

Pore structure of aluminas derived from the alkyl derivatives of boehmite

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1 **Abstract**

2 The alkyl derivatives of boehmite (alkoxyalumoxanes; $\text{AlO}(\text{OH})_{1-x}(\text{OR})_x$) were
3 synthesized by the reaction of aluminum triisopropoxide in straight-chain primary
4 alcohols at 300 °C for 2 h in an autoclave. In the present work, pore structures of
5 aluminas obtained by calcination of the alkyl derivative of boehmite were examined.
6 The alumina obtained from the ethyl derivative of boehmite had a broad pore-size
7 distribution, while the pore-size of the alumina obtained from the dodecyl derivative of
8 boehmite distributed in a narrow range in the mesopore region. The mode pore
9 diameter of the latter alumina increased with the increase in calcination temperature
10 (as-syn., 39 Å; 600 °C, 54 Å; 800 °C, 58 Å; 1000 °C, 68 Å), but narrow pore-size
11 distribution was maintained even after calcination at high temperatures.

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18 *Keywords:* Alumina; Pore structure; Solvothermal reaction

19 **1. Introduction**

20 For the supports of industrial catalysts, alumina is most widely used because it is
21 inexpensive and reasonably stable, and is provided with a wide range of surface areas
22 and porosities suitable for a variety of catalyst application [1]. Preparation of alumina
23 (or alumina-based) supports with the controlled pore structure is still active area: Thus,
24 many papers have been reported [2–8].

25 Boehmite is one of the modifications of aluminum oxide hydroxide, AlOOH , and it
26 can be easily prepared by hydrothermal treatment of aluminum hydroxide [9].
27 Microcrystalline boehmite is called “pseudoboehmite” and is used as a precursor of
28 aluminas. Although boehmite has a layer structure, intercalation of guest molecules
29 into the boehmite layers has never been reported, presumably because of strong
30 hydrogen bonding between the layers. On the other hand, during the course of our
31 long-term study on controlling the pore texture of alumina for use as catalyst supports,
32 we found that the thermal treatment (glycothermal and alcohothermal treatments) of
33 aluminum triisopropoxide (AIP) in organic solvents (glycols and alcohols) yielded
34 novel derivatives of boehmite, in which the alkyl (or hydroxyalkyl) groups were
35 incorporated into the boehmite layers through the covalent bondings [10–13].

36 In the present work, the pore structures of aluminas derived from the alkyl

37 derivatives of boehmite were examined.

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39 **2. Experimental**

40 *2.1. Synthesis of the alkyl derivatives of boehmite*

41 In a Pyrex test tube serving as an autoclave liner, 130 ml of a straight-chain primary
42 alcohol (ethanol, EtOH; 1-butanol, BuOH; 1-pentanol, PeOH; 1-hexanol, HeOH;
43 1-octanol, OcOH; 1-decanol, DeOH; 1-dodecanol, DDOH) and 12.5 g of aluminum
44 triisopropoxide (AIP) were placed, and the test tube was then placed in a 300 ml
45 autoclave. In the gap between the autoclave wall and the test tube was placed an
46 additional 30 ml of the alcohol. The autoclave was thoroughly purged with nitrogen,
47 heated to 300 °C at a rate of 2.3 °C /min, and held at that temperature for 2 h. After the
48 mixture was cooled to room temperature, the resulting precipitate was washed by
49 repeated cycles of agitation with methanol, centrifuging, and decantation, and then
50 air-dried. The obtained product was calcined at the various temperatures by heating at
51 a rate of 10 °C /min and holding at that temperature for 30 min in a furnace in static air.

52 These products will be designated by “A” followed by the abbreviation for the
53 medium used in the alcohothermal treatment and calcination temperature in degree

54 Celsius in parentheses. The original samples will be specified by a term, “as-syn”, in
55 parentheses.

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57 *2.2. Characterization.*

58 Powder X-ray diffraction (XRD) was measured on a Shimadzu XD-D1
59 diffractometer using CuK α radiation and a carbon monochromator. The nitrogen
60 adsorption isotherms were measured at liquid-nitrogen temperature by using a
61 volumetric gas-sorption system, Quantachrome Autosorb-1. The alumina samples were
62 previously outgassed at 300 °C for 30 min. Surface areas were calculated by applying
63 the BET method to the adsorption data, taking the average area occupied by a nitrogen
64 molecule as 0.162 nm². Pore size distributions were calculated from the desorption
65 branch of the nitrogen adsorption isotherm by the BJH method. Infrared spectra were
66 obtained on JASCO FT/IR-470 plus spectrometer using the usual KBr-pellet technique
67 with 128 integration times. Simultaneous thermogravimetric (TG) and differential
68 thermal analyses (DTA) were performed on a Shimadzu DTG-50 analyzer: a weighed
69 amount (ca. 20 mg) of the sample was placed in the analyzer, and then heated at the
70 rate of 10 °C/min. Morphologies of the products were observed with a scanning
71 electron microscope (SEM), Hitachi S-2500X.

72 **3. Results and discussion**

73 *3.1. Alkyl derivatives of boehmite obtained by solvothermal method.*

74 The XRD patterns of the products are shown in Fig. 1. For comparison, the XRD
75 pattern of pseudoboehmite is also given in the figure. The XRD peaks at the high angle
76 side ($2\theta = 50^\circ$ and 65°) correspond to the lattice parameters, a and c , of
77 pseudoboehmite (200 and 002 planes), and the obtained products could be indexed on
78 the basis of the boehmite structure [12,13]. The 020 plane of products shifted toward
79 the lower-angle side with the increase in the carbon number of the alcohol used as the
80 solvent (Fig. 1), suggesting that the alkyl groups derived from the solvent alcohol are
81 incorporated between the boehmite layers. In the XRD patterns of ADeOH(as-syn) and
82 ADDOH(as-syn), the peak due to 020 plane was not clearly shown. The alkyl chains of
83 the decyl and dodecyl groups are so long, that the diffraction peaks for the 020 plane
84 (basal plane) of ADeOH and ADDOH appeared at $2\theta < 3^\circ$. However, they exhibited
85 the 040 and 060 diffraction peaks (these peaks are 2nd and 3rd order diffraction peaks
86 of the basal plane), and therefore the position of the 020 diffraction peak can be
87 calculated precisely from these peaks, which clearly shows the low-angle shift of the
88 020 diffraction peak. To clarify these points, the 040 and 060 diffraction peaks are
89 indexed in Fig. 1. Linear increase of basal spacing with the increase of carbon number

90 of solvents was observed (Supplementary materials, Fig. 1S), which is an incontestable
91 evidence for the incorporation of the alkyl groups derived from the solvent alcohols.

92 In the IR spectra of the products shown in Fig. 2, bands characteristic of the
93 boehmite layers are seen at around 615 and 480 cm^{-1} [14–16], suggesting that the
94 products had the layer structure of boehmite. Bands due to the incorporated organic
95 moieties were also noted at 3000–2850 cm^{-1} (ν_{CH}).

