iii) the residual gas after the one-step reaction with  $WO_3$  (Figure 1(a)).

### **Reaction mechanisms of fluorocarbon gases from decomposed PTFE with WO<sub>3</sub>.** The

formation of CO<sub>2</sub> and CO from C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> evinces that the reaction induced the cleavage of carbon-carbon bonds in the C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> species. Comparing the molecular structure of *c*-C<sub>4</sub>F<sub>8</sub> with those of C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub>, it is also justifiable to consider that the C=C bonds were not only cleaved but also involved with their reactivity. The cleavage of the C=C bond in C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> ( $R_1R_2C=CF_2$  ( $R_1, R_2 = F$  or  $CF_3$ )) forms :CF<sub>2</sub>, C<sub>2</sub>F<sub>4</sub>, and C<sub>3</sub>F<sub>6</sub>, respectively. Therefore, :CF<sub>2</sub> formation can be considered to occur as a “knock-on effect” from the other components. Considering that the release of :CF<sub>2</sub> from C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> under vacuum is thermodynamically unfavorable (see Figure S15 and Table S7 for the calculated Gibbs energies of the :CF<sub>2</sub> dissociation reactions from C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> by a DFT method (B3LYP)), the interaction between the gas components and the WO<sub>3</sub> surface can be postulated to aid in the C=C bond cleavage and the concomitant :CF<sub>2</sub> formation. Difluorocarbene is notably more stable in the singlet state than in the triplet state, unlike its :CH<sub>2</sub> carbene equivalent, due to the  $\pi$ -donation from the F atoms to the divalent carbon atom. (see Figure S16 for the calculated structures of :CF<sub>2</sub> in the singlet and triplet states and their total energies).<sup>85,86</sup> It should also be pointed out that the formation of :CF<sub>2</sub> carbene is a known

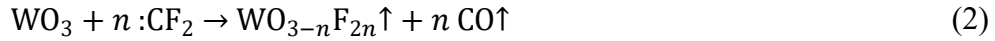
decomposition product of CF<sub>2</sub>-containing species such as HCF<sub>2</sub>Cl and ClCF<sub>2</sub>COONa as well as C<sub>2</sub>F<sub>4</sub>, and can be used to introduce the CF<sub>2</sub> group to organic compounds.<sup>85,87</sup>

In the context of :CF<sub>2</sub> reactivity with WO<sub>3-x</sub>F<sub>x</sub>, the CO<sub>2</sub> and CO confirmed after the reactions are deemed to result from the simultaneous defluorination and oxidation of :CF<sub>2</sub>. As such, the formation of CO<sub>2</sub> and CO is presumed to directly result from the reaction between :CF<sub>2</sub> and WO<sub>3-x</sub>F<sub>x</sub>, and not indirectly from the oxidation of CO and carbon by WO<sub>3-x</sub>F<sub>x</sub>. This is corroborated by previous works, which reported that the oxidation of CO and carbon by WO<sub>3</sub> cannot proceed at low temperatures of 450–550 °C.<sup>88,89</sup> The ratios of CO<sub>2</sub> and CO were estimated using their band intensities in the IR spectra after the reactions at 500 °C. The resulting ratios of CO<sub>2</sub> and CO were found to be closely comparable regardless of the reaction duration or F/W ratios applied (see Figure S17 and Table S8 for partial-pressure ratios of CO<sub>2</sub> and CO after the reactions at 500 °C).<sup>90</sup> Equation (1) below prescribes the CO<sub>2</sub> formation process during the reactions:

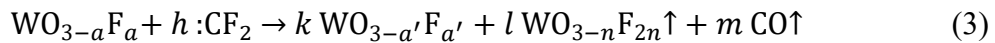


This reaction corresponds to the reductive fluorination (2O–2F exchange), which provides a mixed oxidation state of W(VI) and W(V) in the WO<sub>3-x</sub>F<sub>x</sub> (*x* is the F content

in the final product). On the other hand, the CO formation reaction is exemplified by Equation (2) below:



