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Photoinduced intramolecular charge separation in a polymer solid below the glass transition temperature

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Photoinduced intramolecular charge separation (CS) in a polar polymer glass, cyanoethylated pullulan (CN-PUL), was studied below the glass transition temperature \( T_g = 395 \) K. A series of three carbazole (Cz: donor)-cyclohexane (S: spacer)-acceptor (A: acceptor) molecules (Cz-S-A) was used as intramolecular donor-acceptor dyads. The photoinduced CS rate was evaluated by the fluorescence decay measurement at temperatures from 100 to 400 K. The CS rate \( k_{\text{CS}} \) increased above 200 K even far below \( T_g \) where micro-Brownian motions of the whole polymer chain are frozen. Below 200 K, on the other hand, \( k_{\text{CS}} \) showed weak dependence on temperature. The temperature dependence of \( k_{\text{CS}} \) is discussed in terms of the dielectric relaxation time of the polymer matrix. Consequently, CS below \( T_g \) was well explained by a thermally nonequilibrium electron transfer (ET) formula above 200 K and by a two-mode quantum-mechanical ET formula below 200 K. The increase in \( k_{\text{CS}} \) above 200 K is mainly caused by a thermally activated low-frequency matrix mode originating from the side-chain relaxation of polar cyano groups. The weak temperature dependence of \( k_{\text{CS}} \) can be explained by a nuclear-tunneling effect caused by a high-frequency matrix mode \((\hbar\omega_Q = 250 \text{ cm}^{-1}\) \) and an intramolecular vibrational mode \((\hbar\omega_Q = 1300 \text{ cm}^{-1}\) \). The high-frequency mode of the polymer matrix was attributed to a vibrational or librational motion of polar groups in the CN-PUL glassy solid. © 2005 American Institute of Physics.

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I. INTRODUCTION

Electron transfer (ET) in a polymer solution is controlled by various factors: distance \( r \) from a donor (D) to an acceptor (A), electronic coupling \( V \) of the reactant with the product, free-energy gap of the ET reaction \(-\Delta G^0\), reorganization energy \( \lambda \), solvent polarity, dielectric relaxation time of the solvent \( \tau \), and temperature \( T \).\(^{1-8}\) In particular, the thermal fluctuation of polar solvent molecules leads the reactant to a transition state by exchanging energy and momentum with the reactant. The ET theory developed by Marcus is based on the assumption that a thermal equilibrium is established in the solvent fluctuation during the ET event. Under the thermal equilibrium condition, the ET rate \( k_{\text{ET}} \) is determined not by the solvation dynamics, that is, \( \tau \), but by the static polarity of the solvent. The solvent can be treated as a dielectric continuum and its static polarity is characterized by a static dielectric constant \( \varepsilon_0 \) which dominates \( k_{\text{ET}} \) through \(-\Delta G^0 \) and \( \lambda \). The Marcus theory well describes the various ET reactions in solutions.\(^{2,3,6-9}\) In general, the thermal equilibrium assumption is valid for the ET reaction in a solution where the solvent motion with a dielectric relaxation time of \( \tau \) is sufficiently faster than the ET event, \( 1/\tau \gg k_{\text{ET}} \). On the other hand, the assumption breaks down if the solvent relaxation is equal to \((1/\tau \sim k_{\text{ET}})\) or slower than \((1/\tau < k_{\text{ET}})\) the ET event. Under these thermally nonequilibrium conditions, \( k_{\text{ET}} \) will be dependent on \( \tau \).\(^{7,8,10-14}\) In fact, there have been several reports on the solvent-controlled ET.\(^{15-18}\) Heitele and co-workers studied photoinduced intramolecular charge separation (CS) in highly viscous propylene glycol solvent.\(^{15,16}\) They found a smooth transition of CS from nonadiabatic to solvent-controlled adiabatic behavior with an increase in the solvent relaxation time \( \tau \) at lower temperatures. Recent progress in spectroscopic instruments allows one to directly observe the ultrafast ET in a time domain faster than a picosecond where the thermal equilibrium is not established yet.\(^{7,11-14,19-30}\) Some studies reported that \( k_{\text{ET}} \) is proportional to \( 1/\tau \).\(^{7,13,14,19,20,24,26}\) Others reported the existence of ET faster than orientational motions of the solvent.\(^{7,13,14,21-25,27-30}\) The ultrafast ET, which is faster than \( 1/\tau \), indicates that there must be nuclear motions faster than the orientational motions to satisfy the energy conservation law during the ET process. To explain the ultrafast ET, several researchers introduced intramolecular vibrational motions \(^{10,22,23,29-32}\) or fast solvent motions \(^{12-14,24,25,27,28,33-35}\) such as molecular vibration, inertial motion, and libration as another reaction coordinate under thermal equilibrium in ad-
dian to the reaction coordinate of thermally nonequilibrium solvent orientational motions. Recently, fast solvent motions contributing to solvation have been revealed by subpicosecond time-dependent fluorescence Stokes shift measurements or a variety of nonlinear spectroscopic methods. These solvent motions interact with the reactant on a time scale faster than 0.1 ps and they have a temperature-independent relaxation time. Some theoretical ET models introducing these fast motions predict that $k_{ET}$ is mainly dominated by the fast motions rather than following the orientational motions of the solvent. Even now, however, the influence of those solvent dynamics on the ET reaction is not understood completely.

