## **Synthesis, Thermal Stability, Structural Features and Electromagnetic Properties of**  $\mathbf{Bi}_{2+x} \mathbf{Sr}_{2-x} \mathbf{CuO}_{6+\delta}$  **(0≤***x***≤0.4)**

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The thermal stability and structural modulation were studied systematically in a wide range of  $0 \le x \le 0.4$  for the 2201 phase in the Bi-Sr-Cu-O system,  $Bi_{2+x}Sr_{2-x}CuO_{6+\delta}$  and it was found that these properties varied remarkably at  $x \approx 0.1$ . Compositions  $0 \le x < 0.1$  remained stable only in a narrow low *T*- high  $Po_2$  region and their modulation period changed stepwisely, not continuously, and reversibly between 4.9*b* (oxidized) and 5.5*b* (reduced) when the oxygen content was changed only by 0.65%. In relation to this we propose for  $0 \le x < 0.1$  specifically that the change in oxygen content induces the exchange of small amounts of Bi and Sr ions between the "BiO" and "SrO" sheets. The superconductivity of the cation-stoichiometric composition  $(x = 0)$  was also studied as a function of oxygen content.

Keywords: Phase diagram /  $Bi_{2+x}Sr_{2-x}CuO_{6+\delta}$  / Substitution / Modulation / Superconductivity

The "2201" phase in the  $Bi_2O_3$ -SrO-CuO system is known to adapt itself to various Bi:Sr:Cu ratios. Our previous phase diagramic study done at 840°C in the air [1] showed that the monophasic range was  $0.1 < x < 0.6$  and  $0 < y < x$ 2 for  $\text{Bi}_{2+x} \text{Sr}_{2-x} \text{Cu}_{1+y} \text{O}_z$  and that for  $0 \le x \le 0.1$  three kinds of phases including the Bi-poorest end of the above mentioned solid solution,  $Bi_{17}Sr_{16}Cu_{7}O_{z}$ , and  $Sr_{14}Cu_{24}O_{41}$  coexisted. More recently it has been reported that the solubility range is extended toward  $x = 0$  at high oxygen pressures. Kato et al. successfully obtained a cation-stoichiometric sample with  $x = 0$  at 840<sup>°</sup>C and Po<sub>2</sub> = 30 atm, which was an over-doped metal that became a superconductor when annealed in  $N_2$  [2].

 In this report we will shed a new light on the relation among Bi content, oxygen content, thermal stability, and structural features of the 2201 phase by comparing behaviors of monophasic samples with  $0 \le x \le 0.4$  systematically which were all prepared under conventional conditions like  $Po_2 = 1$  atm and 800°C.

All the samples were prepared by an ordinary ceramic

method from  $Bi_2O_3$ ,  $SrCO_3$ , and CuO, each with a purity of 99.9%. Appropriate mixtures of these starting materials were pressed into pellets and heated at 600°C-840°C for 20h-120h in total with intermittent grinding, mixing and pelletizing processes. Three different atmospheres including an oxygen stream of 1 atm, the air, and an Ar stream of 1 atm were used. Certain samples were postannealed in the Ar atmosphere at different temperatures between 200-700˚C for 12-240h depending upon the temperature.

Cation-stoichiometric  $Bi_2Sr_2CuO_{6+\delta}$  was successfully obtained by firing the starting mixture in flowing  $O_2$  first at 720°C and finally at 800°C. The tetragonal cell parameters of  $a = 5.361$  Å,  $c = 24.65$  Å calculated from the Xray diffraction peaks were almost identical to those (*a* = 5.37 Å,  $c = 24.65$  Å) of Kato et al.'s sample with  $\delta = 0.2$ which was synthesized under  $Po_2 = 30$  atm and post-annealed in flowing  $N_2$ . We note here that small amounts of  $Bi_{17}Sr_{16}Cu_{7}O_{z}$  and others were detected after a further treatment at 820°C, showing the stability of  $\text{Bi}_{2}\text{Sr}_{2}\text{CuO}_{_{6+\delta}}$ 

## SOLID STATE CHEMISTRY — Quantum Spin Fluids—

## *Scope of research*

Quantum oxide systems such as high-T<sub>c</sub> superconducting cuprates,  $La_2xSr_xCuO_4$  and a spin-ladder,  $(Sr,Ca)_{14}Cu_{24}O_{41}$  are *synthesized in the form of single crystals using traveling-solvent-floating-zone and laser abrasion techniques. Detailed equilibrium phase diagram of Bi cuprate systems is investigated. Main subjects and techniques are: mechanism of high-T<sub>c</sub> superconductivity: origin of quantum phase separation in strongly correlated electron systems: spin exitations in quantum spin systems: interplay between spin and charge flow in doped spin systems: neutronscattering by using triple-axis as well as time-of-flight techniques.*



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**Figure 1.** Temperature-oxygen pressure diagram with a border line below which the cation-stoichiometric composition remains stable. The closed squares are the present data and the open square is from ref. 2.

