Surface phonons of C(001)(2×1)-H  

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Surface phonons of the hydrogen-covered diamond surface C(001)(2×1)-H have been studied by electron energy loss spectroscopy (EELS) combined with semiempirical total-energy calculations. EELS resolves several vibrational features in the energy region of the bulk phonon bands in addition to C-H stretch modes. From calculated depth-resolved spectral densities, the vibrational features are found to originate not only from the adsorbed hydrogen and the carbon surface dimer atoms but also from deeper layers of the diamond crystal. On the other hand, a nearly localized character of the vibrations along the surface is found from the measurements of the energy dispersions.

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Diamond has attracted much attention from both experimentalists and theorists owing to its intrinsic properties such as a large band gap, high saturation carrier velocity, and high thermal conductivity. Recently, diamond films can be homoepitaxially grown by chemical vapor deposition (CVD) using a mixture of hydrocarbon and hydrogen. It is believed that hydrogen plays a crucial role in the deposition process. Therefore, a study of vibrational properties of hydrogenated diamond surfaces is of great importance since it can lead to a better understanding of the mechanisms governing growth.

A number of experimental, as well as theoretical, studies has focused on the vibrational properties of C(001)(2×1)-H. The experimental data, mainly obtained using electron energy loss spectroscopy (EELS), show complex vibrational structures ranging from 120 to 180 meV and a well-defined C-H stretch mode at 362–363 meV. Due to their resonance with bulk phonons, the vibrations in the region below 165 meV are not expected to be purely derived from the surface. The assignments of the vibrational losses and their respective displacement patterns, except for the C-H stretch modes, are not clarified yet.

In this study, we investigate vibrational modes of the C(001)(2×1)-H surface by EELS combined with semiempirical total-energy calculations. From the calculations of the depth-resolved phonon spectral densities it is found that the vibrational spectra of C(001)(2×1)-H contain significant contributions from deeper layers of the substrate. The phonon dispersions along the surface are determined from angle-dependent measurements.

The experiments were performed using an ultrahigh vacuum chamber equipped with a high resolution electron spectrometer for EELS (LK-5000, LK Technologies, Inc.), an electron optics for low-energy electron diffraction, and a quadrupole mass spectrometer for gas analysis. The base pressure of the chamber is less than 1×10^{-10} Torr. The B-doped diamond film is epitaxially grown on the (001) surface of synthesized single-crystalline diamond (1b type) using microwave plasma-assisted CVD. The thickness of the film is ~20 μm. The typical B concentration in the CVD diamond film is at the ppm level. The size of the sample was 4×4×0.3 mm³. The sample is mounted on a tantalum holder. The sample can be cooled to 90 K using a liquid nitrogen reservoir, and heated by electron bombardment from the back side. The sample temperature is measured using an alumel-chromel thermocouple attached to the sample holder.

For the EELS measurements, primary energies (E_p) of 1–20 eV, a typical energy resolution of ~3 meV, incidence angles (θ_i) of 60°–80° and emission angles (θ_e) of 25°–80° with respect to the surface normal are used. The phonon dispersion curves are determined along the ΓJ and ΓJ’ high-symmetry lines of the surface Brillouin zone (SBZ) shown in the inset of Fig. 1. Since 2×1 and 1×2 domains coexist on our sample surface, the dispersions along both directions are obtained by the measurements along either direction. We have, therefore, restricted our measurements to the ΓJ’ line. The length of the respective wave vector parallel to the surface is given by \( Q_i = 0.5123 \sqrt{E_p} (\sin \theta_i - \sin \theta_e) \) Å⁻¹. All EELS measurements are made at 90 K.

The calculations have been carried out employing the semiempirical total-energy approach (cf. Ref. 10). In the current calculations the convergence with respect to layer number has been improved. The structure optimization has been achieved with 50 C and two H layers (symmetrical slab) while the spectral densities were evaluated for 436-layer slabs (the above 52 plus 384 carbon layers included by bulk filling in the middle of the slab).

Figure 1 and the inset show a series of EELS spectra of C(001)(2×1)-H in the energy region of the bulk phonons (~165 meV) and the C-H stretch vibrations (~360 meV), respectively, as a function of \( Q_i \) at \( E_p = 2.8 \) eV (the lowest spectrum in Fig. 1), 14.0 eV (the second lowest), and 16.5 eV (the others). In the specular direction (\( Q_i = 0 \) Å⁻¹), we mainly observe three pronounced losses at 102, 149, and 156 meV. In addition, we find a small peak at 164 meV and a...
As assigned to the Rayleigh modes in the two directions. Below 65 meV, strongly dispersing features are observed and at other primary energies that this loss is an intrinsic peak since it is clearly observed that this loss is an intrinsic peak since it is clearly observed.