This reaction is a substitutive fluorination (O–2F exchange) which does not entail a change in the oxidation state of W. Here,  $n$  denotes the number of O atoms substituted. Alternatively, the CO formation can be rationalized by the reaction between  $\text{WO}_{3-a}\text{F}_a$  and  $: \text{CF}_2$ , which also results in a phase separation involving W(VI) species ( $\text{WO}_{3-n}\text{F}_{2n}$ ) and W(VI)-W(V) mixed species ( $\text{WO}_{3-a'}\text{F}_{a'}$ ), as shown in Equation (3) (see Supporting Information for detailed coefficients and compositions of this equation):



Here,  $a$  and  $a'$  are the F contents in the intermediate compounds during the reaction in Equations (1) ( $a < a' < x$ ). Since the resulting W(VI) species,  $\text{WO}_{3-n}\text{F}_{2n}$ , is volatile, it is promptly dissipated from the vicinity of the sample, thus increasing the fluorination

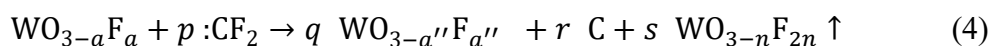


degree of the remaining solid sample and indirectly contributing to the reductive fluorination.

The sublimated W(VI) species are most likely  $\text{WO}_2\text{F}_2$  and  $\text{WOF}_4$ , given their significant vapor pressures at the present reaction temperatures. Accordingly, the color of the sublimed compounds was noted to change from white or blue to yellow upon contact with air indicative of their hydrolysis to  $\text{WO}_3$  (see Figure S13 for photos of the lid of the Ni reactor after the reaction at 500 °C for 36 h with  $F/W = 12.00$ ). Moreover, it is the sublimation of W(VI) species during the CO-formation reaction, and not the sublimation of  $\text{WO}_{3-x}\text{F}_x$ , that provides a more plausible rationale for the decrease in weight observed in the reactions that yielded  $\text{WO}_{3-x}\text{F}_x$  products with low  $x$  values (Table 1 and Figure 1(c)). This is justified by the decreases in weight observed even in the reactions that produced  $\text{WO}_{3-x}\text{F}_x$  with low  $x$  values such as 0.03 and 0.17 (Entries 1 and 2 in Table 1), although  $\text{WO}_{3-x}\text{F}_x$  with  $x$  values below 0.4 are expected to have low vapor pressures given that  $\text{WO}_{3-x}\text{F}_x$  with the  $x$  values at 0.4 is obtainable at 500 °C.

A broader look at the byproducts formed under different reaction conditions reveals that the carbon byproduct is only present in products obtained under severe reaction conditions (see Figure 1(c) and Table 1 for the correlation between reaction conditions and  $C_C$ ). This suggests that carbon is only formed alongside  $\text{CO}_2$  and CO during the

defluorination of :CF<sub>2</sub> in the presence of WO<sub>3-x</sub>F<sub>x</sub> with deficient amounts of oxygen; after fluorination has progressed beyond a certain degree. This reaction process is expressed as Equation (4) below (see Supporting Information for detailed coefficients and compositions of this equation):



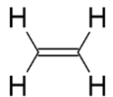
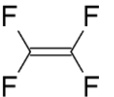
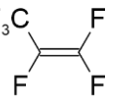
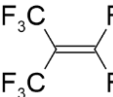
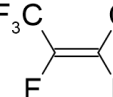
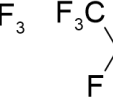
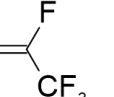
Here,  $a''$  is the F content in the product of the reaction in Equation (4) ( $a'' < a < x$ ). The formation of the sublimable W(VI) species of WO<sub>3-n</sub>F<sub>2n</sub> along with carbon highlighted in this chemical equation is in line with the trends observed in Table 1 and Figure 1(c), where increases in  $C_C$  values correspond with significant decreases in  $R_w$  values. In the same way, the carbon-formation reaction is seen to cause a decrease in the  $x$  values of WO<sub>3-x</sub>F<sub>x</sub>. This suggests that the  $x$  values of the WO<sub>3-x</sub>F<sub>x</sub> in Figure 1 (c) and the concomitant partial sublimation from WO<sub>3-x</sub>F<sub>x</sub> could not exceed a certain limit despite the continuous formation of CO<sub>2</sub> and CO during reactions.