 We tested the synthesis at a lower oxygen pressure as follows. Shown in Fig. 1 is a  $Po_2$ -T diagram with a border line below which the cation-stoichiometric composition remains stable.

Monophasic samples with higher Bi contents of  $0 < x \leq$ 0.4 were also prepared at both  $(T^{\circ}C, Po_{2}/atm) = (800, 1)$ and (730, 0.2). The composition dependences of the lattice parameters, *a* and *c*, are plotted in Fig. 3. As Bi content increases from  $x = 0$  to 0.4, the *c* parameter decreased by 0.8%, while *a* increased by 0.6%. In further detail, the *c* parameter showed a small jump at  $x \approx 0.1$  and, at the same time, the slope,  $da/dx$ , became sharper for  $0.1 \leq x$ . This anomaly concerning the lattice parameters is one of the several features that separate the composition range of  $0 \le x \le 0.4$  into two with a border line at  $x \approx 0.1$ .

In parallel to this, the thermal stability examined at  $P_{\text{O}_2}$  $= 0.2$  atm and 1 atm also showed a gap at  $x \approx 0.1$ . Saying typically, the decomposition temperature was as high as  $\approx$ 880°C for  $x = 0.125$  in the air but it dropped to ≈780°C for  $x = 0.10$  in the same atmosphere.

 It is well-known that the 2201 structure is incommensurately modulated with its wave vector, *q*, lying in the *b*\*-*c*\* plane. We found the same type of modulation in all the present samples by means of XRD and TEM. The coefficients  $b_m$  and  $c_m$  of the vector  $q = b_m b^* +$ *c*<sub>m</sub><sup>c</sup>\* showed an interesting stepwise composition dependence again at  $x \approx 0.1$ . These coefficients were evaluated from the XRD data using the following equation

 $1/d_{hklm}^2 = h^2/a^2 + (k+m b_m)^2/b^2 + (l+m c_m)^2/c^2$ , (1) where  $d_{hklm}$  stands for the  $d$  value of a superlattice peak (*hkl*,  $\pm m$ ). We obtained a set of parameters (*a=b* /Å, *c* /Å,  $b_m$ , $c_m$ ) = (5.361, 24.65, 0.205, 0.455) from the XRD pattern for the  $x = 0$  sample prepared at 800°C and  $Po_2 = 1$ atm.

 Through a cyclic treatment of the former sample at 730°C in the air and at 800°C in flowing  $O_2$  we noticed that the change was quite reversible. We further noticed that  $b_m$ and  $c_m$  were changed stepwisely, not continuously, from  $(b_m, c_m) = (0.205, 0.455)$  to (0.185, 0.288) by reducing treatments as can be seen most typically for the sample annealed at 400°C in Ar (see Fig.3). These two types are mixed in samples annealed under intermediate conditions



**Figure 2.** Composition dependence of the subcell lattice parameters. The triangles are for the samples prepared at 800°C in  $O_2$ , and the circles are for those prepared at 730°C in the air.



**Figure 3.** Partial enlargements of the XRD patterns of  $\mathrm{Bi}_2\mathrm{Sr}_2\mathrm{CuO}_{_{6+\delta}}$  as-prepared in  $\mathrm{O}_2$  at 800°C (a), post-annealed in Ar at  $200^{\circ}$ C (b),  $400^{\circ}$ C (c), and  $600^{\circ}$ C (d).

like 730°C in the air. There seems no doubt that a slight change in oxygen content switches the modulation mode from one to the other, without changing the *a* and *c* parameters remarkably.

 We conducted TEM observations on the two typical samples with  $x = 0$ , one as-prepared in  $O_2$  and the other annealed in Ar at 600°C. The modulation wavelength varied from  $\lambda = 4.9b$  for the as-prepared sample to  $\lambda =$ 5.5*b* for the annealed one, which are consistent with the XRD results of  $\lambda = 4.88b$  ( = *b*/0.205) and 5.41*b* ( = *b*/ 0.185), respectively.

 From the resistance and magnetization measurements it has heen revealed that it is only the portion with  $(b_m, c_m)$  $= (0.185, 0.288)$  that is superconducting but the portion with  $(b_m, c_m) = (0.205, 0.455)$  is an over-doped metal.

## **References**

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