In Fig. 1 shows EELS spectra of the C-H stretch region obtained for $E_p = 16.5$ eV and the same $Q_i$ values as in Fig. 1. In the specular direction an intense peak is observed at 362 meV. With increasing $Q_i$ this peak attenuates and a new peak is observed at 360 meV at $Q_i = 0.62 \text{ Å}^{-1}$. One might attribute the 360-meV peak as originating from the dispersion of the 362-meV loss. Instead, we assign these two losses to different normal modes, i.e., to antisymmetric and symmetric stretching modes, for the following reasons: (1) The spectrum at $Q_i = 0.25 \text{ Å}^{-1}$ consists of two features, a main peak at 362 meV and a shoulder on the low-energy side. Also the main peak at 360 meV for $Q_i = 0.62 \text{ Å}^{-1}$ has an asymmetric shape with a tailing to higher energies. (2) On Si(001)(2×1)-H, which has a structure similar to C(001)(2×1)-H, the antisymmetric and symmetric stretching modes were clearly resolved by infrared spectroscopy at 259 and 260 meV, respectively. Previous calculations for C(001)(2×1)-H showed a negligible dispersion for the two C-H stretching modes (less than 1 meV). Thus we attribute the 360- and 362-meV losses to two different normal modes of the C-H stretch and tentatively assign them, at this point, to the antisymmetric and symmetric modes, respectively. We note that the assignment on the basis of our experimental results is not straightforward because the dipole selection rule may not be applicable on the diamond surface. We come back to this point further below.

In the latest EELS studies of the C(001)(2×1)-H surface, Kinsky et al. found a peak at 183 meV which is not observed in Fig. 1. We have used three samples prepared similarly. Only one of them exhibits an unreplicable loss at 183 meV accompanied by a higher-energy tail of the C-H stretch mode. We conjecture that the 183-meV loss is due to minority species on the surface. Indeed, a recent theoretical study predicts that the dihydride species on C(001) shows a CH₂ bending mode at 183 meV. Thus, we attribute the loss at 183 meV reported previously to the dihydride species possibly present at defect sites of the C(001)(2×1)-H surface.

Figure 2 shows calculated atom- and layer-resolved spectral densities at the $\bar{\Gamma}$ point in the energy range of 0–200 meV. They clearly show significant intermixing of the vibrational motions between the surface hydrogen and substrate carbon atoms due to their resonance with the bulk phonons. We find a considerable number of normal modes existing at the $\bar{\Gamma}$ point of the SBZ. Displacement patterns of the most salient of these modes are schematically depicted (not drawn to scale for clarity sake) in Fig. 3. The single-domain 2×1 surface belongs to the $C_{2v}$ point group. Thus, the surface phonon modes can be classified according to the respective irreducible representations, as indicated in Fig. 3. Since we measure spectra of double-domain C(001)(2×1)-H along the $\bar{\Gamma}J$ and $\bar{\Gamma}J^\prime$ directions at the same time, normal modes of
\(a_1\), \(b_1\), and \(b_2\) symmetries can be observed, whereas those of \(a_2\) symmetry are forbidden according to the selection rule for impact scattering\(^{14}\) and thus cannot be observed by EELS.

The loss peaks are assigned to the calculated normal modes. The \(\sim 70\) meV loss is assigned to \(b_1(1)\). The 100- and 102-meV losses are possibly assigned as \(b_1(2)\) and \(a_1(1)\), respectively. The 120-, 136-, 149-, and 156 meV losses are assigned to \(b_2(1)\), \(b_2(2)\), \(b_1(3)\), and \(b_2(3)\), respectively. As noted above, a previous study assigned the broad loss at \(\sim 152–155\) meV to C-H bending vibrations.\(^{3,4}\) We find, however, from the spectral densities that deeper layers of the substrate contribute significantly in this energy region (see Fig. 2). Consequently, the vibrations in this region are attributed to the substrate modes. The 167-meV mode is assigned to \(a_1(2)\), which is mainly due to the C-C stretch of the surface dimer. We note at this point that the 120-, 136-, 149-, and 164-meV losses can have significant contributions from the bulk phonons, namely, the transverse optical mode at \(W\) and \(X\) and the longitudinal optical mode at \(X\) and \(\Gamma\), respectively.\(^{13}\) Depending on the particular mode, the individual contributions from the first four layers are different, as shown in Fig. 2. For example, the \(a_1(2)\) mode, which is mainly due to the C-C stretch of the surface dimer, is localized on the first three layers. The \(a_1(1)\) mode is most completely localized in the H adlayer. It can be called a “scissors” mode. On the other hand, a contribution of even the fourth layer is apparent in Fig. 2 for the \(b_1(2)\) mode.