For a deeper comprehension of the reaction process, the C=C bond strengths of C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> were ascertained through quantum mechanical calculations at the B3LYP/6-311+G(d) and MP2/6-311+G(d) levels to determine their :CF<sub>2</sub> dissociation

capabilities. The calculated bond orders, bond lengths, and wavenumber of C=C bonds of C<sub>2</sub>F<sub>4</sub>, C<sub>3</sub>F<sub>6</sub>, and *i*-C<sub>4</sub>F<sub>8</sub> are shown in Table 3 alongside those of C<sub>2</sub>H<sub>4</sub>, 2-*cis*-C<sub>4</sub>F<sub>8</sub>, 2-*trans*-C<sub>4</sub>F<sub>8</sub>, 1-C<sub>4</sub>F<sub>8</sub> for comparison. The table also shows the energy gap of C<sub>4</sub>F<sub>8</sub> olefin isomers with respect to *i*-C<sub>4</sub>F<sub>8</sub>. Here, the C<sub>2</sub>F<sub>4</sub> is seen to have a lower C=C bond order than C<sub>2</sub>H<sub>4</sub> – in agreement with the large interaction between the lone pair of fluorine and the antibonding  $\pi$ -orbital of the C=C bond.<sup>91-93</sup> The C=C bond order of C<sub>3</sub>F<sub>6</sub> is lower than that of C<sub>2</sub>F<sub>4</sub> as a result of the negative hyperconjugation of the resonance structure when CF<sub>3</sub> is introduced.<sup>94</sup> The bond order of *i*-C<sub>4</sub>F<sub>8</sub> is further lower than that of C<sub>3</sub>F<sub>6</sub> due to the second CF<sub>3</sub> group, in agreement with the lower wavenumber of the C=C stretching mode of *i*-C<sub>4</sub>F<sub>8</sub> compared to C<sub>3</sub>F<sub>6</sub> (see Figure 3(c) and Table 3 for experimental and computational results, respectively). Based on the theoretical data presented here, the higher reactivity of *i*-C<sub>4</sub>F<sub>8</sub> towards WO<sub>3-x</sub>F<sub>x</sub> in comparison to that of C<sub>3</sub>F<sub>6</sub> (Figures 3(c-i) and (c-ii)) can be attributed to the weaker C=C bond of *i*-C<sub>4</sub>F<sub>8</sub>. Conversely, the bond orders of *cis*-2-C<sub>4</sub>F<sub>8</sub> and *trans*-2-C<sub>4</sub>F<sub>8</sub> are larger than those of C<sub>3</sub>F<sub>6</sub> and even C<sub>2</sub>F<sub>4</sub> due to the different positions of the additional CF<sub>3</sub> group. This trend results from the negative hyperconjugation in *cis*-2-C<sub>4</sub>F<sub>8</sub> and *trans*-2-C<sub>4</sub>F<sub>8</sub>, which form carbocationic units that are stabilized by only single F atoms. The bond orders of 1-C<sub>4</sub>F<sub>8</sub> and C<sub>3</sub>F<sub>6</sub> are similar, corresponding to the similarities in their molecular structures except for CF<sub>3</sub> substituted

with  $C_2F_5$  in 1- $C_4F_8$ . Among the  $C_4F_8$  olefin isomers highlighted in the table, *i*- $C_4F_8$  is shown to have the weakest C=C bond. Besides, the calculated energy gaps  $\Delta E$  in Table 3 reveal that *i*- $C_4F_8$  formation is thermodynamically the most favorable among the  $C_4F_8$  olefin isomers highlighted. In light of this favorable *i*- $C_4F_8$  formation, the PTFE decomposition gas does not contain the other  $C_4F_8$  olefin isomers (see Figure 2(a) for the IR spectra of the PTFE decomposition gases).

