Title

Si emission from the SiO2/Si interface during the growth of SiO2 in the HfO2/SiO2/Si structure

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The downscaling of Si devices has led to the need for fine control techniques to form oxide layer as thin as 1 nm. Therefore, an understanding of Si oxidation is strongly demanded by the industry as well as in materials science. This is also the case even if SiO2 is replaced by high-k materials in future Si devices. An ultrathin (sub nanometer) SiO2 layer is intentionally grown on Si before the formation of the high-k film to get better device performance. Thus, the understanding of Si oxidation is still an important issue.

The oxidation of relatively thick (>10 nm) films is well described by the so-called Deal-Grove model. The oxidation kinetics for very thin films, however, cannot be explained by the Deal-Grove model. The oxidation rate for thin films is much faster than that expected from a linear relationship predicted by the Deal-Grove model. The failure of the Deal-Grove model may be related to the large volume expansion upon oxidation. During oxidation of Si, oxidant diffuses through the disordered oxide network and reacts with silicon substrate at the SiO2/Si interface. The volume of the increased oxide region is about twice larger than that of the decreased Si region. The volume expansion results in high stress on the interface between Si and SiO2, which should be released when the oxidation proceeds. Recent theoretical studies predicted that Si species are emitted from the SiO2/Si interface to release the stress induced by oxidation as well as in materials science.1

Recent theoretical studies predicted that Si species are emitted from the SiO2/Si interface to release the stress induced by oxidation as well as in materials science.4 The oxidation rate for thin films is much faster than that expected from a linear relationship predicted by the Deal-Grove model. The failure of the Deal-Grove model may be related to the large volume expansion upon oxidation. During oxidation of Si, oxidant diffuses through the disordered oxide network and reacts with silicon substrate at the SiO2/Si interface. This Si species can outdiffuse through a HfO2 film.16 The emitted Si species from the SiO2/Si interface, therefore, are expected to accumulate at the surface of the HfO2 film. Thus, the emitted Si can be separately observed from the interfacial SiO2 layer using HRBS.

A HfO2 film of 4 nm thickness was grown by atomic layer deposition (ALD) using tetrakis(ethylmethylamino)hafnium (H[3N(CH2)(C2H5)]4) with O2 as an oxidant after preparing a SiO2 layer of 0.7 nm thickness on Si(001). In order to prepare a high quality HfO2 film, the sample was annealed in O3 at 275 °C for 10 min after deposition. The samples thus prepared were annealed in an infrared furnace (MLA 3000) at 500–900 °C in 0.1 Torr dry oxygen for 2–20 min. For comparison, the sample was annealed also in an ultrahigh vacuum chamber (base pressure 4 × 10−10 Torr) at 800 °C for 2 min.

These samples were observed by HRBS using 400 keV He+ ions as a probe. The details of the HRBS measurement were described elsewhere.18 In the present measurements, samples were aligned to a channeling configuration with the [111] axis parallel to the incident beam, reducing the substrate contribution to the signal. Energy spectra of He+ ions scattered at 50° were measured by a 90° sector magnetic spectrometer.

Figure 1 shows the observed HRBS energy spectra. The solid line shows the spectrum of the as-grown sample. There are three peaks in the spectrum. The large peak at ~390 keV corresponds to Hf in the HfO2 layer. The peak at ~350 keV corresponds to Si in the SiO2/Si interface region. The Si interface peak consists of contributions from the interfacial SiO2 layer and the substrate Si atoms which can be seen by the channeling ions. The latter contribution mainly comes from the topmost Si atomic layer in the substrate Si (this is...
essentially the same as the so-called surface peak in a usual channeling spectrum) and is almost independent of the thickness of the interfacial SiO2 layer unless the SiO2 layer is extremely thick. The increase of the Si interface peak yield is, therefore, attributed to the growth of the SiO2 layer. The third peak seen at \( \sim 330 \text{ keV} \) corresponds to O in both the HfO2 and the interfacial SiO2 layers. The thickness of the interfacial SiO2 layer is estimated to be 0.7 nm from the observed Si peak. In contrast to the Si and the O peaks, the shape of the Hf peak does not change but a small shift \((\sim 0.3 \text{ keV})\) towards lower energies is observed. A similar shift was also observed for the leading edge of the Si interface peak. These shifts suggest the formation of an overlayer on the HfO2 surface. In addition to these changes, a new peak appears around 330 keV. No pinhole was observed, confirming the formation of the interfacial SiO2 layer.