96 Figure 3 shows the results for thermal analyses of the products. A large weight
97 decrease is seen around 100 and 400 $^{\circ}\text{C}$ for all of the samples. In the DTA profiles of
98 the products, one endothermic and two exothermic processes took place at around 100,
99 300 and 400 $^{\circ}\text{C}$, respectively. The first process is attributed to desorption of
100 physisorbed water/methanol and the second and third processes are caused by
101 combustion of the alkyl moieties incorporated between the boehmite layers. Collapse
102 of the boehmite layers yielding amorphous alumina seems to take place simultaneously
103 with the last process because of large exothermic effects. Because the molecular
104 weight of ethanol is small as compared to the other alcohols used in this study, the
105 intensity of the exothermic peak as well as the total weight decrease is the smallest
106 among the samples. The exothermic peak shifted slightly toward lower temperature
107 when a long chain alcohol was used (AEtOH(as-syn); 425 $^{\circ}\text{C}$, AHeOH(as-syn); 407 $^{\circ}\text{C}$

108 ADDOH(as-syn); 373 °C). This result may be attributed to the fact that the octane
109 number of an alcohol having a shorter alkyl chain is higher than that of alcohol having
110 a longer alkyl chain [17]. The results of thermal analyses of the products obtained by
111 the reaction of AIP in primary alcohols are summarized in Table 1. All these results are
112 consistent with the fact that the alkyl moieties are incorporated between the boehmite
113 layers.

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115 *3.2. Morphological aspects.*

116 The scanning electron micrographs of the products are shown in Fig. 4. Particles of
117 the product obtained in ethanol (AEtOH(as-syn)) had a rod shape. On the other hand,
118 particles of APeOH(as-syn) had irregular shapes but rod shape particles were also seen
119 together with the irregularly-shaped particles. AHeOH(as-syn) and ADDOH(as-syn)
120 were composed of large aggregate particles having irregular shapes.

121

122 *3.3. Calcined products.*

123 Phase transformation of the alkyl derivatives of boehmite was investigated and Fig.
124 5 shows the transformation of AHeOH as the representative results. AHeOH
125 maintained the boehmite structure at 300 °C and converted to amorphous alumina at

126 400 °C (Fig. 5). The γ -phase appeared by calcination in air at 800 °C for 30 min. The
127 sample calcined at 1000 °C was composed of θ - and α -phases. The single-phase of
128 α -phase was obtained by calcination at 1200 °C.

129 All of the other alkyl derivatives of boehmite were converted into γ -alumina
130 through an amorphous phase by calcination at 1000 °C in air, and the XRD patterns
131 were essentially identical irrespective of the solvents used.

132 Nitrogen adsorption isotherms, and pore size distribution curves of some the
133 products are shown in Figs. 6–8 and t -plots of all the products examined in this paper
134 are given in Fig. 9.

135 Although the aluminas derived from the ethyl derivative of boehmite exhibited rather
136 complex N_2 -adsorption isotherms (Fig. 6a), the t -plots (Fig. 9a) derived from the
137 isotherms can be divided into 5 segments. At low t (statistical thickness; 0–3 Å) region
138 (corresponding to low partial pressure of nitrogen), the plot seems to have a rather
139 steep slope going through the origin, although only a few data points are available. In
140 the second segment (3–7 Å), the slope decreased. The slope increased in the third
141 segment (7–12 Å) and decreased again in the fourth segment (12–18 Å) with final
142 increase at higher t region (>18 Å; fifth segment). The decrease of the slope from the
143 first to second segment is a typical phenomenon for microporous materials and is

144 explained by micropore filling. It was reported that calcination of well-crystallized
145 boehmite gives alumina having micropores with a slit-shape [18,19]. The crystallite
146 sizes of the present products were relatively large, and much larger than
147 pseudoboehmite usually obtained in aqueous systems. Therefore, the boehmite layer
148 structure seems to give the slit-shaped micropores on calcination. Hysteresis loop
149 observed in the isotherms (Fig. 6a) supports the presence of slit-shaped pores in the
150 AEtOH(600).

151 The increase in the slope at the middle partial pressure region (third segment) can be
152 explained by capillary condensation of adsorbate molecules into mesopores. Gradual
153 decrease in the slope in the third and fourth segments suggests that the pore size
154 distributed widely, which is verified by the pore-size distribution curve calculated by
155 the BJH method (Fig. 6b). These pores seem to be formed in the rod-shaped
156 pseudomorphous particles because of the difference between the true densities of
157 γ -alumina and the ethyl derivative of boehmite. The large slope in the fifth segment
158 indicates the presence of macropores, which can be also recognized from the pore-size
159 distribution curve shown in Fig. 6b. Since the pore size is in the order of the size of
160 rod-shaped particles, these pores are formed between pseudomorphous rod-shaped
161 particles.

162 Microporous structure of AEtOH(as-syn) is due to the elimination of a part of the
163 alkyl groups during the drying stage prior to the N₂-adsorption measurement.
164 Calcination of the product developed micro- and meso-pores because of elimination of
165 ethyl groups and collapse of the boehmite layer structure. Significant increase in
166 mesopore volume is apparent. Further increase in the calcination temperature caused a
167 significant decrease in micropore volume while mesopore volume was slightly
168 decreased. This result suggests that the primary particles of alumina separated by
169 micropores are easily sintered by heat treatment. On the other hand, macropores
170 formed between the rod-shaped particles was not affected by calcination. This can be
171 recognized by essentially identical slopes in the fifth segment and also by the pore-size
172 distribution curves.

173 The aluminas derived from ABuOH showed *t*-plots (Fig. 9b) different from those
174 obtained from AEtOH. The decrease of the slope from the first to second segments was
175 not seen, indicating that the micropores were not present in the samples. The increase
176 in the slope from the second to third segment was significant and third and fourth
177 segments are clearly distinguished, indicating that the pore size distributed in a narrow
178 range, which is verified by the pore-size distribution curve (Fig. 7b). Macropores were
179 not recognized. Hysteresis loop of these products suggests that tubular pores with

180 narrow constriction were formed (Fig. 7a).

181 Nitrogen adsorption isotherms of AOcOH exhibited a similar tendency with ABuOH.
182 However, an abrupt increase in the slope at high t region (fifth segment) was observed
183 in AOcOH (Fig. 9e), indicating that macropores were present in these samples.

184 As for APeOH(600) and AHeOH(600), micropores were not recognized and the first
185 and second segments are merged into one line going through the origin (Figs. 9c and
186 9d). This result can be explained by the boehmite layers separated by long alkyl groups.
187 The fourth and fifth segments are also merged. Usually the slope at high t region (fourth
188 segment) corresponds to the outer surface area after mesopores are filled by adsorbate
189 molecules by capillary condensation. However, this is not the case because the slope at
190 this region is much larger than the slope at low t region (the first segment;
191 corresponding to the total surface area). Therefore, pores are formed between the
192 irregularly-shaped particles (Fig. 4c) and the size of these pores distributes widely
193 from meso- to macro- pore region.