Compared with these extensive studies on ET in solutions under both thermal equilibrium and thermally nonequilibrium conditions, there have been few experimental studies on ET in polymer solids. Different from a polar Debye solvent, polymer solids have various dielectric relaxation modes and each mode has a wide distribution in $\tau$; motions of a long segment in the main chain, local motions of a short segment in the main chain, rotational motions of a polar group in the side chain, and damped oscillations of a polar group. Above the glass transition temperature ($T_g$), the motions of the whole polymer chains are allowed and the side-chain relaxation time merges with the main chain. Our previous studies showed that photoinduced CS above $T_g$ can be explained by the ET formula with a quantum mode of intramolecular vibration and a classical mode of low-frequency orientational motions of the polymer matrix under a thermal equilibrium condition. In other words, even in polymer solids, thermal fluctuations of polar groups leading to the ET reaction can be considered in a thermal equilibrium above $T_g$.

Below $T_g$, on the other hand, the motion of the main chain is frozen and dielectric relaxations are mainly due to a variety of local motions induced by the side-chain relaxation. Inhomogeneity in the relaxation time is more remarkable in the side-chain relaxation rather than the main-chain relaxation. Thus, it is difficult to specifically determine the relaxation time $\tau$ that contributes to the ET event below $T_g$. In polymer glass, most part of the dielectric relaxation is slower than the ET event. Providing that ET is effectively promoted by these slow motions, it would proceed under a thermally nonequilibrium condition. However, even below $T_g$, local fast fluctuations still remain and a high-frequency tail of dielectric relaxation extends into a time scale faster than nanoseconds. In fact, there is a temperature-independent solvation owing to inertial motions in polymer glass as in a solution. Providing that ET is effectively promoted by these fast motions, it would proceed under a thermal equilibrium condition. Owing to the wide distribution of dielectric relaxations in polymer glass, ET in polymer solids should be different from that in a polar Debye solution. However, there have been no systematic studies dealing with this problem.

Here we studied photoinduced CS in carbazole-spacer-acceptor (Cz-S-A) dyad molecules embedded in a cyanoethylated pullulan (CN-PUL) polymer glassy solid. Dielectric motions in the polymer were examined at relaxation time $\tau$ with the Cole-Cole distribution function $\phi(\tau)$. The averaged CS rate $\langle k_{CS}(\tau) \rangle$ was calculated from an integral of $k_{CS}(\tau)$ weighted by $\phi(\tau)$ over a wide range of $\tau$. The CS of Cz-S-A is suggested to be effectively coupled to fast dielectric motions corresponding to the lifetime of the Cz donor in the excited state. The effective dielectric constant $\varepsilon_{eff}$ coupled with CS can be directly evaluated by using a fluorescent polarity probe of carbazole-terephthalate cyclophane (Cz-TP). Consequently, the photoinduced CS in the glassy polymer solid was well explained by the thermally nonequilibrium ET formula from $T_g$ to 200 K and by the two-mode quantum-mechanical ET formula under the thermal equilibrium condition below 200 K. The quantum-mechanical ET formula includes a high-frequency mode of the polymer matrix and a high-frequency intramolecular vibrational mode.

II. EXPERIMENT

A. Materials

1. Intramolecular donor-spacer-acceptor compounds

A series of three carbazole (Cz: donor)-cyclohexane (S: spacer)-acceptor (A: acceptor) (Cz-S-A) molecules was synthesized and used as intramolecular D-A dyads. N-Ethylcarbazole (EtCz) was synthesized and used as a reference. Details of the synthesis have been described elsewhere.

2. Cz-TP fluorescent polarity probe

Cz-TP was synthesized by photodeselenation of the selena carbazolophane. Details of the synthesis have been described elsewhere.