Figure 3 implies that the \(b_2(2)\) mode is characterized by a “wagging” motion of the surface dimer. This characterization is not straightforward, however, due to the strong coupling of this mode to the substrate.

To corroborate the general picture of the vibrational modes at the C(001)(2×1)-H surface developed so far, we have investigated, as well, the isotope-substituted C(001)(2×1)-D surface by EELS. For this system we have observed the \(a_1(1)\), \(b_2(1)\), and \(b_1(3)\) modes at 85, 90, and 110 meV, respectively. It is noted that, even if the mode is the same, the displacement patterns for (2×1)-H and (2×1)-D are not exactly the same. These isotope shifts indicate that substantial H (D) motions are involved in these modes which is in agreement with the calculations. We note that the remaining modes for C(001)(2×1)-D are observed nearly at the same energies as those for C(001)(2×1)-H indicating that the effective reduced masses entering these modes are dominated by the C atoms.

Figure 4 shows the wave-vector dependence of the energy losses taken with \(E_p=1–20\) eV, \(\theta_p=60^\circ–80^\circ\), and \(\theta_e=25^\circ–80^\circ\). The calculated phonon bands are shown for comparison by solid curves. For clarity sake, neither the symmetry type nor localization properties of the involved modes are explicitly indicated. Since the dispersion of the losses along both the \(\Gamma\bar{J}\) and \(\Gamma\bar{J'}\) directions have to be superimposed under the present experimental conditions, the same experimental results are plotted along both directions. Overall, we find a very good agreement between theory and
and 167 meV are very close to calculated bands throughout the entire SBZ (see Fig. 4). It should be noted at this point that, for the sake of clarity, the calculated phonon bands are plotted only for the 2×1 SBZ. Thus, they are no superpositions from the 2×1 and the 1×2 SBZ’s, as opposed to the experimental data. From Fig. 4 it is obvious, however, that backfolding the calculated bands from J to Γ onto the line from ΓJ'/2 to J' brings the theoretical results into coincidence with the measured Rayleigh modes along both the JΓ and ΓJ' lines. All measured modes, except for the Rayleigh modes, show only small dispersions of less than 5 meV, indicating the localized character of the vibrations along the surface. Some of the calculated bands are not observed in EELS. This is thought to be due to their symmetry or to their weak spectral density. Further up in energy, there occur two C-H stretch modes. One is symmetric with respect to the two C-H bonds (see the side view in the top panel of Fig. 2) while the other is asymmetric. The top panel of Fig. 4 shows the two measured stretch-mode bands around 360 and 362 meV. The calculated stretch-mode bands result to be a little bit lower in energy. We obtain the symmetric stretch mode at 356 and the asymmetric stretch mode at 358 meV from our calculations. The size of the splitting of these two dispersionless bands is practically the same in both experiment and theory, but the symmetry assignment is opposite. The theory yields the symmetric mode to be lower in energy, as compared to the asymmetric mode. This can be rationalized by the fact that the asymmetric stretch mode involves slightly stronger bond-length changes in the H-C-H unit, as compared to the symmetric stretch mode.

In summary, we have investigated the vibrational modes for the C(001)(2×1)-H surface by means of EELS combined with semiempirical total-energy calculations. We observe losses at ~70, 100, 102, 120, 136, 149, 156, 164, 167, 360, and 362 meV near the Γ point. From the calculations of the depth-resolved spectral densities we find that the second or third layers of the diamond substrate significantly contribute to the vibrational spectra of C(001)(2×1)-H. The assignments are made on the basis of our theoretical results. The phonon dispersions along the surface have been determined from angle-dependent measurements, where we find a nearly localized character of the vibrations along the surface.

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