Table 3. Summary of C=C bond parameters of C<sub>2</sub>H<sub>4</sub> and fluoroolefin species and energy gaps of C<sub>4</sub>F<sub>8</sub> isomers with respect to *i*-C<sub>4</sub>F<sub>8</sub> calculated at B3LYP/6-311+G(d,p) and MP2/6-311+G(d,p)

|          | C <sub>2</sub> H <sub>4</sub>   | C <sub>2</sub> F <sub>4</sub>  | C <sub>3</sub> F <sub>6</sub>   | <i>i</i> -C <sub>4</sub> F <sub>8</sub>   | 2- <i>cis</i> -C <sub>4</sub> F <sub>8</sub>  | 2- <i>trans</i> -C <sub>4</sub> F <sub>8</sub>                                      | 1-C <sub>4</sub> F <sub>8</sub>   |        |
|----------|---|--|---|---|---|---|---|--------|
| Molecule |  |  |  |  |  |  |  |        |
|          | C=C bond order  | 2.049  | 1.733   | 1.723   | 1.713   | 1.764   | 1.765   | 1.721  |
| DFT      | C=C bond length (Å)   | 1.329  | 1.322   | 1.330   | 1.334   | 1.333   | 1.332   | 1.331  |
| (B3LYP)  | Wavenumber for C=C <sup>a</sup> (cm <sup>-1</sup> )                               | (1684)   | (1909)  | 1824  | 1766  | 1765  | (1777)  | 1816   |
|          | $\Delta E^b$ (kJ mol <sup>-1</sup> )  | –  | –   | –   | 0   | 44.2  | 40.6  | 72.2   |
|          | C=C bond order  | 2.040  | 1.806   | 1.787   | 1.753   | 1.832   | 1.833   | 1.7842 |
| MP2      | C=C bond length (Å)   | 1.339  | 1.328   | 1.336   | 1.340   | 1.341   | 1.341   | 1.338  |
|          | Wavenumber for C=C <sup>a</sup> (cm <sup>-1</sup> )                               | (1675)   | (1922)  | 1843  | 1795  | 1768  | (1784)  | 1833   |
|          | $\Delta E^b$ (kJ·mol <sup>-1</sup> )  | –  | –   | –   | 0   | 55.0  | 52.8  | 84.5   |

<sup>a</sup>Wavenumbers in bracket indicate the vibration is IR inactive, <sup>b</sup>Energy gap from *i*-C<sub>4</sub>F<sub>8</sub>

**Solid-solid reaction between  $\text{WO}_3$  and  $\text{WO}_{3-x}\text{F}_x$ .** Until this juncture, the present results and discussion have extensively mentioned that the fluorination of  $\text{WO}_3$  occurs as a gas-solid reaction that involves gaseous  $i\text{-C}_4\text{F}_8$ ,  $\text{C}_3\text{F}_6$ , and  $\text{C}_2\text{F}_4$  to form  $\text{CO}_2$  and  $\text{CO}$ . According to a previous study, the degree of fluorination is dependent on the metal oxide used. For instance, the fluorination of some oxides such as  $\text{ZnO}$  and  $\text{Co}_3\text{O}_4$  only occurred on the surface, whereas other metal oxides such as  $\text{CuO}$  and  $\text{V}_2\text{O}_5$  underwent bulk fluorination.<sup>52</sup> Although the reaction in this study is instigated on the  $\text{WO}_3$  surface, fluorination was observed throughout the bulk of  $\text{WO}_3$ . This suggests the possible occurrence of  $\text{F}^-$  and  $\text{O}^{2-}$  diffusion within  $\text{WO}_3$  during the reaction at around  $500\text{ }^\circ\text{C}$ . To verify this phenomenon,  $\text{WO}_3$  and  $\text{WO}_{2.64}\text{F}_{0.36}$  powders (synthesized at  $500\text{ }^\circ\text{C}$  for a reaction time of 36 h and F/W of 6.00, Table 1) were pelletized in a weight ratio of 90:10 and heated at  $500\text{ }^\circ\text{C}$  for 24 h. The schematic of this solid-solid reaction and the XRD pattern of the reaction product is provided in Figure 4(a). The XRD pattern of the solid-solid reaction product is provided alongside those of the starting  $\text{WO}_3$  and  $\text{WO}_{2.64}\text{F}_{0.36}$  (Figure 4(b)). After the reaction, the diffraction peaks of  $\text{WO}_3$  are observed to completely disappear, resulting in a new diffraction pattern that is somewhat similar to that of  $\text{WO}_{2.64}\text{F}_{0.36}$ . Although the reaction product is not a single phase, it sufficiently demonstrates that  $\text{F}^-$  and  $\text{O}^{2-}$  diffusion occurs at  $500\text{ }^\circ\text{C}$  even in solid state, further

promoting the fluorination of  $\text{WO}_3$ .