The spectrum of the sample annealed at 800 °C in O2 (short dashed line in Fig. 1) is quite different from that of the as-grown sample. Both the Si peak as well as the O peak become wider after annealing, indicating the growth of the interfacial SiO2 layer. The increase of the interfacial SiO2 layer is estimated to be 0.6 nm from the observed Si peak. In contrast to the Si and the O peaks, the shape of the Hf peak does not change but a small shift \((\sim 0.3 \text{ keV})\) towards lower energies is observed. A similar shift was also observed for the leading edge of the Si interface peak. These shifts suggest the formation of an overlayer on the HfO2 surface. In addition to these changes, a new peak appears around 330 keV.

Only the signals for Si and O are shown because the Hf peak did not change except for the small energy shift mentioned above. Widths of both Si and O peaks increase with increasing temperature, indicating that the thickness of the interfacial SiO2 layer increases with annealing temperature. The yield of the surface Si also increases with increasing temperature, showing that there is a strong correlation between the growth of the interfacial SiO2 layer and the SiO2 overlayer. This strongly suggests that the origin of the surface Si is the Si emission from the SiO2/Si interface during the growth of the interfacial SiO2 layer. There are, of course, other possibilities, e.g., the heating itself may cause the surface accumulation of Si independent of the oxidation. In order to see if this is the case, the sample was annealed in vacuum at 800 °C for 2 min. The observed HRBS spectrum (a dashed line in Fig. 1) is very similar to the spectrum of the as-grown sample. Neither the surface Si peak nor the growth of the interfacial layer is seen. This clearly indicates that the observed surface Si is a result of the oxidation and is not a result of the heating itself.

Finally, we studied the annealing time dependence of the Si emission. The samples were annealed at 900 °C in 0.1 Torr oxygen for various annealing times. The increase of the Si areal density of the surface SiO2 layer, \( \Delta D_s \), and that of the interfacial SiO2 layer, \( \Delta D_i \), were measured and the results are shown as a function of the annealing time in Fig. 3. The ratio of the increase of the surface Si to that of the interfacial Si, \( \Delta D_s/\Delta D_i \), is also shown to clear up their quantitative relation. The interfacial SiO2 layer increases with increasing annealing time. This growth rate slows down when the annealing time exceeds 5 min. The surface Si also increases with annealing time but saturates after 5 min. This means that more than 23% (\( =0.3/1.3 \)) of Si atoms are emitted from the SiO2/Si interface during oxidation. This is in good agreement with the result of the first principles calculation, which showed that one Si atom is kicked out of the SiO2/Si interface when every three Si atoms are oxidized.\(^{11} \)
The observed ratio, $\Delta D_s/\Delta D_i$, decreases with increasing annealing time, indicating that a part of emitted Si atoms is incorporated in the interfacial SiO$_2$ layer. After 5 min the surface Si does not increase while the interfacial SiO$_2$ layer still grows. This means that the emitted Si species are almost completely incorporated in the SiO$_2$ layer and cannot reach the surface when the thickness of the SiO$_2$ layer exceeds $\sim$2 nm in the present conditions (900 °C in 0.1 Torr oxygen).

In conclusion, the growth of the interfacial SiO$_2$ layer and the simultaneous surface accumulation of Si were observed when HfO$_2$/SiO$_2$/Si was annealed in oxygen. A strong correlation was found between the growth of the interfacial SiO$_2$ layer and the Si surface accumulation. The observed result indicates that silicon species are emitted from the SiO$_2$/Si interface during the growth of the interfacial SiO$_2$ layer to release the stress which is induced by oxidation. The fraction of the emitted Si was estimated to be more than 23%, which is consistent with the recent theoretical studies.

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