194 The most significant difference between the t plots of AHeOH(1000) and
195 ADDOH(1000)A is found in the slope at higher t region (Fig. 9f). Since the slope for
196 ADDOH(1000) is smaller than the slope at low t region, the former corresponds to the
197 outer surface area of the irregularly-shaped particles shown in Fig. 5d. The particles

198 size was so large that the space between these particles is not recognized as pores by
199 the nitrogen adsorption method.

200 The pore sizes of the aluminas derived from ADDOH(as-syn) can be assessed either
201 by closure points of the hysteresis in the isotherms (Fig. 8a), by the regions of the third
202 segments in the t -plots (Fig. 9f) or by the pore-size distribution curves (Fig. 8b). All the
203 data suggest that the pore size increased with the increase in the calcination
204 temperature. However, the increase in the pore size by the increase in the calcination
205 temperature is much smaller than that for the ordinary aluminas derived from
206 pseudoboehmite and crystalline aluminum hydroxides [20]. This indicates robustness
207 of the pore structure of the alumina derived from the dodecyl derivative of boehmite,
208 which is attributed to the well-developed layer structure of the precursor due to the
209 strong interaction between the alkyl groups. Table 2 summarized the results for the
210 physical properties of the products.

211

212 **4. Conclusions**

213 The alumina derived from the ethyl derivative of boehmite had a broad pore-size
214 distribution, while the pore-size of the aluminas obtained from the alkyl derivatives of
215 boehmite with long alkyl chains distributed in a narrow range in the mesopore region.

216 The mode pore diameter of the latter aluminas increased with an increase in calcination
217 temperature (as-syn., 39 Å; 600 °C, 54 Å; 800 °C, 58 Å; 1000 °C, 68 Å; for ADDOH),
218 but narrow pore-size distribution was maintained even after calcination at high
219 temperatures. These results indicate that van der Waals interaction between the alkyl
220 chains facilitated the formation of the boehmite layers having a fewer number of
221 defects, and that the collapse of the well-developed boehmite layers gave aluminas
222 with narrow pore-size distributions.

223

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259 **Figure captions**

260 Fig. 1. XRD patterns of pseudoboehmite (h) and the products (a–g) obtained by
261 reaction of AIP in alcohols: a, AEtOH(as-syn); b, ABuOH(as-syn); c, APeOH(as-syn);
262 d, AHeOH(as-syn); e, AOcOH(as-syn); f, ADeOH(as-syn); g, ADDOH(as-syn).

263 Fig. 2. IR spectra of the products obtained by the reaction of AIP in alcohols: a,
264 AEtOH(as-syn); b, AHeOH(as-syn); c, ADDOH(as-syn).

265 Fig. 3. Thermal analyses of the products obtained by the reaction of AIP in the alcohols
266 specified in the figure, in a 40 ml/min flow of dried air at the heating rate of 10 °C
267 /min: a, TG; b, DTA.

268 Fig. 4. Scanning electron micrographs of the products: a, AEtOH(as-syn); b,
269 APeOH(as-syn); c, AHeOH(as-syn); d, ADDOH(as-syn).

270 Fig. 5. XRD patterns of aluminas obtained by calcination of the hexyl derivative of
271 boehmite at various temperature in air: a, 300 °C; b, 400 °C; c, 500 °C; d, 600 °C; e,
272 800 °C; f, 1000 °C; g, 1100 °C; h, 1200 °C.

273 Fig. 6. Nitrogen adsorption isotherms (a) and *t*-plots (b) of the aluminas obtained by
274 calcination of the ethyl derivative of boehmite at various temperatures.

275 Fig. 7. Nitrogen adsorption isotherms (a) and *t*-plots (b) of the aluminas obtained by
276 calcination of the butyl derivative of boehmite at various temperatures.

277 Fig. 8. Nitrogen adsorption isotherms (a) and *t*-plots (b) of the aluminas obtained by
278 calcination of the dodecyl derivative of boehmite at various temperatures.

279 Fig. 9. *t*-Plots of the aluminas obtained by calcination of the alkyl derivatives of
280 boehmite at various temperatures: (a) ethyl derivative boehmite; (b) butyl derivative
281 boehmite; (c) pentyl derivative boehmite; (d) hexyl derivative boehmite; (e) octyl
282 derivative boehmite; (f) dodecyl derivative boehmite.

Table 1

Summary of thermal analysis of the alkyl derivatives of boehmite obtained by solvothermal method.

Sample	Weight ratio (BD ^a /Al ₂ O ₃)	Ignition temperature (°C)	Molecular fomular ^b
AEtOH	1.25	425	AlO(OH) _{0.86} (OEt) _{0.14}
ABuOH	1.25	402	AlO(OH) _{0.89} (OBu) _{0.11}
APeOH	1.51	417	AlO(OH) _{0.76} (OPe) _{0.24}
AHeOH	1.36	407	AlO(OH) _{0.89} (OHe) _{0.11}
AOcOH	1.51	408	AlO(OH) _{0.85} (OOc) _{0.15}
ADeOH	1.41	400	AlO(OH) _{0.91} (ODe) _{0.09}
ADDOH	1.66	373	AlO(OH) _{0.85} (ODD) _{0.15}

a: Alkyl derivatives of boehmite.

b: Empirical formulas, AlO(OH)_{1-x}(OR)_x were calculated from ignition losses determined by TG analysis for the products; Et, CH₃CH₂-; Bu, CH₃(CH₂)₃-; Pe, CH₃(CH₂)₄-; He, CH₃(CH₂)₅-; Oc, CH₃(CH₂)₇-; De, CH₃(CH₂)₉-; DD, CH₃(CH₂)₁₁-

Table 2

Physical properties of the alumina obtained by calcination of alkyl derivatives of boehmite at various temperatures.

Sample	Calcination Temp. (°C)	Surface area (m ² /g)	Pore volume (cc/g)	mode pore size (Å)
AEtOH	as syn.	389	0.74	74
	600	346	1.02	68
	800	167	0.64	63
	1000	117	0.64	80
ABuOH	as syn.	543	0.63	37
	400	414	0.73	50
	600	390	0.73	51
	1000	112	0.37	58
APeOH	600	364	1.0	50
	1000	105	0.56	110
AHeOH	as syn.	389	0.47	90
	600	392	0.69	68
	1000	119	0.51	154
AHOcOH	400	530	1.1	37
	600	513	1.27	50
	1000	136	0.65	62
ADeOH	as syn	444	0.71	37
	400	592	1.3	37
	600	469	1.2	42
	1000	157	0.76	62
ADDOH	as syn.	392	0.54	39
	600	338	0.69	54
	800	258	0.59	58
	1000	111	0.32	68