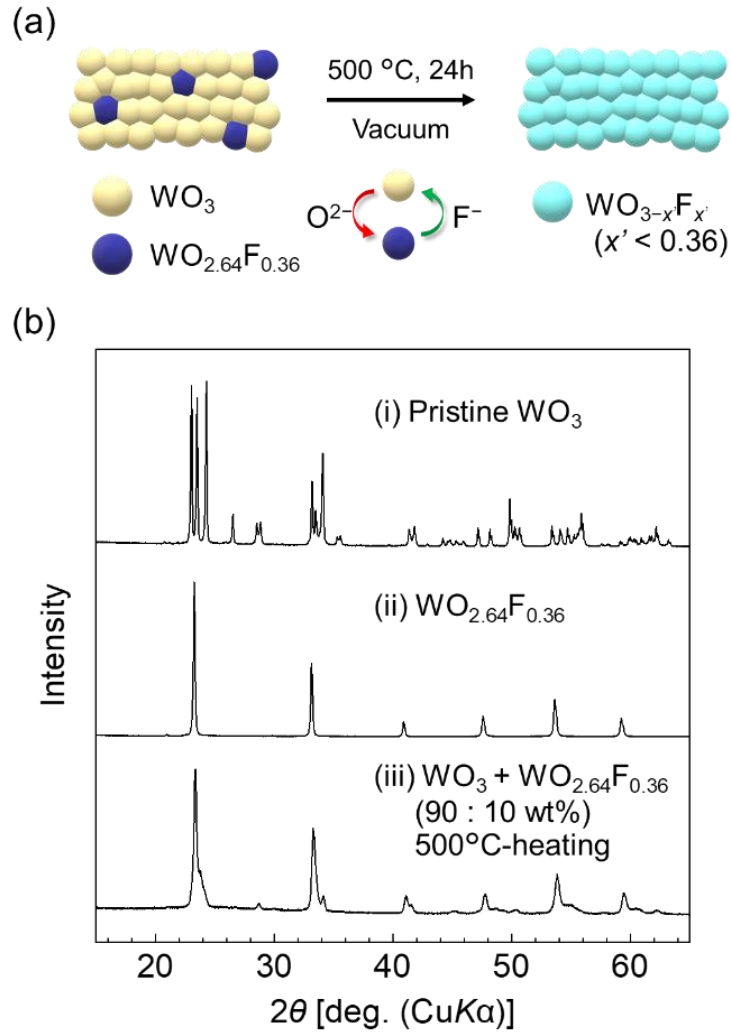


Figure 4. (a) Schematic of the solid-solid reaction between  $\text{WO}_3$  and  $\text{WO}_{2.64}\text{F}_{0.36}$ . (b) XRD patterns of (i) pristine  $\text{WO}_3$ , (ii)  $\text{WO}_{2.64}\text{F}_{0.36}$  (synthesized at 500 °C for 36 h with the F/W ratio of 6.00), and (iii) the product of the reaction between  $\text{WO}_3$  and  $\text{WO}_{2.64}\text{F}_{0.36}$  (90:10 wt%) at 500 °C for 24 h under vacuum (pelletized for solid-solid reaction).