3. Cyanoethylated pullulan

As a polymer matrix, cyanoethylated pullulan (CN-PUL) was used in this study, which was kindly donated from Shin-Etsu Ltd. The glass transition temperature of CN-PUL was $T_g = 395$ K, which was measured by differential scanning calorimetry. The refractive index of CN-PUL film was $n = 1.50$ at 300 K, which was measured by using an Abbe refractometer (Atago) with an index-matching fluid ($n_{DP} = 1.520$ at 298 K). The CN-PUL was purified by reprecipitation from an acetone solution into hexane three times before use.

Figure 1 shows the chemical structures of Cz-S-A, EtCz, Cz-TP, and CN-PUL polymer.

B. Methods

1. Preparation of polymer film samples

Polymer films of CN-PUL doped with EtCz or Cz-S-A were prepared by the solution casting method. The EtCz and Cz-S-A compounds were dissolved in an acetonitrile (Nacalai Tesque, spectroscopic grade) solution of CN-PUL so that the concentration of chromophores in the final polymer film was in the order of $10^{-3}$ mol L$^{-1}$. The polymer solution was dropped onto a quartz plate (15 $\times$ 15 $\times$ 1 mm$^3$), dried for several days under atmospheric pressure at room temperature, and for 12 h under reduced pressure above $T_g$. Polymer films had a thickness of about 50 $\mu$m.
2. Fluorescence decay measurements

Fluorescence decay was measured by the time-correlated single-photon-counting method. The excitation light source was third-harmonic pulses (295 nm) generated from a mode-locked Ti:sapphire laser (Spectra-Physics, Tsunami 3950) that was pumped by an Ar+ laser (Spectra-Physics, BeamLok 2060). The fluorescence emission was detected with a photomultiplier (PMT; Hamamatsu Photonics, R3234) through a monochromator (Ritsu, MC-10N) with a cut-off filter (UV-34) for the excitation light. The details of this apparatus have been described elsewhere. The total instrument response function has a full width at half maximum of ca. 750 ps at the excitation wavelength. The decay data were fitted with sums of exponentials that were convoluted with the instrument response function by the nonlinear least-squares method.

3. Temperature control

The temperature of film samples cast on a quartz plate was controlled in a cryostat (Iwatani Plantech Corp., CRT510) with a temperature control unit (Scientific Instruments, Model 9650).

4. Dielectric relaxation measurements of CN-PUL

Dielectric relaxation spectra were measured for the CN-PUL film with a LCR meter (Ando Electrics, Japan) at a temperature of 100–400 K and frequency of 0.1–100 kHz.

Fluorescence decays $I(t)$ of a reference EtCz and Cz-S-A were measured from 100 to 400 K. Figure 2 shows the temperature dependence of $I(t)$ for EtCz and Cz-S-CN. The fluorescence decay of EtCz (broken line) was independent of temperature from 100 to 400 K. On the other hand, the fluorescence of Cz-S-CN decayed faster at temperatures higher than 200 K. At temperatures below 200 K, $I(t)$ of Cz-S-CN was close to that of EtCz, but even at 100 K there remained a clear difference in $I(t)$ between Cz-S-CN and EtCz. A similar temperature dependence of the fluorescence decay was observed for Cz-S-2CF$_3$. The fluorescence decay of Cz-S-NO$_2$ was rapid and independent of temperature. As shown in Fig. 2, $I(t)$ did not obey a single-exponential decay over the whole temperature range measured. The multiexponential decay of $I(t)$ is probably due to a temporal distribution of the dielectric relaxation time around Cz-S-A molecules in CN-PUL rather than spatial inhomogeneity, because the deviation from a single-exponential decay was more remarkable at higher temperatures. All the decay curves of Cz-S-A were well fitted with the sum of three exponential functions [Eq. (1)]

$$I(t) = I_0 \sum A_i \exp(-t/\tau_i).$$

Photoinduced CS rate $k_{CS}$ was calculated by $k_{CS} = 1/\langle \tau \rangle - 1/\tau_f$, where $\tau_f$ is the fluorescence lifetime of EtCz and $\langle \tau \rangle$ is the averaged fluorescence lifetime of Cz-S-A.

Figure 3 shows the temperature dependence of $k_{CS}$ of the three Cz-S-A molecules. For Cz-S-NO$_2$, $k_{CS}$ was invariant over the whole temperatures measured from 100 to 400 K. In contrast, $k_{CS}$ of Cz-S-2CF$_3$ and Cz-S-CN increased with increasing temperature above 200 K while it was weakly dependent on temperatures below 200 K. The critical temperature $T_c=200$ K, far below $T_c=395$ K, corresponds to a subtransition temperature of CN-PUL, which is ascribed to a local relaxation of side chains with polar cyano groups.