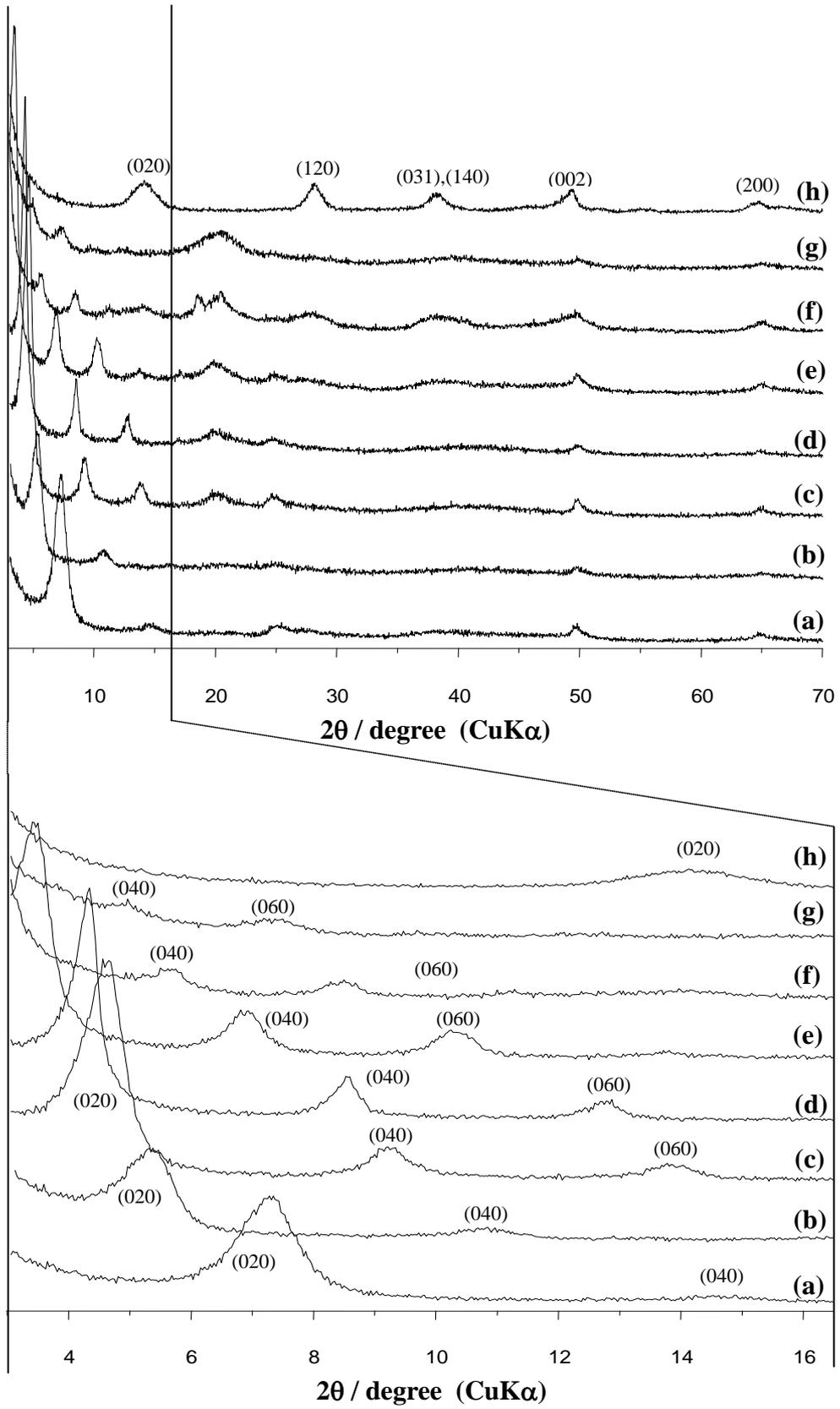


Fig. 1. Kim, et al.

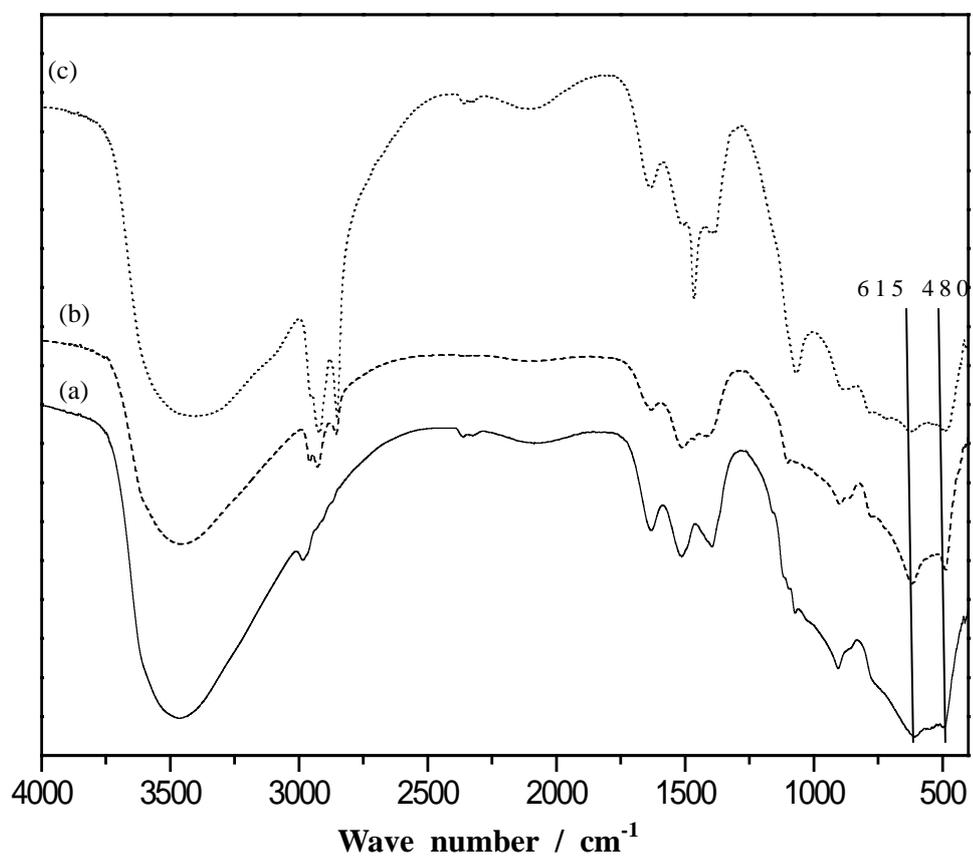


Fig. 2. Kim, et al.