Figure 5 illustrates a summary of the reaction mechanism between  $\text{WO}_3$  and PTFE-decomposition gases. As shown, the decomposition of PTFE at high temperatures produces four different gaseous species:  $\text{C}_2\text{F}_4$ ,  $\text{C}_3\text{F}_6$ , *i*- $\text{C}_4\text{F}_8$ , and *c*- $\text{C}_4\text{F}_8$ . The interaction between the  $\text{C}_2\text{F}_4$ ,  $\text{C}_3\text{F}_6$ , and *i*- $\text{C}_4\text{F}_8$  and the surface of  $\text{WO}_3$  results in the formation of  $:\text{CF}_2$ , which in turn induces reductive fluorination by exchanging the oxygen in  $\text{WO}_3$  with fluorine. During the reaction  $:\text{CF}_2$  carbene is also converted to CO via a O–2F exchange accompanied by sublimation of fluorine-containing W(VI) compounds and  $\text{CO}_2$  is formed via a 2O–2F exchange during the reaction. The  $\text{F}^-$  ion in the solid phase further diffuses from the surface to the bulk of  $\text{WO}_3$  by substituting  $\text{O}^{2-}$  eventually reaching a steady state of  $\text{WO}_{3-x}\text{F}_x$ .



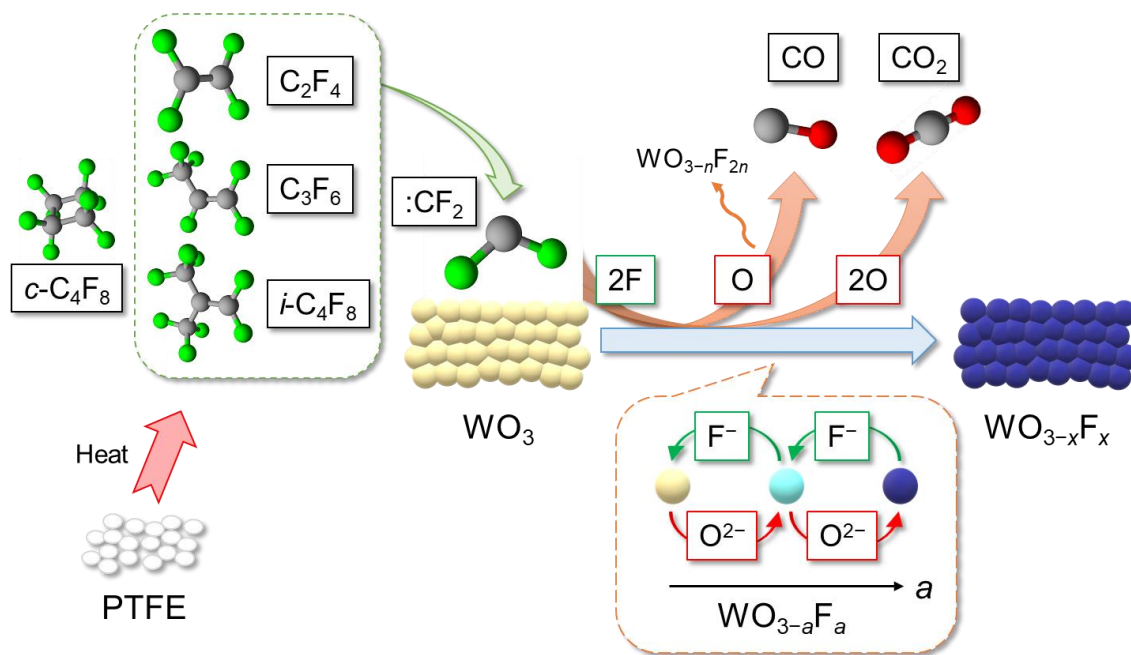


Figure 5 Schematic of the reaction mechanism for the reductive fluorination of  $\text{WO}_3$  by PTFE-decomposition gas.

#### 4. CONCLUSIONS

The present study investigates the mechanisms behind the reductive fluorination reaction between  $\text{WO}_3$  and PTFE-decomposition gases to optimize reaction conditions, identify reactive chemical species, and ascertain the pertinent gas-solid and solid-solid reaction paths. The reactions are conducted in nickel reactors as an alternative to quartz reactors to avoid reactions between PTFE-decomposition gas and quartz.

