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<td>Author(s)</td>
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<tr>
<td>Citation</td>
<td>Applied Physics Letters (2009), 95(12)</td>
</tr>
<tr>
<td>Issue Date</td>
<td>2009-09</td>
</tr>
<tr>
<td>URL</td>
<td><a href="http://hdl.handle.net/2433/87300">http://hdl.handle.net/2433/87300</a></td>
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<tr>
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<tr>
<td>Type</td>
<td>Journal Article</td>
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<td>Textversion</td>
<td>publisher</td>
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Kyoto University
Photocarrier recombination dynamics in highly excited SrTiO$_3$ studied by transient absorption and photoluminescence spectroscopy

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(Received 3 August 2009; accepted 5 September 2009; published online 25 September 2009)

We studied photocarrier recombination processes in highly excited SrTiO$_3$ crystals using pump-probe transient absorption (TA) and photoluminescence (PL) spectroscopy at room temperature. TA signals of nondoped SrTiO$_3$ crystals clearly appear in the visible and infrared spectral region under intense interband photoexcitation, and TA spectra show Drude-like photon-energy dependence. Both TA and PL decay curves are well explained by the same simple rate equation including three-body Auger recombination and single-carrier trapping. © 2009 American Institute of Physics. [doi:10.1063/1.3238269]

Over the past decade, perovskite oxides and their heterostructures have attracted a great deal of attention as unique device materials because they have a wide variety of fascinating and multifunctional properties. Among perovskite oxides, SrTiO$_3$ has been regarded as one of the most important materials because SrTiO$_3$ has been widely used as a substrate material for transparent oxide electronics. In addition, by electron doping, SrTiO$_3$ itself shows multifunctional electrical properties ranging from insulating to semiconducting, metallic, and superconducting. The interest to SrTiO$_3$ is further increased by unique electronic and magnetic properties of two-dimensional electron gases formed at the interfaces between SrTiO$_3$ and other oxides. Despite extensive studies of electronic structures and electrical properties of SrTiO$_3$ and related heterostructures, the carrier dynamics that determines electronic and optical properties remains unclear even in SrTiO$_3$ bulk crystals. Impurities and defects in SrTiO$_3$ crystals affect optical responses and the carrier recombination dynamics, and thus there have been so far no quantitative discussions on intrinsic carrier recombination processes.

Recent discovery of the blue photoluminescence (PL) in SrTiO$_3$ provides a chance to deep understand of carrier dynamics and electronic properties of SrTiO$_3$ crystals and SrTiO$_3$ based heterostructures. Broad blue PL at around 2.9 eV was reported in electron-doped SrTiO$_3$ (Ar$^+$-irradiated SrTiO$_3$) at room temperature, and similar blue PL was observed in nondoped SrTiO$_3$ under intense photoexcitation. Under intense photoexcitation, defect and impurity PL usually saturates, and intrinsic carrier recombination processes appear and determine the room-temperature blue PL dynamics. Thus, we anticipate that blue PL spectrum and lifetime are used as a probe for determining spatial profiles of carriers and defects. Study of the carrier dynamics by different evaluation methods is essential for understanding of the PL origins in SrTiO$_3$ and for quantitative characterization of SrTiO$_3$ thin films and heterostructures.

In this letter, we performed transient absorption (TA) and PL measurements in nondoped SrTiO$_3$ crystals under intense photoexcitation at room temperature. TA spectra in nondoped SrTiO$_3$ crystals are similar to linear absorption spectra of electron-doped SrTiO$_3$ crystals, originating from doped carriers in the samples. The carrier recombination dynamics evaluated from TA and PL experiments is well explained by a simple model involving the nonlinear Auger recombination and single-carrier trapping processes. From the analysis of the carrier decay times, we determine Auger recombination and single-carrier trapping coefficients.