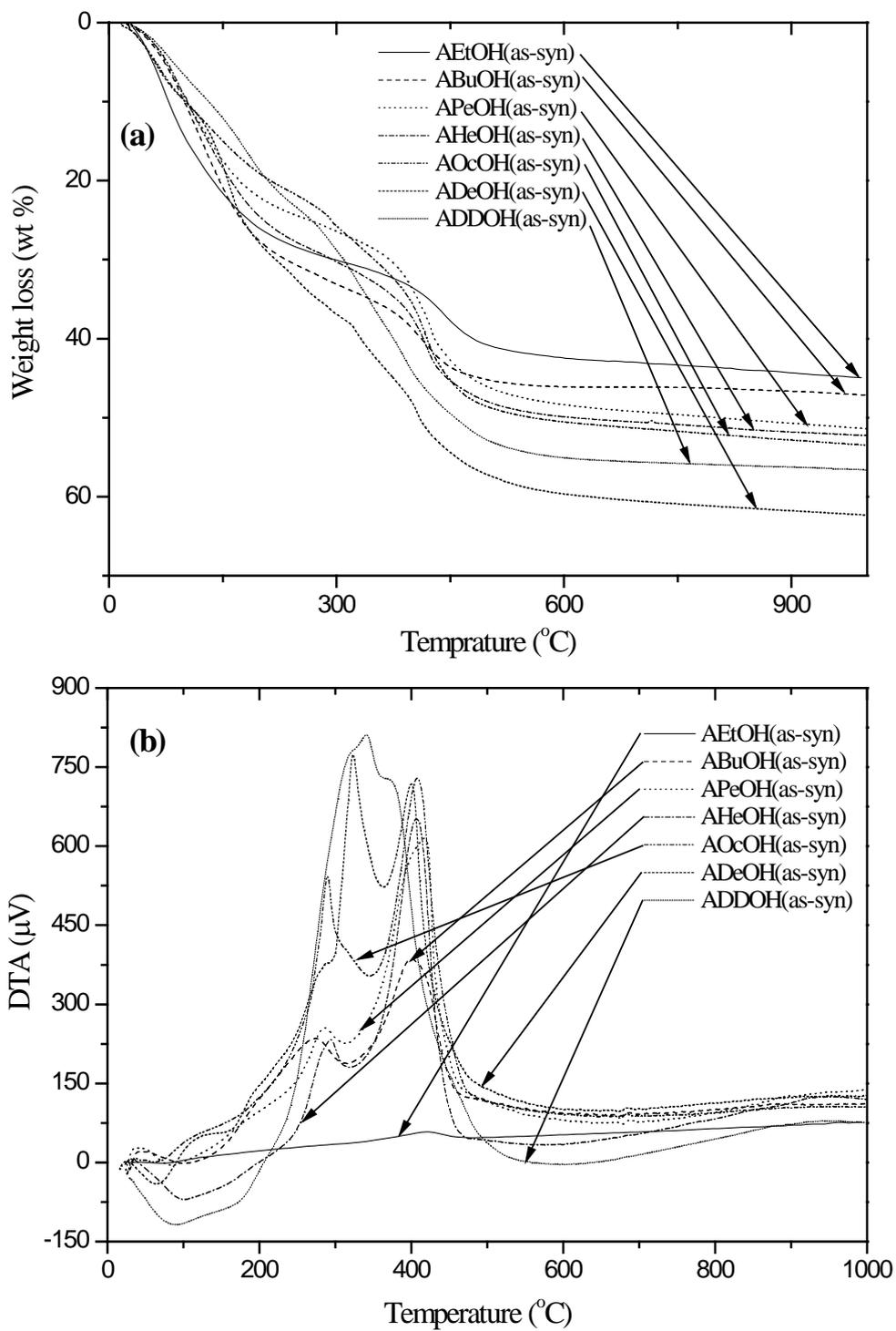


Fig. 3. Kim, et al.

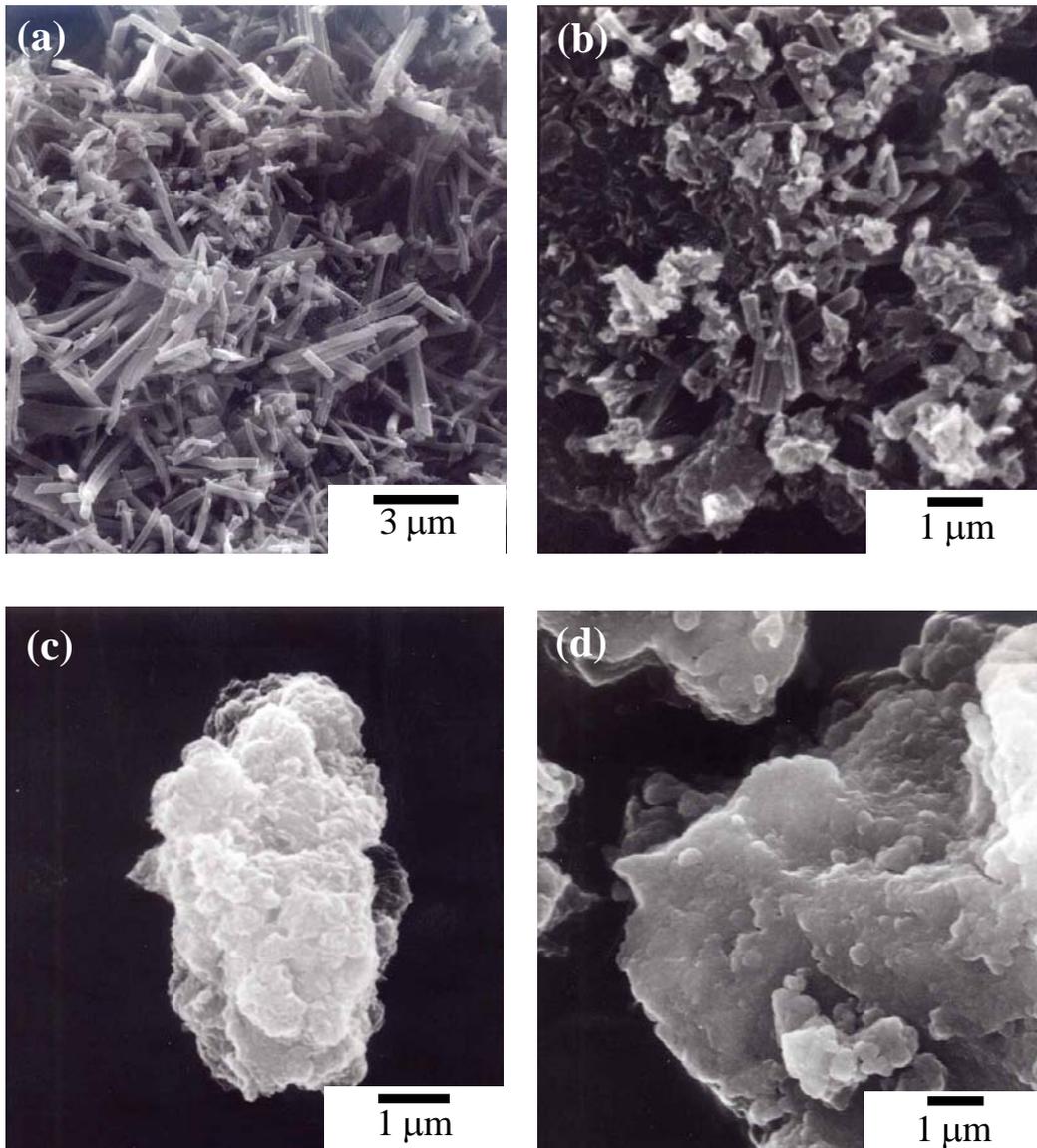


Fig. 4. Kim, et al.