The polar cyano groups having a large dipole moment are
Thus, the increase in solid.

locating at the end of the side chains where there is free volume and larger flexibility. This is why the local motions of the polar cyano groups are activated even far below $T_c$. Thus, the increase in $k_{CS}$ of Cz-S-2CF$_3$ and Cz-S-CN above $T_c$ is also ascribable to the local orientational motions of the polar cyano groups in the side chains. This dielectric behavior differs from that of cyanoethylated $O$-(2-hydroxypropyl)cellulose (CN-HPC) and poly(alkyl methacrylate) where most dielectric motions are frozen below the $T_c$. Above $T_c$, photoinduced CS of Cz-S-A in CN-HPC and poly(alkyl methacrylate) solids could be explained by the thermal equilibrium.\textsuperscript{53,54} In CN-PUL glass, however, a wide distribution in $\tau$ should be considered to discuss CS below $T_c$ because there are various dielectric motions not only slower but also faster than the CS event. In Sec. III B, the temperature dependence of $k_{CS}$ will be discussed in detail considering the wide distribution in the relaxation time of the CN-PUL glassy solid.

**B. Analysis of $k_{CS}$ based on dielectric relaxation dynamics of CN-PUL glassy solid**

The rate of nonadiabatic CS is expressed by Eqs. (2) and (3).\textsuperscript{2,67} Here, a static dielectric constant $\varepsilon_0$ of the solvent matrix represents thermal fluctuation of polar solvent molecules. Considering the contribution of a quantum effect involving a high-frequency intramolecular vibration to the CS reaction, $k_{CS}$ is expressed by a sum of nonadiabatic CS rates $k_{NA}^{0-n}$ from the reactant in the ground vibrational state to the product in the $n$th excited vibrational state,

$$k_{CS}^N = \sum_n k_{NA}^{0-n},$$

$$k_{NA}^{0-n} = \frac{2\pi}{\hbar} \sqrt{\frac{1}{4\pi\varepsilon_0 k_B T}} \frac{S^n \exp(-S)}{n!} \times \exp \left[ -\frac{(\Delta G_{CS} + \varepsilon S + nh\omega_Q)^2}{4\lambda_s k_B T} \right],$$

$$S = \frac{\lambda^2}{\hbar \omega_Q},$$

where $S$ is the electronic coupling matrix element, and $S$ is the electron vibrational coupling strength given as Eq. (4) by using both reorganization energy for intramolecular vibration $\lambda_Q$ and its vibrational quantum $\hbar \omega_Q$. The free-energy gap for the photoinduced CS $\Delta G_{CS}$ and reorganization energy for solvent $\lambda_S$ are given by Eqs. (5) (Refs. 68 and 69) and (6), respectively, where $E(S)$ is the energy level of the lowest excited singlet state of EtCz, $E_{ox}$ and $E_{red}$ oxidation potential of donor (Cz) and reduction potential of acceptor (A), respectively, $\varepsilon$ the elementary charge, $\varepsilon_0$ the vacuum permittivity, $\varepsilon_S$ static dielectric constant of the matrix, $r_{DA}$ separation distance between the donor and acceptor, $r_D$ and $r_A$ spherical radii of donor and acceptor moieties, respectively, and $\varepsilon_\infty$ optical dielectric constant of the matrix which is equal to the square of the refractive index of the matrix $n^2$,

$$k_{CS}(\tau) = \sum_n \frac{1}{1 + \kappa_{0-n}(\tau)} k_{NA}^{0-n},$$

$$k_{0-n}(\tau) = \frac{4\pi\tau^2 V^2 S^n \exp(-S)}{\hbar \lambda_S n!}.$$
binary unit. Figure 4 shows the temperature dependence of \( \tau_m \) (closed circles) and \( \beta \) (triangles). First, \( k_{CS} \) of Cz-S-2CF\(_3\) was calculated by the ET formula considering the thermally nonequilibrium effect [Eq. (7)] with the average dipolar relaxation time \( \tau_m \) because of the wide distribution in relaxation time. This approximation by a single relaxation time is generally used for a polar Debye solvent. In the calculation, parameters of \( V \), \( \lambda^0 \), and \( h_0\alpha_0 \) were used as follows: \( V = 3.5 \) cm\(^{-1} \), \( \lambda^0 = 0.56 \) eV, and \( h_0\alpha_0 = 1300 \) cm\(^{-1} \).\(^{53,54} \) The static dielectric constant \( \varepsilon_s \) in Eqs. (5) and (6) was evaluated by the dielectric analysis with the Cole-Cole function. The optical dielectric constant \( \varepsilon_\infty \) was calculated as the square of the refractive index \( n^2 \) of CN-PUL \((n = 1.50)\) and assumed to be constant over the whole temperature range. As shown in Fig. 5 (dashed-dotted line), \( k_{CS}(\tau_m) \) fell off with decreasing temperature and considerably deviated from the experimental result. The large deviation is clearly due to the disregard of a variety of relaxation times except for \( \tau_m \) in the CN-PUL glassy solid.