The reaction conditions were explored under one-step process, and appropriate reaction temperature, duration, and PTFE quantity (F/W) were found to be essential to obtain cubic  $\text{WO}_{3-x}\text{F}_x$  because formation of amorphous carbon byproduct and sublimation of fluorine-containing W(VI) compounds occurred at high temperature (550 °C under F/W = 3.56 and 36 h) and high PTFE quantity (F/W  $\geq$  10.00 under 500 °C and 36 h). The analysis of the PTFE decomposition gases performed at 25 °C and 500 °C (*in situ*) after the decomposition at 500 °C under vacuum confirmed the generation of  $\text{C}_2\text{F}_4$  at the first step and its conversion into *c*- $\text{C}_4\text{F}_8$ ,  $\text{C}_3\text{F}_6$ , and *i*- $\text{C}_4\text{F}_8$ . The two-step reaction process revealed that  $\text{C}_3\text{F}_6$  and *i*- $\text{C}_4\text{F}_8$  are also reactive species in addition to  $\text{C}_2\text{F}_4$ . The reductive fluorination reaction was induced by  $:\text{CF}_2$  formed by the cleavage of the C=C bond in the  $\text{C}_2\text{F}_4$ ,  $\text{C}_3\text{F}_6$ , and *i*- $\text{C}_4\text{F}_8$  upon contact with the surface of  $\text{WO}_3$ . The reaction between  $\text{WO}_3$  and  $:\text{CF}_2$  notably occurred via two mechanisms: (i) an O–2F exchange reaction that leads

to CO formation accompanied by the sublimation of fluorine-containing W(VI) compounds; and (ii) a 2O–2F exchange that produces CO<sub>2</sub>. Quantum chemical calculations showed that *i*-C<sub>4</sub>F<sub>8</sub> has a weaker C=C bond than C<sub>3</sub>F<sub>6</sub>, which is consistent with the higher reactivity of *i*-C<sub>4</sub>F<sub>8</sub> than that of C<sub>3</sub>F<sub>6</sub> confirmed in the two-step reaction. The formation of *i*-C<sub>4</sub>F<sub>8</sub> is thermodynamically the most favorable compared with that of the other C<sub>4</sub>F<sub>8</sub> fluoroolefin isomers, thereby rationalizing the presence of only *i*-C<sub>4</sub>F<sub>8</sub> among the C<sub>4</sub>F<sub>8</sub> isomers in the PTFE decomposition gases. The diffusion process of O<sup>2-</sup> and F<sup>-</sup> in the solid state was also confirmed by a direct reaction between WO<sub>3</sub> and WO<sub>2.64</sub>F<sub>0.36</sub> at 500 °C.

The reaction model proposed here provides an unprecedented insight into the reductive fluorination of metal oxides using PTFE. Although the optimal conditions and rate-determining steps are dependent on the metal oxide species, this report presents an fundamental framework for future explorations. Further, we expect this study to result in interest in reductive fluorination mechanisms involving different oxide materials and other fluoropolymers for more valuable insights.

## **SUPPORTING INFORMATION.**

The Supporting Information is available free of charge at DOI: xx. Details on some PTFE-related reactions; infrared and XPS spectra, XRD pattern and related structural analyses, and thermogravimetric curve of reaction products; photos of the reactor; thermodynamic and structural data of theoretical calculations (PDF).

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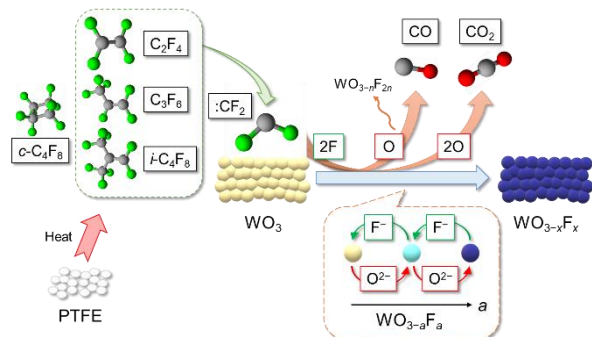
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Graphical Abstract (8.25 cm x 4.45 cm)



The reaction mechanisms behind the reductive fluorination of  $\text{WO}_3$  using polytetrafluoroethylene have been explored under varied reaction conditions (temperature, duration, and F/W ratio) to suppress the formation of carbon byproducts, minimize the dissipation of fluorine-containing tungsten (VI) compounds, and achieve a high fluorine content.