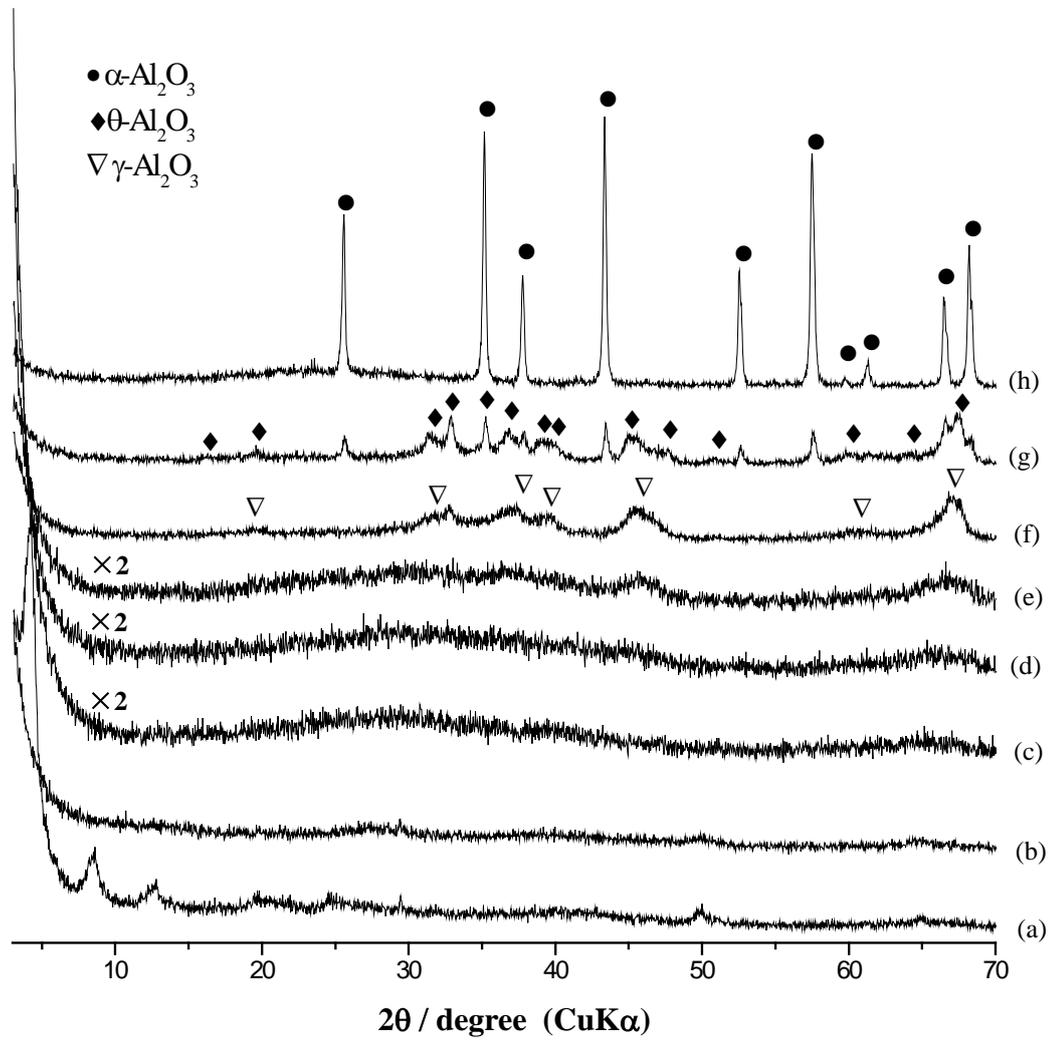


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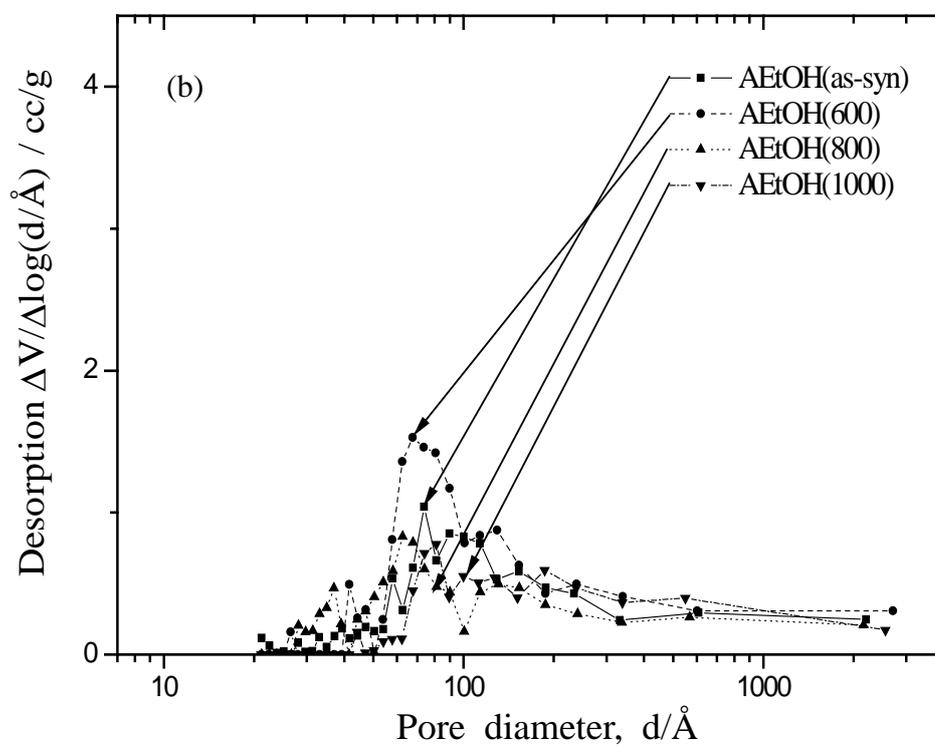
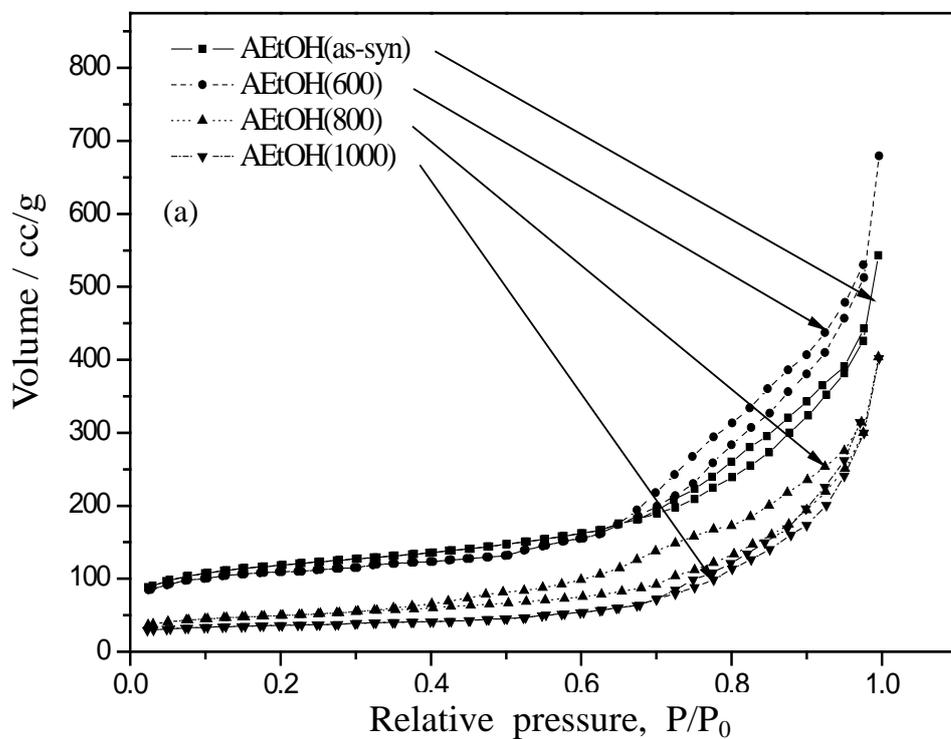