Next, the averaged CS rate \( \langle k_{CS}(\tau) \rangle \) was calculated from the integral of \( k_{CS}(\tau) \) weighted by the distribution function \( \phi(\tau) \) over \( \tau \) [Eq. (10)] to discuss the effect of the wide distribution in dielectric relaxation time \( \tau \) on the CS rate. In a Cole-Cole formula, \( \phi(\ln \tau) \) is expressed by Eq. (11).\(^{48,50,61} \)

\[
\langle k_{CS}(\tau) \rangle = \int_0^{\infty} k_{CS}(\tau) \phi(\tau) d\tau,
\]

\[
\phi(\ln \tau) = \frac{1}{2\pi} \sin \beta \pi \left[ \frac{1}{\cosh[\beta \ln(\tau/\tau_m)] + \cos \beta \pi]}.\right.
\]

Compared with \( k_{CS}(\tau_m) \), theoretical prediction of \( \langle k_{CS}(\tau) \rangle \) (broken line in Fig. 5) demonstrated a marked increase in CS rate above 200 K and a weak temperature dependence below 200 K, although \( \langle k_{CS}(\tau) \rangle \) is still much smaller than the experimental result. This improvement suggests that CS in the CN-PUL glassy solid is coupled not to a representative motion with a relaxation time of \( \tau_m \) but to a variety of motions. The remaining deviation of \( \langle k_{CS}(\tau) \rangle \) is probably due to slow relaxation motions, which should be considered to be frozen during the lifetime of the excited initial state. This time scale is comparable to the lifetime of EtCz in the excited state. Thus, dielectric dynamics with a relaxation time faster than the lifetime of EtCz in the excited state (13 ns) have to be evaluated.

To discuss CS in CN-PUL below \( T_s \) on the basis of the effective dielectric dynamics with a relaxation time faster than the lifetime of EtCz in the excited state, we need an effective dielectric constant \( \varepsilon_{\text{eff}} \) and effective distribution function \( \phi(\tau)_{\text{eff}} \) corresponding to the CS reaction. As for \( \varepsilon_{\text{eff}} \), it was previously demonstrated that the fluorescence polarity probe method using the carboxyl-terephthalate cyclophane (Cz-TP) is a powerful tool for evaluating \( \varepsilon_{\text{eff}} \) reflecting dielectric motions in a time domain shorter than the lifetime of EtCz in the excited state.\(^{62} \) Therefore, \( \varepsilon_{\text{eff}} \) of the CN-PUL glassy solid was evaluated with the Cz-TP probe whose lifetime in the excited state is 35 ns.\(^{72} \) As for \( \phi(\tau)_{\text{eff}} \), the distribution function \( \phi(\tau) \) should be modified according to the time window of the initial excited state, because not all dielectric motions can contribute to CS in Cz-S-A. The dielectric motions contributing to CS should be limited to those with a relaxation time \( \tau \) shorter than the lifetime of EtCz in the excited state. Thus, we introduced a modified distribution function \( \phi(\tau)_{\text{eff}}(I) \) as \( \phi(\tau)_{\text{eff}} \). Here, the fluorescence decay \( I(\tau) \) of EtCz represents the population of EtCz in the excited state at a time \( \tau \). Figure 6(a) shows the distribution function of the side-chain relaxation time \( \phi(\tau) \) of CN-PUL (solid line) at each temperature and the fluorescence decay of EtCz \( I(\tau) \) (broken line). As shown in the figure, most part of the \( \phi(\tau)_{\text{eff}} \) at each temperature is distributed in a time domain longer than that represented by the broken line \( I(\tau) \), that is, the time window of the lifetime of EtCz in the excited state. This is why \( \langle k_{CS}(\tau) \rangle \) deviates from the experimental results. On the other hand, as shown in Fig. 6(b), the time domain of the effective distribution function \( \phi(\tau)_{\text{eff}}(I) \) is consistent with the time domain of \( I(\tau) \). Thus, the CS rate was evaluated by Eq. (12) with \( -\Delta G_{CS}(\varepsilon_{\text{eff}}) \) and \( \lambda_{\varepsilon}(\varepsilon_{\text{eff}}) \).

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FIG. 4. Temperature dependence of the average dipolar relaxation time \( \tau_m \) (closed circles) and a parameter \( \beta \) describing the distribution of relaxation time (triangles). These are obtained by analyzing dielectric relaxation spectra of the CN-PUL with a single Cole-Cole function [Eq. (9)].