We used nondoped SrTiO$_3$ and electron-doped SrTi$_{1-x}$Nb$_x$O$_3$ (Nb-doped SrTiO$_3$) single crystals with several dopant concentrations (commercially available from Furui Chemical Co.). Nondoped samples were annealed under oxygen flow for 24 h at 700 K to reduce oxygen vacancies. The samples were 0.5 mm thick. Time-resolved PL and TA measurements were performed using an optical parametric amplifier system based on a regenerative amplified mode-locked Ti:sapphire laser with a pulse duration of 150 fs and a repetition rate of 1 kHz. The excitation photon energy was 3.49 eV. The laser spot size on the sample surface was measured carefully using the knife-edge method. The femtosecond time-resolved TA measurements were performed by using a pump-probe technique. An aperture with 10 $\mu$m diameter was attached on the sample surface to ensure the homogeneous excitation laser intensity and the overlap of the pump and probe pulses. In temporal profile measurements, the photon energy of the probe pulse was 1.55 eV and a photodiode with a lock-in amplifier was used for detection. In TA spectral measurements, we used white light as a probe pulse obtained by focusing 1.55 eV laser pulse on the sapphire plate, and used a charge coupled device and an InGaAs diode array for detection. We also measured TA decay curves in the nanosecond time region using a photodiode and a digital oscilloscope. The probe light source was a continuous-wave diode laser whose photon energy was 1.57 eV and the time resolution of this experimental setup was approximately 2 ns. In time-resolved PL measurements, we used a streak camera and a monochromator, where the time resolution of our setup was 40 ps. All measurements were performed at room temperature. While the PL intensity is determined by the product of the electron and hole densities, the intraband TA signal directly reflects the electron or hole density. From both TA and PL measurements, we discuss...
photocarrier decay dynamics in highly excited SrTiO$_3$ in more detail.

The inset of Fig. 1 shows the linear absorption spectra of nondoped SrTiO$_3$ and electron-doped SrTiO$_3$ (Nb-doped SrTiO$_3$) with several doping concentrations. In nondoped SrTiO$_3$, there is no optical absorption below the band gap of 3.2 eV in the visible region. The 0.16 eV peak is assigned to an electronic transition of the defects. On the other hand, electron-doped SrTiO$_3$ shows a Drude-like broad absorption band in the visible and infrared region. The spectrum shape of electron doped SrTiO$_3$ is dependent on the doped carrier density. The relatively narrow 2.4 eV band is assigned to optical transition from a deep impurity level to the conduction band.

Figure 1 shows TA spectra of nondoped SrTiO$_3$ at the delay times of 5 ps and 2 ns under 1.0 mJ/cm$^2$ excitation. Just after laser excitation, a broad TA band appears in the infrared and visible region above the band-gap energy. The 0.16 eV peak is assigned to an electronic transition of the defects. On the other hand, electron-doped SrTiO$_3$ shows a Drude-like broad absorption band in the visible and infrared region. The spectrum shape of electron doped SrTiO$_3$ is dependent on the doped carrier density. The relatively narrow 2.4 eV band is assigned to optical transition from a deep impurity level to the conduction band.

Figure 1 shows TA spectra of highly excited SrTiO$_3$ at the delay times of 5 ps and 2 ns under 1.0 mJ/cm$^2$ excitation. Just after laser excitation, a broad TA band appears in the infrared and visible region above the band-gap energy. The 0.16 eV peak is assigned to an electronic transition of the defects. On the other hand, electron-doped SrTiO$_3$ shows a Drude-like broad absorption band in the visible and infrared region. The spectrum shape of electron doped SrTiO$_3$ is dependent on the doped carrier density. The relatively narrow 2.4 eV band is assigned to optical transition from a deep impurity level to the conduction band.

Figure 2 shows temporal changes in TA signals at different excitation densities of 0.5, 4.4, 8.9, and 13.3 mJ/cm$^2$. Since the broad TA band has no fine structures in the range of 1–2 eV, we used the 1.55 eV laser as a probe pulse for temporal changes of the carrier density. Under weak excitation of 0.5 mJ/cm$^2$, the TA signal shows very slow decay and the decay time is much longer than 2 ns. As shown in the inset of Fig. 2, the TA signal shows a single exponential decay with a decay time of 60 ns under the weak excitation condition of 0.8 mJ/cm$^2$. Under high excitation densities above about 3 mJ/cm$^2$, the fast and nonexponential decay component clearly appears in the subnanosecond time region, and the TA decay time becomes faster as increasing the excitation density.