Fig. 6. Kim, et al.

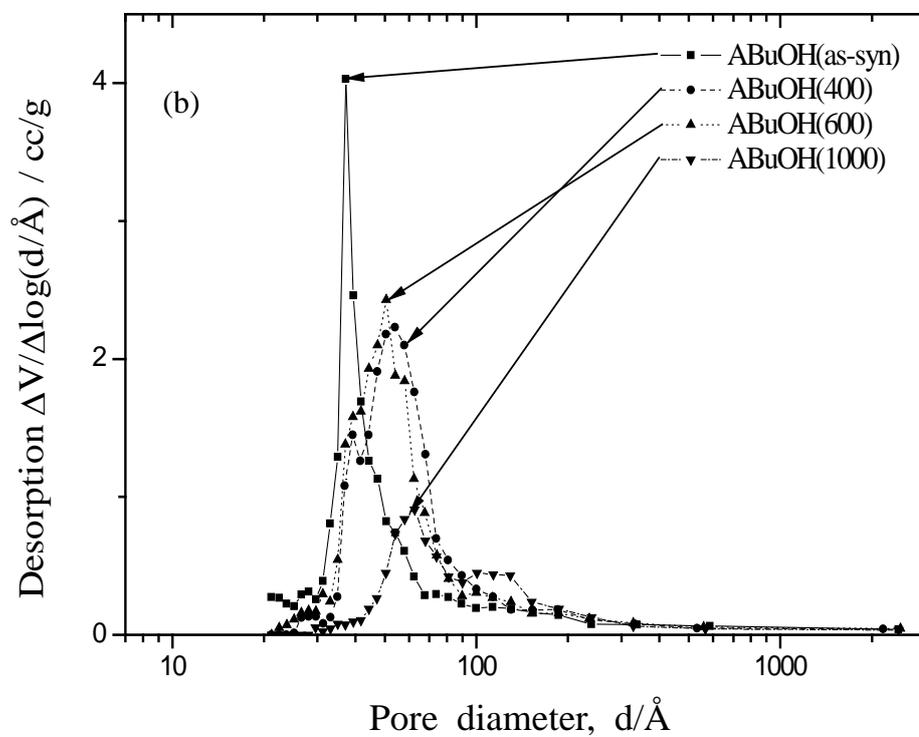
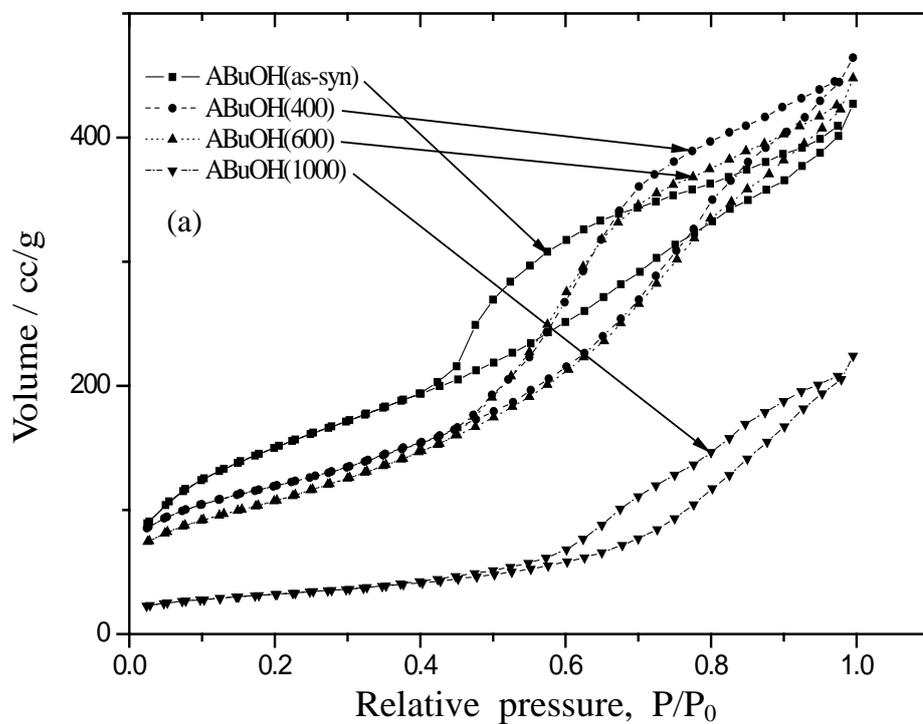


Fig. 7. Kim, et al.