FIG. 5. Temperature dependence of the CS rate calculated by each ET equation. The dashed-dotted line represents \( k_{CS}(\tau_m) \) [Eq. (10)], the solid line \( k_{CS}(\tau)_{\text{eff}} \) [Eq. (12)], the dotted line \( k_{CS}(\tau_{\text{eff}})^{\text{NC}-\text{et}} \) [Eq. (2)], and circles are the experimental CS rate of Cz-S-2CF\(_3\). Fitting parameters in each equation were \( V = 3.5 \) cm\(^{-1} \), \( \lambda^0 = 0.56 \) eV, and \( h_0\alpha_0 = 1300 \) cm\(^{-1} \). Both \( -\Delta G_{CS} \) and \( \lambda_{\varepsilon} \) were evaluated with \( \varepsilon_s \) for \( k_{CS}(\tau_m) \) and \( \langle k_{CS}(\tau_{\text{eff}}) \rangle \) and with \( \varepsilon_{\text{eff}} \) for \( k_{CS}(\tau)_{\text{eff}} \) and \( \langle k_{CS}(\tau_{\text{eff}}) \rangle_{\text{eff}} \) respectively.
shows that dielectric motions with a relaxation time $\tau$ cause a nonadiabatic CS and therefore, within the time domain of thermally nonequilibrium CS and $k_{eff}$, solid lines of Cz-S-2CF$_3$ at 300 K calculated by Eq. (7).

$$k_{CS}(\tau)_{eff} = \int_{0}^{\infty} k_{CS}(\tau) \phi(\tau) I(\tau) d\tau.$$  \hspace{1cm} (12)

As shown in Fig. 5 (solid line), $k_{CS}(\tau)_{eff}$ was in good agreement with the experimental CS rate from 200 K to $T_g$. The nonadiabatic CS rate $k_{CS}^{NA}_{eff}$ was also calculated by Eq. (2) with $-\Delta G_{CS}(e_{eff})$ and $k_{CS}^{NA}$, and decreased linearly with temperature whereas the experimental CS rate shows a nonlinear dependence on temperature. The deviation of $k_{CS}^{NA}_{eff}$ from the experimental CS rate and the well reproduction of that by thermally nonequilibrium $k_{CS}(\tau)_{eff}$ indicates that CS in Cz-S-2CF$_3$ proceeds under thermally nonequilibrium condition from 200 K to $T_g$. In other words, dielectric relaxation time influences the CS rate.

For the discussion on the contribution of the effective dielectric motions to thermally nonequilibrium CS in more detail, the relaxation time $\tau$ dependence of CS rate $k_{CS}(\tau)$ of Cz-S-2CF$_3$ was calculated by using Eq. (7) at 300 K [dashed-dotted line in Fig. 6(b)]. The relaxation time dependence of $k_{CS}(\tau)$ shows that dielectric motions with a relaxation time $\tau < 10^{-9}$ s cause a nonadiabatic CS and $k_{CS}(\tau)$ is independent of $\tau$ while those with a relaxation time $\tau > 10^{-9}$ s cause a thermally nonequilibrium CS and $k_{CS}(\tau)$ depends on $\tau$. Therefore, within the time domain of $\phi(\tau)_{eff}$, dielectric motions contributing to thermally nonequilibrium CS are limited to those with a relaxation time $\tau > 10^{-9}$ s above 200 K. On the other hand, there still remains disagreement below 200 K. At lower temperatures, the CS rate shows a weak temperature dependence while $k_{CS}(\tau)_{eff}$ and $k_{CS}^{NA}_{eff}$ predict a monotonous decrease with decreasing temperature. Another explanation should be considered for the weak temperature dependence of the CS rate below 200 K.

![FIG. 6. Temperature dependence of dielectric distribution function from 200 to 330 K. (a) the distribution function of the side-chain relaxation time in CN-PUL $\phi(\tau)$ (solid lines) calculated by Eq. (11), and (b) the effective distribution function $\phi(\tau)I(\tau)$ (solid lines) obtained by multiplying $\phi(\tau)$ by the fluorescence decay $I(\tau)$ of EtCz (broken line). The dashed-dotted line represents $k_{CS}(\tau)$ of Cz-S-2CF$_3$ at 300 K calculated by Eq. (7).](https://repository.kulib.kyoto-u.ac.jp/)
and Cz-S-CN above 200 K and weak temperature dependence below 200 K. These results clearly show that CS in Cz-S-A was effectively promoted by the dielectric relaxation with \( \varepsilon_{\text{eff}} \) even below \( T_g \). The \( \varepsilon_{\text{eff}} \) of poly(butyl methacrylate) solids below \( T_g \) was as small as the optical dielectric constant. This difference results from the polarity and flexibility of the side chain. Orientational motions of the polar groups in poly(alkyl methacrylate) solids are highly restricted below \( T_g \) because the polar ester group is bound at the root of the main chain. On the other hand, polar cyano groups of CN-PUL are highly mobile even below \( T_g \), because they are bound at the end of the side chain. As a result, there exist dielectric relaxations of the polar group even in the CN-PUL glassy solid, which are fast enough to effectively promote the photoinduced CS.