We also studied the PL decay dynamics of nondoped SrTiO$_3$ for comparison with TA studies under the same experimental conditions. Figure 3(a) shows PL decay curves of nondoped SrTiO$_3$ in the subnanosecond time region under excitation densities of 1.5, 3.8, 5.2, and 5.9 mJ/cm$^2$. The monitored PL photon energy was 2.9 eV, which is the peak energy of the broad PL spectrum as shown in the inset of Fig. 3(b). Under low excitation density, the PL dynamics shows slow decay. As shown in the inset of Fig. 3(a), the PL decay dynamics shows almost single exponential decay with the decay time of 29 ns under weak excitation density of 0.4 mJ/cm$^2$. As increasing the excitation density, fast and nonexponential decay component appears, and the PL decay time becomes shorter. These PL decay behaviors are quite similar to TA decay ones. Figure 3(b) shows the excitation-intensity dependence of the PL intensity just after excitation: $I_0$. The PL intensity shows quadratic dependence on the excitation density. This means that the blue PL originates from the two-body (bimolecular) recombination of photocarriers, because excitonic states are unstable due to the large dielectric constant ($\epsilon \sim 300$ at 300 K).

The excitation-density dependence of the TA and PL decay times is summarized in Fig. 4. Here, we define the effective decay time $t_{1/e}$ as the time at the carrier density of $n_0/e$. The initial photogenerated carrier density $n_0$ is estimated from the incident photon number and the optical absorption coefficient at the excitation photon energy in Ref. 18. Note that $I_{PL}(t_{1/e}) = I_{PL}(0)/e^2$ and $\alpha_{TA}(t_{1/e}) = \alpha_{TA}(0)/e$ are used because of the relations $I_{PL} \sim n^2$ and $\alpha_{TA} \sim n$, where $I_{PL}$ and $\alpha_{TA}$ are the PL and TA densities. We plot $t_{1/e}$ derived from PL (denoted by circle) and TA (square) decay curves as a function of the excitation density in Fig. 4. The effective decay times $t_{1/e}$ of TA and PL signals show good accordance.
Here, we discuss the carrier dynamics in highly photoexcited SrTiO$_3$ using the rate equation of photocarriers. In semiconductors, the rate equation for the photocarriers can be simplified and written as follows:\textsuperscript{13,19}

\[
\frac{dn}{dt} = -An - Bn^2 - Cn^3, \tag{1}
\]

where $n$ is the photocarrier density. $An$ represents the single-carrier trapping. $Bn^2$ consists of bimolecular radiative recombination process and nonradiative trap-Auger recombination process.\textsuperscript{10,20} $Cn^3$ represents the band-to-band (intrinsic) Auger recombination process, involving electron-electron-hole or electron-hole-hole processes, where the nonradiative recombination energy of the electron-hole pairs is transferred to the kinetic energy of the other electron (hole). According to Eq. (1), the effective decay time $t_{1/e}$ is approximately written as

\[
t_{1/e} = (A + Bn_0 + Cn_0^2)^{-1}. \tag{2}
\]

We fit the decay times of TA and PL dynamics shown in Fig. 4 by using Eq. (2). The best fit-parameters are $A=1.7 \times 10^7$ s$^{-1}$ and $C=1.2 \times 10^{-32}$ cm$^6$ s$^{-1}$ for TA dynamics, and $A=1.7 \times 10^7$ s$^{-1}$ and $C=1.2 \times 10^{-32}$ cm$^6$ s$^{-1}$ for PL dynamics. Here, we confirm that $Bn^2$ is negligibly small compared to $An$ and $Cn^3$. In fact, the solid calculated curves shown in the Fig. 4 ($B=0$) reproduce the experimental results very well. This result means that the recombination rate of two-body processes (i.e., the radiative bimolecular recombination and trap Auger recombination) is very small in SrTiO$_3$. This result is consistent with the low PL quantum efficiency (PL quantum efficiency $<0.01$). The calculated TA and PL decay curves using the above best-fit-parameters also reproduce the experimental decay curves as shown in Figs. 2 and 3(a). From the TA and PL measurements and the calculation using the same equation, we determine the coefficients, $A=1.7 \times 10^7$ s$^{-1}$ and $C=(1.5 \pm 0.3) \times 10^{-32}$ cm$^6$ s$^{-1}$. No difference between TA and PL decay times clearly shows that the intrinsic Auger recombination process determines the photocarrier dynamics under high excitation density. The microscopic carrier recombination mechanisms determining $A$ and $C$ coefficients remain an open question.

In conclusion, we studied TA and PL decay dynamics in highly excited SrTiO$_3$ at room temperature. Both PL and TA decay curves are well explained by the same rate equation. We analyzed the excitation-density dependence of the TA and PL decay times and determined single-carrier trapping and Auger recombination coefficients.

Part of this work was supported by a Grant-in-Aid for Scientific Research on Innovative Area “Optical Science of Dynamically Correlated Electrons” (Grant No. 20104006) from MEXT, Japan.