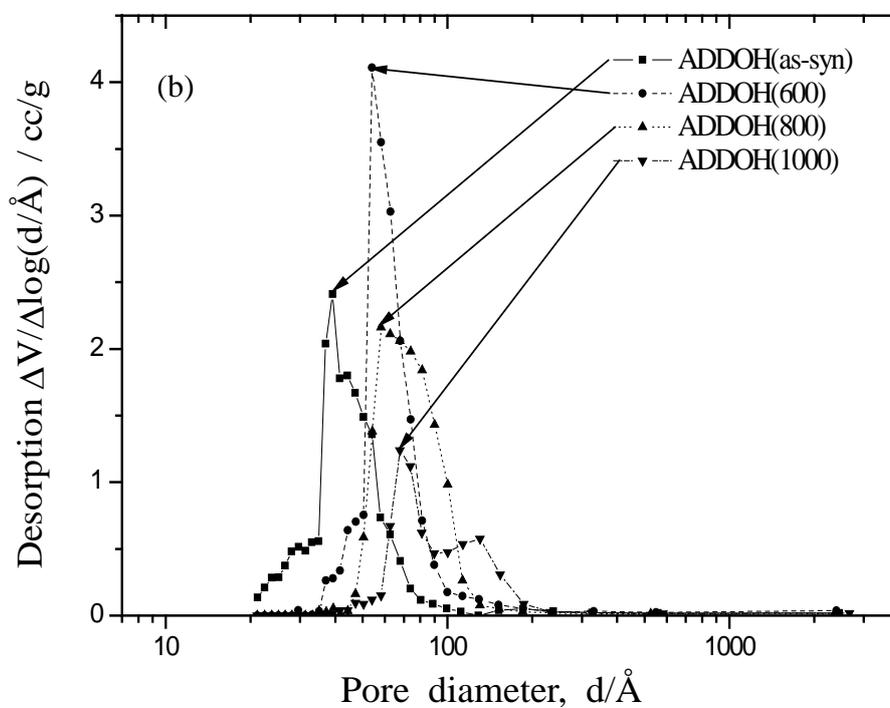
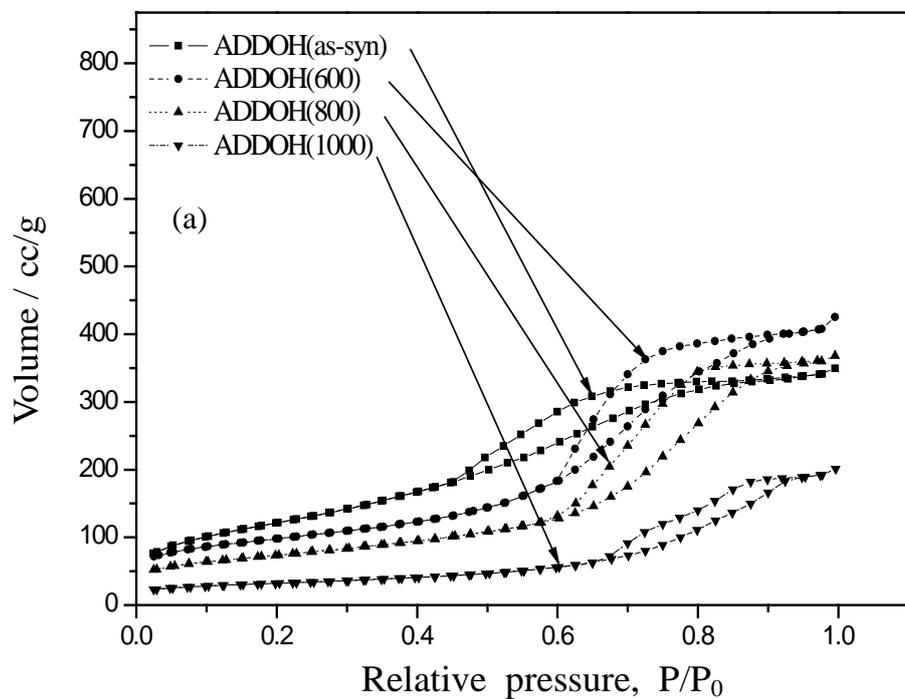


Fig. 8. Kim, et al.

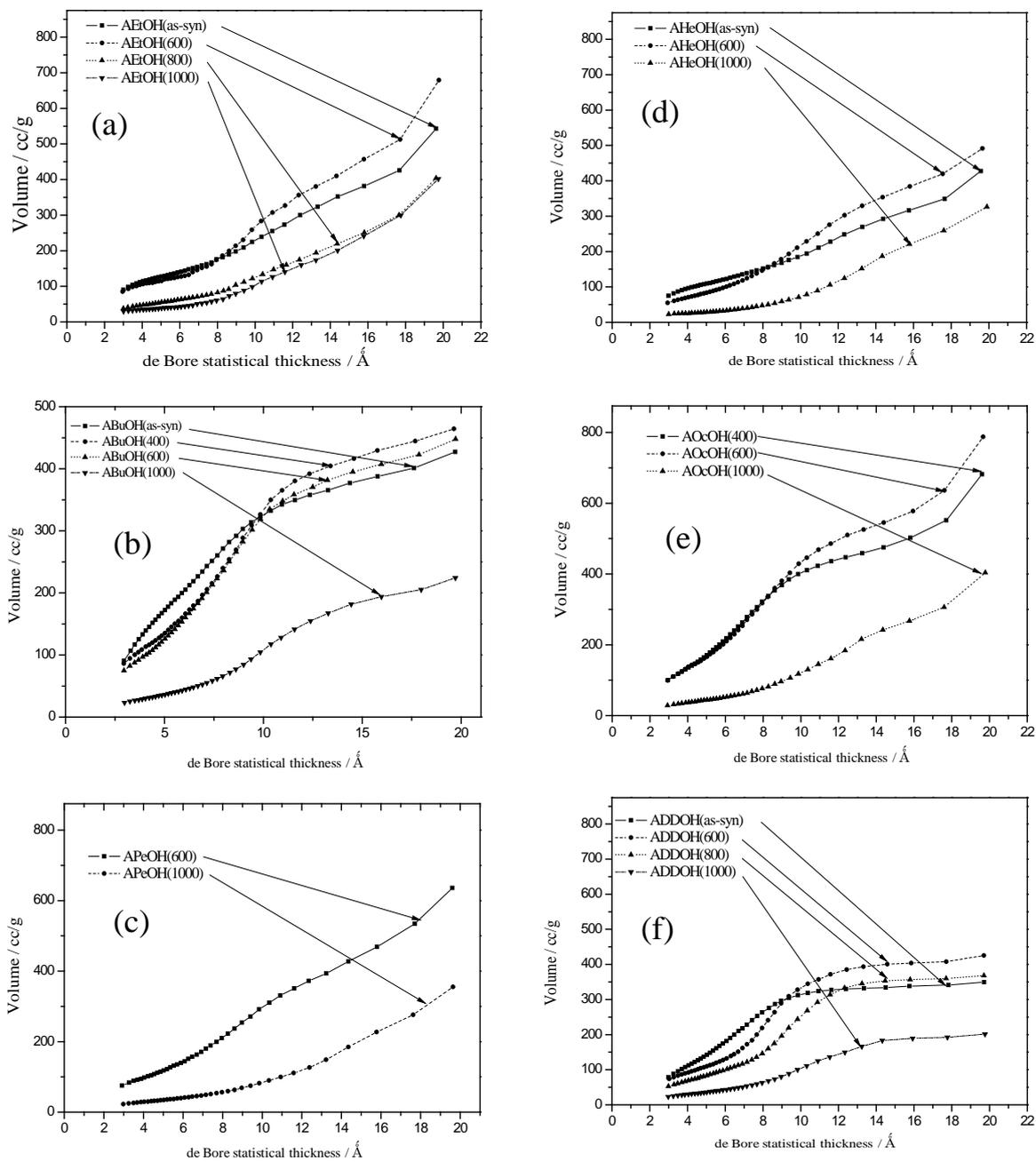


Fig. 9. Kim, et al.