The effective reorganization energy \( \lambda_{\text{eff}} \) of the CN-PUL glassy solid can be estimated by \( \varepsilon_{\text{eff}} \) (Fig. 8). The \( \lambda_{\text{eff}} \) also increased from 200 K and remained almost constant below 200 K. The former part above 200 K was attributed to the reorganization due to low-frequency orientational motions of cyano groups (\( \lambda_{\text{cyano}} \)) and the latter part below 200 K was attributed to that due to high-frequency vibrational or librational motions of the CN-PUL glassy solid (\( \lambda_{\text{high}} \)), respectively. Recent studies on solvation around probe molecules in solutions revealed the existence of fast motions in the time region shorter than the orientational motions. These fast motions are ascribed to inertial motions due to vibrational or librational motions of the solvent, which interact with the reactant on a time scale from ca. 10\(^4\) to 10\(^5\) fs and the relaxation time is temperature independent. Even in polymer glassy solids, similar solvation dynamics due to libration has been observed in the same time domain.

Thus, the CS reaction in Cz-S-A is driven by both the low- and high-frequency motions at temperatures above 200 K, and mainly by the high-frequency motion remaining at temperatures below 200 K. Providing that the fast motions are coupled with the CS reaction, nuclear-tunneling effects would be expected at low temperatures because of their large \( \hbar \omega_{\text{L}} \), which causes the weak temperature dependence of the CS rate. To discuss the contribution of the high-frequency motions to the CS rate at low temperatures, the CS rate of Cz-S-2CF\(_3\) was quantitatively evaluated by using a two-mode quantum-mechanical ET formula [Eq. (13)].

\[
k_{\text{CS}}(\hbar \omega_H, \hbar \omega_Q) = \frac{2 \pi}{h^2 \hbar \omega_H} V^2 \exp\left(-\frac{S_H}{2} + \frac{S_Q}{2} - \frac{S_{\text{Q}}}{2} + \frac{1}{2} \right) \times \sum_{m=-\infty}^{\infty} \left( \frac{\bar{n}_H + 1}{\bar{n}_H} \right)^{p(m)/2} \times I_{p(m)}\left(2 S_H \sqrt{\bar{n}_H (\bar{n}_H + 1)}\right) \times \left( \frac{\bar{n}_Q + 1}{\bar{n}_Q} \right)^{m/2} I_{m}\left(2 S_Q \sqrt{\bar{n}_Q (\bar{n}_Q + 1)}\right),
\]

where

\[
\bar{n}_i = \left( \frac{\exp \left( \frac{\hbar \omega_i}{k_B T} \right) - 1}{\exp \left( \frac{\hbar \omega_i}{k_B T} \right) + 1} \right)^{-1}, \quad S_i = \frac{\lambda_i}{\hbar \omega_i} \quad (i = H, Q),
\]

\[
p(m) = -\frac{\Delta G_{\text{CS}} + m \hbar \omega_Q}{\hbar \omega_H}.
\]

Here, \( p(m) \) is taken as an integer closest to the value of \( p(m) \). At temperatures below 200 K, there exist a high-frequency matrix mode \( (i=H) \) and an intramolecular vibrational mode of Cz-S-A \( (i=Q) \) because the low-frequency matrix mode is frozen. The matrix mode \( (i=H) \) is ascribed to fast motions with a high vibrational quantum \( \hbar \omega_H \) and the corresponding reorganization energy \( \lambda_H \) is equal to \( \lambda_{\text{high}} \) (see Fig. 8). The intramolecular vibration is the highest-frequency quantum mode \( (i=Q) \) with a vibrational quantum \( \hbar \omega_Q \), which contributes to the reorganization energy \( \lambda_Q \) of Cz-S-A. Using \( \varepsilon_{\text{eff}} \), \( -\Delta G_{\text{CS}} \) and \( \lambda_{\text{high}} \) were evaluated by Eq. (5) and as \( \lambda_{\text{eff}} \) \( =0.56 \text{ eV} \) at 100 K, respectively. Other parameters were assumed as follows: \( \lambda_Q = 0.56 \text{ eV}, \hbar \omega_Q = 1300 \text{ cm}^{-1}, \) and \( V = 3.5 \text{ cm}^{-1} \). Figure 9 shows the theoretical prediction with \( \hbar \omega_H = 250 \text{ cm}^{-1} \) as a high-frequency mode of the CN-PUL glassy solid. The temperature dependence of the CS rate:
rate below 200 K was well explained by $k_{CS}(\hbar \omega_{\text{CS}} = 250 \text{ cm}^{-1}, \hbar \omega_{\text{Q}} = 1300 \text{ cm}^{-1})$. With $\hbar \omega_{\text{Q}}$ smaller than 250 cm$^{-1}$, $k_{CS}(\hbar \omega_{\text{CS}}, \hbar \omega_{\text{Q}})$ would decrease monotonically below 200 K. This suggests that the weak temperature dependence of the CS rate at low temperatures originates in nuclear tunneling due to the high-frequency mode in the CN-PUL glassy solid. The high vibrational quantum $\hbar \omega_{\text{CS}} = 250 \text{ cm}^{-1}$ corresponds to a motion with a relaxation time of ca. 0.1 ps, which is probably ascribable to librations in polymer glasses. Thus, it was concluded that CS below 200 K is effectively coupled to the fast motions due to librations with $\hbar \omega_{\text{Q}} = 250 \text{ cm}^{-1}$ leading to weak temperature dependence of the CS rate at low temperatures by nuclear tunneling.

IV. SUMMARY

Photoinduced CS of Cz-S-A in the CN-PUL glassy solid was studied in a temperature range of 100–400 K. For Cz-S-2CF$_3$, $k_{CS}$ increased above 200 K while it was independent of temperature below 200 K. The increase in $k_{CS}$ corresponded to the side-chain relaxation of the CN-PUL glassy solid. The dielectric analysis of the side-chain relaxation with the Cole-Cole distribution function revealed a wide distribution in the dielectric relaxation time $\tau$ for the CN-PUL glassy solid. Thus, the temperature dependence of $k_{CS}$ was theoretically examined considering the wide distribution of $\tau$. As for the increase in $k_{CS}$ above 200 K, $k_{CS}$ was well explained by a thermally nonequilibrium ET formula $\langle k_{CS}(\tau) \rangle_{\text{eff}}$ with the effective dielectric constant $\varepsilon_{\text{eff}}$ and the effective dielectric distribution function $\phi(\tau)_{\text{eff}} = \phi(\tau)(\tau)$. The effective dielectric constant $\varepsilon_{\text{eff}}$ was evaluated with a fluorescent probe of Cz-TP cyclophane. The effective dielectric distribution function $\phi(\tau)_{\text{eff}}$ was an effective modification function of $\phi(\tau)$, which was weighted by the population of EtCz in the excited state $I(\tau)$.

As for the weak temperature dependence of $k_{CS}$ below 200 K, $k_{CS}$ was well explained by a two-mode quantum mechanical ET formula with a high-frequency matrix motion $\hbar \omega_{\text{CS}} = 250 \text{ cm}^{-1}$ and a high-frequency intramolecular motion $\hbar \omega_{\text{Q}} = 1300 \text{ cm}^{-1}$. Both motions are so rapid that ET can be treated under the thermal equilibrium condition. The high-frequency matrix motion was attributed to librational motions of the polar cyano groups in the CN-PUL glassy solid. The wide variations in temperature dependence of $k_{CS}$ originate from not only the large dielectric intensity owing to the polar cyano group but also the wide distribution in the dielectric relaxation time, which extended into a time domain as fast as the CS event.

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In the model calculations [Eqs. (10) and (12)], the averaged rates are evaluated by direct averaging of the individual rates, which is a reasonable way for calculating the averaged rates in inhomogeneous media. Therefore, it is true that the averaged experimental rate constant by using the following equation would be more reasonable: 

\[ k = \frac{1}{\tau} - \frac{1}{\tau_0}, \]

where \( n \) represents the number of different relaxation modes in CN-PUL, \( \Lambda_i \) is the relative amplitude of the fluorescence decay component \( i \) with lifetime \( \tau_i \), and \( \tau_0 \) is the fluorescence lifetime of the reference compound. We think, however, that this approach is unrealistic because we cannot decide the number of relaxation modes \( n \). Moreover, the number of exponential functions available for fitting experimental data is usually limited to two or three, although it should be larger number \( n \). From these reasons, we calculate the experimental CS rate by using the averaged fluorescence lifetime.

One of the three components is ascribable to an impurity Cz fluorescence that lacks the A moiety, which is a minor component less than 5%. Thus, \( k_{CS} \) was calculated by \( k_{CS} = 1/\tau_{i} - 1/\tau_{0} \), where \( \tau_{i} \) is the averaged lifetime of the two major components in the fluorescence of Cz-S-A. See Ref. 53.